

IN VITRO STUDIES OF BONE-CEMENT INTERFACE AND RELATED WORK ON CEMENTED ACETABULAR REPLACEMENT

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The thesis is submitted in fulfilment of the requirements for the award of the degree of

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Abstract

The lasting integrity of the bond between bone cement and bone defines the long-term stability of cemented acetabular replacements. Although several studies have been carried out on bone-cement interface at continuum level, micromechanics of the interface has been studied only recently for tensile and shear loading cases. Furthermore, the mechanical and microstructural behaviour of this interface is complex due to the variation in morphology and properties that can arise from a range of factors. In this work in vitro studies of the bone-cement interfacial behaviour under selected loading conditions were carried out using a range of experimental techniques.

Damage development in cemented acetabular reconstructs was studied under a combined physiological loading block representative of routine activities in a saline environment. A custom-made environmental chamber was developed to allow testing of acetabular reconstructs in a wet condition for the first time and damage was monitored and detected by scanning at selected loading intervals using micro-focus computed tomography (μ CT). Preliminary results showed that, as in dry cases, debonding at the bone-cement interface defined the failure of the cement fixation. However, the combination of mechanical loading and saline environment seems to affect the damage initiation site, drastically reducing the survival lives of the reconstructs.

Interfacial behaviour of the bone-cement interface was studied under tensile, shear and mixed-mode loading conditions. Bone-cement coupons were first mechanically tested and then μ CT imaged. The influence of the loading angle, the extent of the cement penetration and the failure mechanisms with regard to the loading mode on the interfacial behaviour

were examined. Both mechanical testing and post failure morphologies seem to suggest an effect of the loading angle on the failure mechanism of the interface. The micromechanical performance of bone-cement interface under compression was also examined. The samples were tested in step-wise compression using a custom-made micromechanical loading stage within the μ CT chamber, and the damage evolution with load was monitored. Results showed that load transfer in bone-cement interface occurred mainly in the bone-cement contact region, resulting in progressively developed deformation due to trabeculae bending and buckling.

Compressive and fatigue behaviour of bovine cancellous bone and selected open-cell metallic foams were studied also, and their suitability as bone analogous materials for cemented biomechanical testing was investigated. Whilst the morphological parameters of the foams and the bone appear to be closer, the mechanical properties vary significantly between the foams and the bone. However, despite the apparent differences in their respective properties, the general deformation behaviour is similar across the bone and the foams. Multi-step fatigue tests were carried out to study the deformation behaviour under increasing compressive cyclic stresses. Optical and scanning electron microscopy (SEM) were used to characterise the microstructure of foams and bone prior to and post mechanical testing. The results showed that residual strain accumulation is the predominant driving force leading to failure of foams and bones. Although foams and bone fail by the same mechanism of cyclic creep, the deformation behaviour at the transient region of each step was different for both materials. Preliminary results of foam-cement interface performance under mixed-mode loading conditions are also presented.

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Declaration

Whilst registered as a candidate for the above degree, I have not been registered for any other research award. The results and conclusions embodied in this thesis are the work of the named candidate and have not been submitted for any other academic award. Most of the results presented in this thesis have already been published.

Signed:

Date:

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Dedication

To my family

Chapter 1

Introduction

1. Introduction

With the increase in life expectancy and a growing number of younger active patients receiving joint replacement treatment, there is an urgent need to revisit cemented total hip replacements (THR) procedures in order to improve the survival lives of cement fixation beyond 10 to 20 years, so that the need for revision surgeries can be drastically reduced. In the UK, according to the National Joint Registry (NJR Annual Report 2011, UK), of the 68,907 primary hip replacement procedures undertaken in 2010, 36% were cemented THRs.

Aseptic loosening has been identified as the most common symptom of long-term instability leading to gross migration of the implants and failure of THRs (Stocks et al., 1995; Thanner et al., 1999). Many theories have been proposed to explain the loosening process (Huiskes et al., 1993), although its multifactorial etiology is generally accepted now, where both biological and mechanical events are involved in the failure of the fixation (Huiskes et al., 1993; Sundfeldt et al., 2006).

Although late loosening of cemented acetabular cups has been reported to be three times that of femoral components 20 years after operation (Shulte et al., 1993), few studies have been reported with regard to the failure of acetabular fixation under long-term physiological loading conditions (Heaton-Adegbile et al., 2006; Zant et al., 2007, 2008; Tong et al., 2008; Wang et al., 2009). All the previous in vitro studies, however, have been conducted in dry conditions, where the effect of environment on the fixation integrity was not considered.

To achieve a good fixation in cemented implants, the mechanical interlock between the bone cement and the cancellous bone is considered of vital importance (Krause et al., 1982; Bugbee et al., 1992; Majkowski et al., 1993; Mann et al., 1997), and cementing techniques

such as pressurization were found useful in increasing the mechanical interlocking and stability of the interface (Oates et al., 1995; Flivik et al., 2005).

Numerous experimental studies have been carried out on bone-cement interfacial strength under tensile (Mann et al., 1997), shear (Mann et al., 1999), mixed mode (Mann et al., 2001) and creep-fatigue (Kim et al., 2004) loading conditions. Recently, the micromechanical behaviour of bone-cement interface was investigated under tension (Mann et al., 2008), tension-compression (Miller et al., 2009), shear fatigue (Mann et al., 2009) and multi-axial (Miller et al., 2011) loading conditions. However, the local deformation and damage development at the interface were only captured on the exterior surfaces of the samples hence the analysis was limited to a 2D investigation, whilst internal volume responses could not be interrogated.

Biomechanical testing of orthopaedic implants in cadaver bones has provided valuable information on pre-clinical assessment of implant performance. However, the method is limited by the availability of cadaveric tissues. Furthermore, the reproducibility of such test results is generally poor, because of the large variation in the mechanical properties of cancellous bones due to a number of variables such as anatomic site, disease state and age (Goldstein, 1987; Linde, 1994; Keaveny & Hayes, 1993; Keaveny et al., 2001; Gibson, 2005). For this reason it is desirable, in orthopaedic related research, to employ bone substitutes in mechanical characterisation and testing in order to remove some of the variables in cadaver tissues; and, furthermore, to simulate a range of cancellous bones properties by controlling the morphological parameters in bone analogue materials.

The aim of this work is to provide a better understanding of how the bone-cement interfacial behaviour from a macro level (acetabular reconstructs) to a micro level (bone-cement interface coupons) is related to environmental (i.e. wet conditions, temperature), structural (i.e. depth of cement penetration, contact area, bone volume fraction) and operational (i.e. loading condition) parameters. Open-cell metallic foams suitability, as cancellous bone substitutes for cemented biomechanical characterisation of the interface, is also explored. This aim was realised through the following objectives:

- The fatigue behaviour of the bone-cement fixation in acetabular replacements under physiological loading conditions in a saline environment was studied by in vitro testing on a specially designed hip simulator, through a new experimental protocol.
- The bone-cement interfacial behaviour under mixed-mode loading conditions and its micromechanical compressive response were evaluated experimentally by means of experimental techniques, including μ CT examination and image-guided failure assessment (IGFA).
- Selected metallic open-cell foams were mechanically and morphologically characterised under uniaxial and cyclic loading conditions, and the results compared with those obtained from bovine cancellous bone. Preliminary testing of foam-cement interface was carried out under mixed-mode loading conditions.

Chapter 2 provides some background knowledge to the present work and a review of the relevant literatures. Chapter 3 presents the results of fatigue testing on the hip simulator with a new environmental chamber to study the long-term mechanical response of cemented acetabular reconstructs under a combined loading block in a wet environment. In Chapter 4 the mechanical performance of the bone-cement interface under mixed-mode

loading conditions and its compressive micromechanical behaviour are reported. Chapter 5 presents the mechanical and morphometrical characterisation of open-cell metallic foams to determine their suitability as cancellous bone substitutes for cemented biomechanical applications. Furthermore, foam-cement interface coupons were prepared and a preliminary determination of their mechanical performance under mixed-mode loading conditions is given. Chapter 6 presents a final discussion, the main conclusions and suggests some ideas for future studies.

Chapter 2

Literature Review

2.1. Bone and bones

2.1.1. The human skeleton

The skeletal tissues, such as cartilage and bone, are highly specialised type of connective tissues which have a primary role of supporting structure (Cowin, 2001; Williams, 1995). The main constituent of the skeleton is bone (Fig. 2.1), which is different from the other connective tissues (i.e. cartilage, ligaments and tendons) due to the high mineralization of the extra cellular matrix. Thus, the main characteristics of this type of human tissue are rigidity and hardness that ensure the necessary stiffness of the skeleton to maintain the shape of the body, to transmit muscular forces through the body during movement, to protect the soft tissues of the cranial, thoracic and pelvic cavities and to supply the framework for the bone marrow.

2.1.2. Bone morphology

According to their shape, bones can be divided into different families (Williams, 1995):

- Long bones, such as femur, tibia, humerus;
- Short bones, such as carpus, tarsus;
- Flat bones, such as cranial vault, scapulae;
- Irregular bones, any bones not easily assigned to one of the former groups.

A typical example of the macroscopic morphology of bone can be observed in long bones. As shown in Fig. 2.2, they consist of a cylindrical shaft (or diaphysis) and two wider and rounder ends, also called epiphyses. Conical regions, or the metaphyses, connect the diaphysis with the epiphyses. Most long bones have the ends wider than their central part, with the joints covered by articular cartilage.

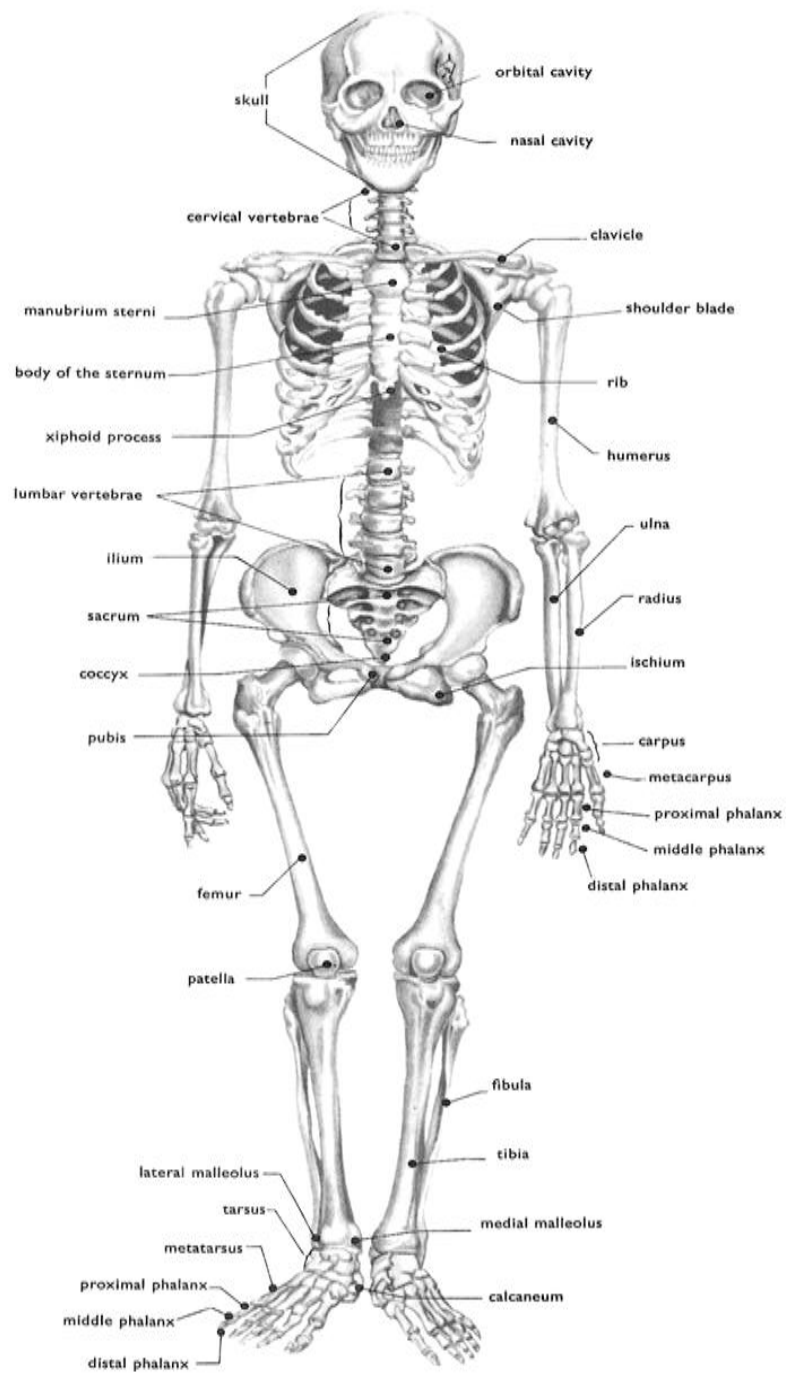


Figure 2. 1: A human adult skeleton: frontal view
 (<http://www.daviddarling.info/encyclopedia/S/skeleton.html>).

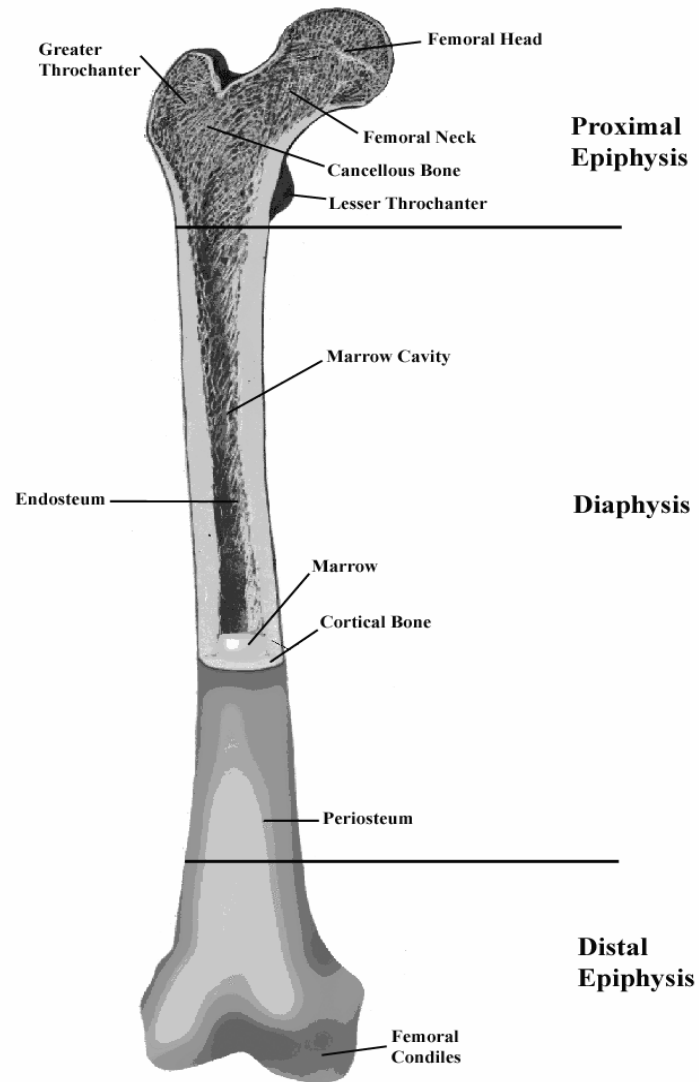


Figure 2. 2: Schematic representation of a human femur (Varini, 2007).

2.1.3. Bone microstructure

Bone composition

Bone consists of approximately 65% mineral and 35% organic matrix, cells and water. Bone mineral, usually presented in the form of small crystals, is placed between the collagen fibers. Bone mineral is mainly impure hydroxyapatite, $\text{Ca}_6(\text{PO}_4)_6(\text{OH})_2$, containing carbonate, citrate, fluoride and strontium. The organic matrix consists of 90% collagen and about 10%

non collagenous proteins. From a mechanical point of view the bone matrix is comparable to a composite material: the organic component is responsible for providing the toughness, while the inorganic component provides the stiffness (Cowin, 2001).

Woven and lamellar bone

Mammalian bone is divided into two main groups: woven and lamellar. In a developing embryo, the bone tissue is of woven type. It is a matrix of interwoven coarse collagen fibres, containing osteocytes distributed in an approximate random way. Woven bone can be seen as a provisional material, eventually resorbed and replaced by lamellar bone. Adult bone is mainly composed of lamellar bone, which is built up of 3µm to 7µm thick unit layers (lamellae) containing collagen fibres distributed forming a parallel pattern with each other (Cowin, 2001).

Cortical and trabecular bone

Adult bone is composed of two basic structures (Cowin, 2001), called cortical (or compact) and trabecular (or cancellous) bone (Fig. 2.3). Cortical bone is a solid and compact material, with the exception of microscopic canals (Haversian canals). Approximately 80% of the skeletal mass in the adult skeleton is composed by cortical bone. It forms the outer wall of all bones, and is mainly responsible for the supportive and protective functions of the skeleton. The remaining 20% of the bone mass is trabecular bone, which is found in the internal parts of the skeleton and appears as a lattice of plates and rods, called trabeculae. A typical thickness for the trabeculae can range from 100µm to 350µm. Trabecular bone is more compliant and weaker than cortical bone due to its porous structure. Its mechanical properties are more varied compared to those of cortical bone due to its microstructure

irregularity. However, trabecular bone has very important roles where it 1) stiffens the structure by connecting the outer shell of cortical bone; 2) supports the cortex layer and distributes loads; and 3) protects the cortical bones from instability (buckling).

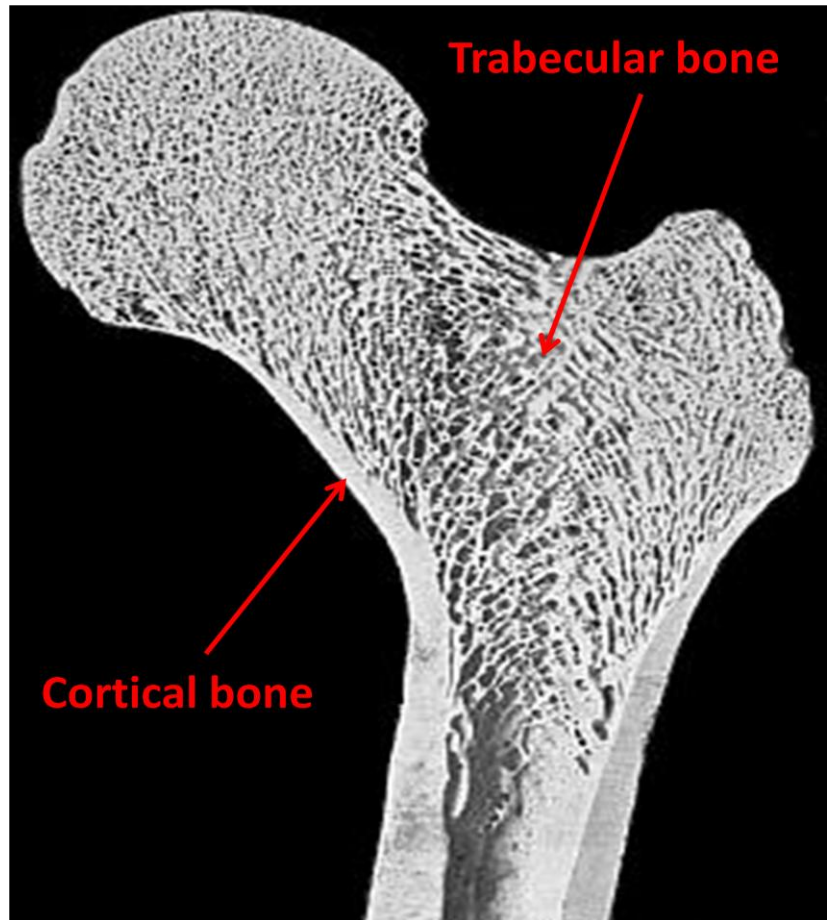


Figure 2. 3: Trabecular and cortical bone distribution in the proximal femur
(<http://www.ilsb.tuwien.ac.at/~daxner/ictam-projekt/my-femur-section-small.jpg>).

Bone is limited to appositional growth by the non-expandable nature of its mineralized bone tissue (Cowin, 2001), for this reason all bone activities occur at the surfaces. Bone tissue has two principal surfaces: periosteal and endosteal. The periosteum covers the external bone surface, and is composed of a sheet of fibrous connective tissue and an inner layer of undifferentiated cells, with the potential to form bone during growth and fracture healing.

The periosteum is absent in regions where tendons or ligaments are attached to the bone, on bone ends covered with articular cartilage, in subscapular areas and in the neck of the femur. The endosteum is a thin cellular layer of bone surface cells (osteoclasts, osteoblasts, and bone lining cells), that covers the marrow cavity of the diaphysis and the cavities of cortical and trabecular bone.

Cortical bone

The main structural unit of cortical bone is the osteon or Haversian system (Fig. 2.4).

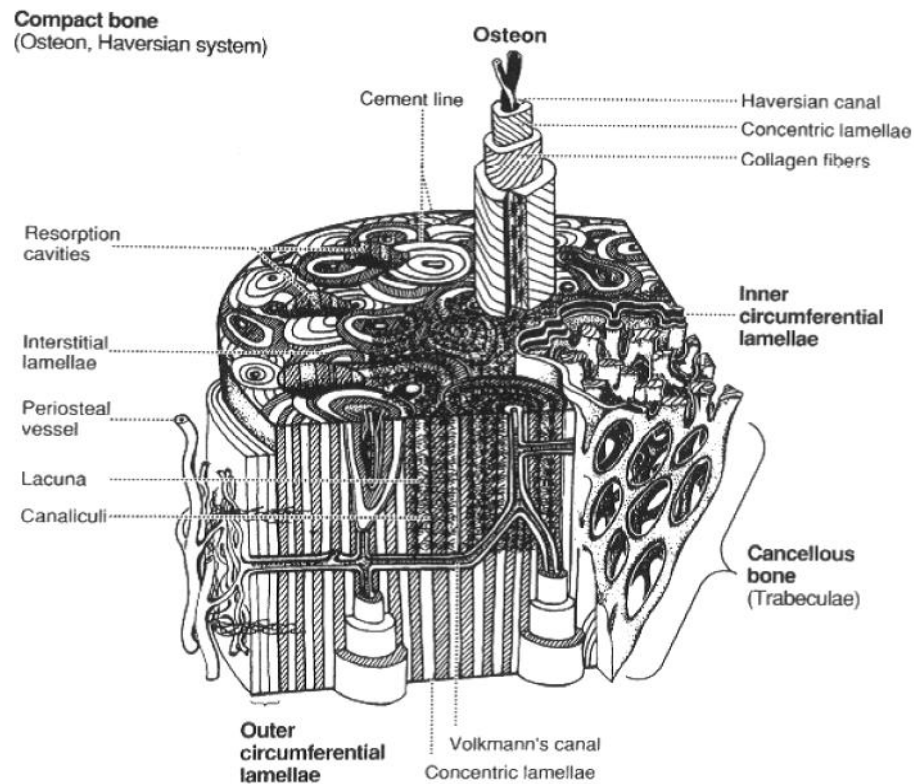


Figure 2. 4: A schematic of a long bone shaft portion, showing details of cortical bone structure (Cowin, 2001).

A typical osteon is a cylinder about 200µm in diameter, consisting of a central canal (Haversian canal) surrounded by about 20-30 concentric lamellae. The cortical bone presents

several layers of osteons in the shaft of long bones. In the gaps between osteons interstitial lamellae appear as angular fragments of previous concentric and circumferential lamellae. Blood flows through the Haversian canals where lymphatics vessels and nerves are running. Transverse canals, also called Volkmann canals, have the function to interconnect the Haversian canals allowing the communication between periosteum and bone marrow. The outer border of each osteon is surrounded by a cement line, which is a thick layer of mineralized matrix with dimensions ranging from 1µm to 2µm, deficient in collagen fibres. Throughout the bone, small cavities (lacunae) containing entrapped bone cells (osteocytes) are visible. Microscopic tubular canals (canaliculi) connect the lacunae to each other and to the Haversian canals (Cowin, 2001).

Trabecular bone

Trabecular bone is characterised by the absence of Haversian systems, and consists of an array of interconnected rods and plates (trabeculae). The trabeculae are in general of a thickness less than 0.2mm and are variable in shape (Fig. 2.5). The trabecular tissue is formed by a mosaic of angular segments of parallel sheets of lamellae, preferentially aligned with the orientation of the trabeculae. The angular grouping of lamellae are called trabecular packet and define the structural unit of trabecular bone. As with cortical bone, cement lines hold the trabecular packets together (Ohman, 2011).

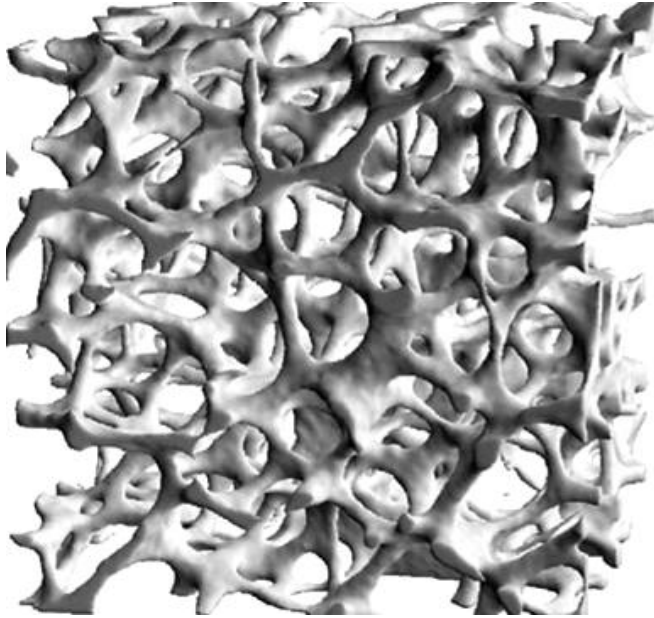


Figure 2. 5: Three-dimensional trabecular bone architecture of a lumbar spine bone biopsy (Nazarian & Muller, 2004).

There are great differences in the apparent density of trabecular bone in different locations in the human skeleton, and this is the main reason why it is possible to find trabecular bone with very different characteristics. Moreover the orientation of the trabeculae is related to the anatomical site and aligned to the loading regime experienced locally. Hence there is a clear dependence between the mechanical properties of trabecular bone and the trabecular orientation (Ohman et al., 2007). The trabecular structure is mainly oriented along the primary loading direction, clearly defined as main trabecular direction (MTD). However, in a joint, the loading direction varies during routine activities and may be affected by local anatomy, muscle forces and subject-specific gait, resulting in complex trabecular structures.

2.1.4. Mechanical properties of cancellous bone

According to the main aim of this thesis and the materials used in the experimental work, the principal emphasis will be placed on the trabecular bone properties. The reported mechanical properties of cancellous bone exhibit a high degree of variation due to many factors such as anatomical site, age and disease state, not to mention its viscoelastic and anisotropic nature. The measured values of strength and modulus also depend on the experimental arrangements used for testing (i.e. direction of loading and strain rate of testing), as reviewed by Keaveny & Hayes (1993) and Keaveny et al. (2001).

Cancellous bone also exhibits strength asymmetry (difference in compressive and tensile strength), with a lower strength in tension compared with compression, and still lower in shear (Keaveny et al., 2001). Some reported values for elastic modulus and strength of cancellous bone tested under tension and compression are shown in Table 2.1.

Ford & Keaveny (1996) measured the shear failure properties of bovine tibial trabecular bone, where specimens were loaded to failure in torsion longitudinally or transversely relative to the primary trabecular orientation (MTD). The difference in ultimate stress for longitudinal ($6.35(\pm 2.00)$ MPa) and transverse ($4.92(\pm 1.48)$ MPa) is a clear indication of the typical cancellous structure anisotropy.

Table 2. 1: Mechanical properties of cancellous bone tested under tension and compression from selected sources.

Reference	Anatomic site	Tensile strength (MPa)	Compressive strength (MPa)	Elastic modulus (MPa) (\pm SD)
Rohlmann et al. (1980)	Human femur	NR	6.8 ± 4.8	441 ± 271
Stone et al. (1983)	Bovine humerus	2.63	8.29 ± 1.82	NR
Mosekilde et al. (1987)	Human vertebra	NR	2.4 ± 16	67 ± 45
Linde & Hvid (1989)	Human tibia	NR	5.3 ± 2.9	445 ± 257
Swartz et al. (1991)	Calf lumbar sacral	NR	7.1 ± 3.0	173 ± 97
Poumarat & Squire (1992)	Calf femoral condyles	NR	8.52 ± 2.79	117 ± 61.53
Keaveny et al. (1994)	Bovine tibiae	16.9 ± 4.24	23.6 ± 8.28	2630 ± 727 (Tension) 2380 ± 777 (Compression)
Kalpan et al. (1997)	Bovine humerus	7.60 (CI=2.20)	12.4 (CI=3.20)	NR
Kopperdhal et al. (1998)	Human vertebrae	2.23 ± 0.76	2.23 ± 0.95	301 ± 100 (Tension) 291 ± 113 (Compression)
Majumdar et al. (1998)	Human calcaneus	NR	1.4 ± 1.3	68 ± 84
	Human femur		2.5 ± 1.9	130 ± 117
	Human vertebrae		1.3 ± 1.0	66 ± 63
Uchiyama et al. (1999)	Human vertebrae	NR	NR	126 ± 97
Lim et al. (2000)	Bovine vertebrae	NR	NR	220.5 ± 130.7
Guedes et al. (2004)	Bovine femurs	NR	15.74 ± 9.59	310 ± 161
Hing (2004)	Multiple sites	NR	3.6-9.3	260-900
Thompson et al. (2004)	Human acetabular roof	NR	NR	117 ± 88
	Human femoral head			47.4 ± 29.5
Ohman et al. (2007)	Human femoral head	NR	18 ± 6.4	2730 ± 1060
Thumer et al. (2007)	NR	NR	6.1 ± 1.6	104 ± 30
Guillen et al. (2011)	Bovine iliac crest	NR	9.36 ± 1.45	800 ± 138

NR: Not Reported

The properties of cancellous bone can be defined at the apparent level or at the tissue level.

The apparent level considers bone as a continuum with the average properties of a representative bone volume. This definition was introduced by Harrigan et al. (1988) for a cube with a minimum of five trabecular lengths, which equates to approximately 5mm. The properties of the individual trabeculae are defined at tissue level.

Although the tissue-level elastic properties of trabecular bone have been reported in numerous studies (Cowin, 2001), there remain substantial discrepancies in these data and

there are almost no data available for tissue-level failure properties. Nanoindentation technique is extensively used in materials science for probing the mechanical properties of thin films, small volumes and small microstructural features. Since many of these features of interest in bone are several micrometres or more in dimension, nanoindentation provides a direct and accurate determination of the elastic properties at tissue level (Rho et al., 1997).

Another approach to direct mechanical testing at the tissue level is to use finite element models built from micro-computed tomography (μ CT) scans, together with specimen-specific experimental data at the apparent level to calibrate “effective” elastic properties of the tissue (Van Rietbergen et al., 1995). Table 2.2 shows some reported values of the tissue properties obtained utilising different investigation methods.

Table 2. 2: Values of elastic modulus of cancellous bone tissues from selected sources.

Reference	Anatomic site	Method	Elastic modulus (GPa) (\pm SD)
Ashman & Rho (1988)	Bovine distal femora	Ultrasonic technique	10.9 ± 1.5
Rho et al. (1997)	Human Vertebra	Nanoindentation	13.4 ± 2.0
Ulrich et al. (1997)	Human Femoral head	Experiment-FEA	3.5-8.6
Hou et al. (1998)	Human Vertebra	Experiment-FEA	5.7 ± 1.6
Ladd et al. (1998)	Human Vertebra	Experiment-FEA	6.6 ± 1.0
Turner et al. (1999)	Human distal femur	Nanoindentation	18.1 ± 1.7
		Acoustic microscopy	17.5 ± 1.1
Zysset et al. (1999)	Human femoral neck	Nanoindentation	11.4 ± 5.6
Niebur et al. (2000)	Bovine tibia	Experiment-FEA	18.7 ± 3.4
Bayraktar et al. (2004)	Human femoral neck	Experiment-FEA	18.0 ± 2.8
Verhulp et al. (2008)	Bovine tibia	Experiment-FEA	6.5 ± 1.1

Due to the degree of variation in the mechanical properties of cancellous bone, many attempts have been made to describe the strength and modulus of bone as a function of apparent density (or bone volume fraction). Cancellous bone exhibits an open cellular structure. For cellular materials, mechanical properties are strongly related to apparent density. For compressive behaviour, a number of equations have been proposed in literature to describe the correlation between the mechanical behaviour of bone and bone density (Carter & Hayes, 1977; Anderson et al., 1992; Hodgkinson & Currey, 1992; Linde et al., 1992; Li & Aspden, 1997; Kopperdahl & Keaveny, 1998; Ciarelli et al., 2000; Keaveny et al., 2001; Morgan, et al., 2003; Kaneko et al., 2004). Carter & Hayes (1977) tested human and bovine specimens of cortical and cancellous bone, and showed a quadratic relationship between the compressive strength and the apparent density; and a cubic relationship between the apparent modulus and the density. Hodgkinson & Currey (1992) tested bovine, horse and human cancellous bone cubes so that a range of bone densities was considered. They reported a relationship that was closer to quadratic, with an exponent of density ranging from 1.57 to 2.47. According to Cowin (2001), bone volume fraction accounted for 92% of the variation in measured Young's modulus, 88% of the variation of shear modulus and 76% variation in Poisson's ratio. Keaveny et al. (2001) reported power law regressions between the ultimate compressive stress and the apparent density of human trabecular bone specimens, from a range of anatomic sites, with values of r^2 up to 0.93. However a recent literature review (Helgason et al., 2008) found that, although normalised according to density measurement unit and loading rate, the equations reported in literature still showed considerable scatters and it was difficult to draw any definitive conclusions.

An alternative approach, for generalisation of the apparent level mechanical properties of bone, based on cellular theory, is to create a model of an idealised structure of cancellous bone. The repetitive structure, based on unit cells, is used to represent the structure of bone. Structural analysis (which can be in 2D or 3D) is then performed on this idealised structure to derive theoretical relationships between bone volume fraction and mechanical properties (Kowalczyk, 2003; Gibson & Ashby, 1997; Zhu et al., 1997). These idealised versions of bone structure can be combined with finite element analysis (FEA) to isolate and examine the effect of microstructure on the continuum properties. The structure of the unit cell is of vital importance in the determination of the mechanical properties. Kim & Al-Hassani (2002) examined the effect on the mechanical properties of strut taper in an analytical cellular model of vertebral bone, showing how the strut taper model exhibited mechanical properties in the order of 1.8-2.2 times higher than those with uniform struts. This study highlights one of the disadvantages of using only idealised models to simulate the complex architecture of cancellous bone.

Due to the irregularity in native trabecular bones, synthetic cellular materials are often used as cancellous bone substitutes for in vitro biomechanical testing due to their uniformity, consistency in properties and availability. A set of materials with known consistent and controllable mechanical properties, within the range of properties of cancellous bone, can provide valuable alternatives to cadaver bones as test substrates (Szivek et al., 1995). Three dimensional cellular solids are often known as foams. The foam properties generally depend on the bulk material. Rigid polyurethane and aluminum foams have been reported to exhibit similar mechanical (Szivek et al., 1995; Nazarian & Muller, 2004; Thompson et al., 2003) and morphological (Johnson & Keller, 2007) properties to those of cancellous bones.

Metallic foams have been favoured for orthopaedic research and testing purposes for their open-celled, interconnected structure with essentially homogeneous but adjustable densities. The properties of the metallic foams depend on the bulk metal properties and the relative density of the foam, the distribution of the material within the foam and the architecture of the foam (Gibson & Ashby, 1997; Gibson, 2005). Open-cell metal foams exhibit a combination of high interconnecting porosity, high deformability and high specific stiffness (Krupp et al., 2006). These characteristics make them excellent candidates for light-weight designs that combine functional and structural properties (Antsiferov et al., 2005). Mechanical behaviour of metallic foams have been investigated in numerous studies (Andrews et al., 1999; Evans et al., 1998; Prakash et al., 1995; Weber et al., 1994), although a testing protocol had not been available until recently (DIN 50134, 2008). This protocol allows a more consistent approach in the measurement of the mechanical properties, and particularly the elastic modulus. Open-cell aluminum foams have been used to determine the accuracy of a time-lapsed μ CT imaging technique for bone and cellular solids (Nazarian & Muller, 2004; Nazarian et al., 2005). However, the main goals of these studies were other than exploring these materials as cancellous bone substitutes. This will be further discussed in Chapter 5.

2.1.5. Limitations in determining the mechanical properties of bone

The structure and properties of bones vary with many factors such as age, sex, disease and species. This, together with experimental errors, accounts for the large variation in the measured values of strength and modulus of bones (Keaveny et al., 2001). Experimentally, the preparation of samples can affect the measured properties of bone also. Preparation of specimens can damage individual trabeculae leading to an underestimation of modulus;

similarly chemicals and storage can also have a deleterious effect on the properties of bone (Cowin, 2001). A common testing procedure to determine the bone compressive properties involves the use of platens, where the bone is sandwiched between two parallel plates and compressed at a set strain rate until failure occurs. End artefacts such as non-uniform deformation of the bone surface or friction between the sample and the platens are primary sources of errors in the measured properties. Keaveny et al. (1994, 1997) showed that the end effects can result in 20-40% errors in the determined values of modulus, and suggested that the ends of the specimens should be embedded into a sample holder to reduce the artefacts. The measured bone properties are also dependant on the specimen geometry (Linde et al., 1992; Keaveny et al., 1993), strain rate used for testing (Linde et al., 1991) and the test environment (Keaveny et al., 1997). Keaveny et al. (1997) reported a lower band of experimental errors in the modulus measured using an extensometer attached to the surface of the sample. However, the extensometer blades can damage the bone surface, inducing errors also. An alternative way is to attach the extensometer to the holders or use non-contact extensometers.

2.1.6. Post-yield and fatigue behaviour of cancellous bone

Post yield behaviour of bones has been determined at both apparent and tissue levels. Hayes & Carter (1976) tested specimens of bovine cancellous bone under uniaxial compression and examined the post yield behaviour relative to energy absorption characteristics. These tests showed an initial linear elastic response until the yield stress is reached, followed by a region of “strain softening” due to pore collapsing, where the stress-strain curve appears almost horizontal. In this region, fracture and buckling of the trabeculae are both responsible for the energy absorption and load dissipation in bone. Gibson & Ashby (1997) reported that at low

relative densities ($\rho^*/\rho_s < 0.3$, where ρ^* is the density of the cellular material and ρ_s the density of the solid sample from the same material) bending and buckling of trabeculae tend to dominate the failure of trabeculae, whereas for higher relative densities ($\rho^*/\rho_s > 0.3$) trabeculae tend to fail by the formation of plastic hinges in bending and subsequent microcracks generation. Experiments on specimens of trabecular bone have shown that when a bone is loaded past its yield point, there is a residual strain when it is unloaded to zero stress. Reloading repeatedly will result in a reduced modulus. Keaveny et al. (1999) found that for strains up to 3%, the modulus reduction of vertebral trabecular bone is in the range of 5.2-91%. These modulus reductions are not immediately related to the change in bone density (Kopperdahl & Keaveny, 1999), but strongly depend on the applied strain. Reductions in strength are also associated with loading beyond the yield stress of bone.

At the tissue level, the reductions in modulus and strength as a result of overloading have been associated with microdamage accumulation in the bone. Microdamage has been shown to occur preferentially in trabecular bone rather than in cortical bone, and more likely to appear in the form of microcracks than complete fracture of the trabeculae (Fyhrie & Shaffler, 1994; Vashishth et al., 2000; Wachtel & Keaveny, 1997). Fyhrie & Shaffler (1994) reported the compressive failure behaviour of human vertebral cancellous bone subjected to 15% strain. Their study showed that trabeculae orientated horizontally with respect to the applied load were the first to fracture, whereas vertically orientated trabeculae appeared to buckle due to the accumulation of microdamage. It was also suggested that, since disconnected trabeculae are rapidly resorbed, complete fracture of trabeculae can lead to the loss of repair potential.

It has been proposed that cellular materials follow characteristic cyclic deformation behaviour (Gibson, 2005; Ohrndorf et al., 2006; Dendorfer et al., 2009; Michel et al., 1993; Moore & Gibson, 2003), however investigations into the fatigue behaviour of bone are relatively limited. Michel et al. (1993) examined the fatigue behaviour of bovine trabecular bone. The stress-strain curves exhibited increasing non-linearity, hysteresis loops and decreasing secant modulus with time. The number of cycles to failure, defined by a 5% reduction in secant modulus, was found to correlate with the initial global maximum strain ($r^2=0.78$). A difference in modulus degradation for low and high cycle fatigue was also reported, suggesting a possibility that both creep and damage accumulation contribute to fatigue failure of trabecular bone. Two types of failure were observed from the microscopic analysis: brittle fracture and buckling. Moore & Gibson (2003) reported the effect of damage accumulation under cyclic compressive loading conditions on the mechanical behaviour of bovine trabecular bone. The specimens were loaded to various combinations of normalised stress ($\Delta\sigma/E_0$) and maximum compressive strain (ϵ_{\max}). The reduction in normalised secant modulus (E_{sec}/E_0) and specimen residual strain (ϵ_{res}) increased with the increasing of maximum compressive strain (with $r^2=0.66$ and $r^2=0.76$ respectively), and a weak correlation was found between the change in the elastic modulus and the normalised stress used to load the specimens (i.e. $r^2=0.34$ from E_{sec}/E_0 vs $\Delta\sigma/E_0$). It was concluded that the effects of normalised stress and number of cycles on deformation were relatively small, suggesting that trabecular bone failure is largely strain-based. A comparison between the cyclic compressive behaviour of bovine and human cancellous bone was reported by Haddock et al. (2004). Human vertebral samples were tested under cyclic compression and compared against the literature data for bovine cancellous bone. It was found that, despite the two different

origins, bovine and human trabecular bones exhibit consistent qualitative and quantitative aspects of cyclic behaviour. The results showed, for both types, a progressive loss of secant modulus and strain accumulation with the number of cycles. These caused failure at load levels much lower than those required for monotonic failure. More recently Dendorfer et al. (2009) used optical deformation analysis to provide the deformation and damage mechanisms of human and bovine cancellous bones under cyclic loading conditions. The results showed highly non-uniform deformations where damage appeared in form of increasing residual strains, hysteresis area and a decrease in secant modulus. Damage of single trabeculae appeared in form of highly localised strains, which were linked to microcrack initiation and propagation. These findings suggested that cyclic creep plays a vital role in the deformation of cancellous bones under fatigue loading conditions consistently with other authors (Michel et al., 1993; Bowman et al., 1998). However, Moore et al. (2004) seems to believe otherwise, i.e. creep does not contribute to fatigue in trabecular bones.

Due to the complex behaviour of cancellous bone under fatigue loading, analogue bone materials are often used. Open-cell foams in particular are favoured for their compressive fatigue behaviour and their morphometric similarity to human bones (Johnson & Keller, 2007; Harte et al., 1999).

2.2. The human hip joint

2.2.1. Anatomy and kinematics

The hip joint is a ball and socket joint, with the femoral head articulating within the matching acetabular socket (Fig. 2.6).

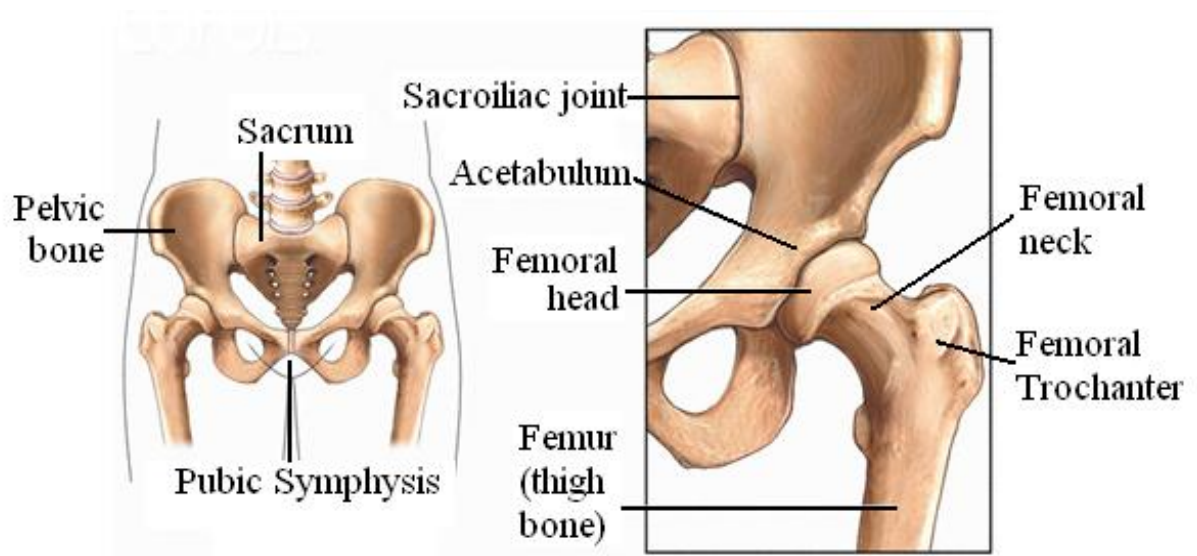


Figure 2. 6: Natural human hip joint (<http://www.corbisimages.com/stock-photo/rights-managed/42-17799773/human-hip-and-pelvic-bones>).

Although is not an exact spherical ball and socket joint in the meridian section, the mean deviation from a conchoids shape is quite small. This special shape makes the joint less likely to sublux compared to a normal ball and socket joint as reported by Menschik (1997). A smooth cushion of articular cartilage covers the femoral head and the acetabulum, and the articular cartilage is kept lubricated by a viscous fluid known as the synovial fluid. Its main function as a fluid medium is to provide nutrition and lubrication within the joint. Since the cartilage is smooth and slippery, the bones move against each other easily and without pain. The hip joint capsule is a dense fibrous structure extending from the base intertrochanteric

region of the femur to the acetabular rim. Large and strong ligaments, tendons, and muscles around the joint capsule hold the bones in place helping the joint motions and supporting stability (Nordin & Frankel, 2001).

The acetabulum is formed at the junction of the ilium, ischium and pubis, as shown in Fig. 2.7.

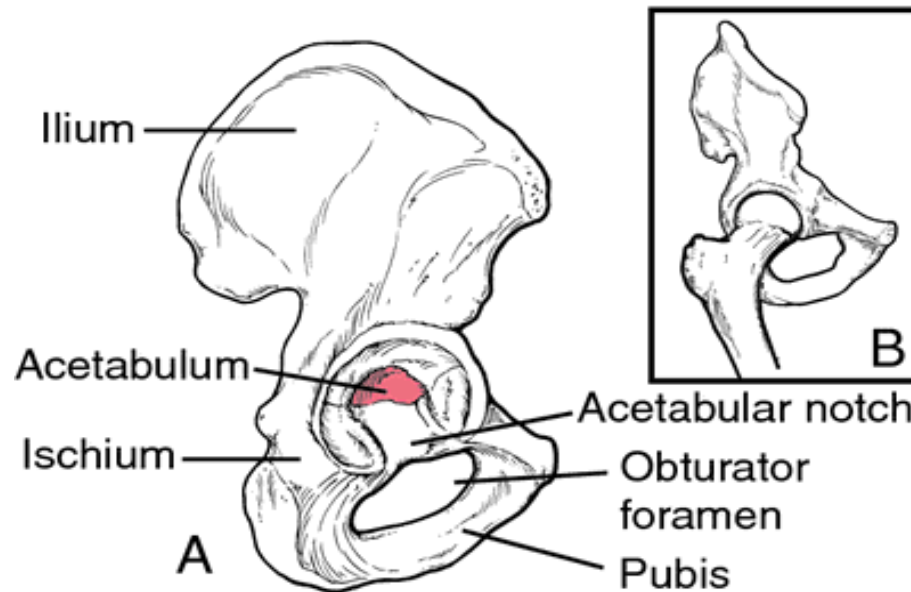


Figure 2. 7: Acetabulum, showing the cup-shaped cavity (A), and its articulation with the femur (B) (<http://medical-dictionary.thefreedictionary.com/acetabulum>).

The ilium is a large flat bone providing broad surfaces for muscular attachment, which extends posteriorly and forms the superior dome of the acetabulum. The ischium joins the ilium superiorly and the pubis inferiorly, forming the posterior acetabulum. The pubis, which serves as insertion site of the musculature of the abdominal wall as well as the site of origin for the adductor muscles of the thigh, forms the anterior acetabulum. The lateral opening of the acetabulum forms a circular horseshoe with the open end directed inferiorly. The acetabulum has a deep notch, the acetabular notch, which is continuous with a circular non-

articular depression at the bottom of the cavity known as acetabular fossa (Wiesel & Delahay, 2010). The acetabular fossa contributes to a more uniform articular contact stress distribution and a consequent decrease in the peak stress, optimising the contact stress distribution in the hip joint (Daniel et al., 2005). The rest of the acetabulum is formed by a curved articular surface, the lunate surface, for articulation with the head of the femur.

The femur is the longest, heaviest and strongest bone in the body. As previously shown in Fig. 2.2 the proximal part of the femur comprises of the femoral head, the neck, the greater and the lesser trochanters. The femoral head is almost spherical in shape and smooth except for a depression, the femoral fovea, which serves as the site of attachment of the ligamentum teres and also houses a blood vessel that supplies blood to the head. The femoral neck connects the head to the main body of the femur (Wiesel & Delahay, 2001). Both the trochanters act as muscle attachment sites. The greater trochanter is located laterally at the junction of the femoral head and neck with the lesser trochanter. The shaft of the femur is bowed anteriorly with the distal end of the femur forming the medial and lateral condyles, which articulates with the tibia. Two superior aspects, known as epicondyles for both the medial and lateral condyles are sites for muscle attachment (Puthumanapully, 2010).

The motion of the hip takes place in all three planes (sagittal, coronal and transverse), as shown in Fig. 2.8. Flexion-extension occurs through the transverse plane with a range of movements normally of 0-120° in flexion and 0-20° in extension. Assisted flexion involves overcoming a significant proportion of the protective tension in the hamstring muscles and joint capsule, resulting in the generation of an extra 25-30° of flexion. Assisted extension of the thigh only produces a further 10° of movement. Medial and lateral rotations occur in the

sagittal plane about the mechanical axis of the femur through a maximum of 90°. Abduction-adduction occurs in an anterior-posterior plane, each within a 45° range (Heaton-Adegbile, 2005).

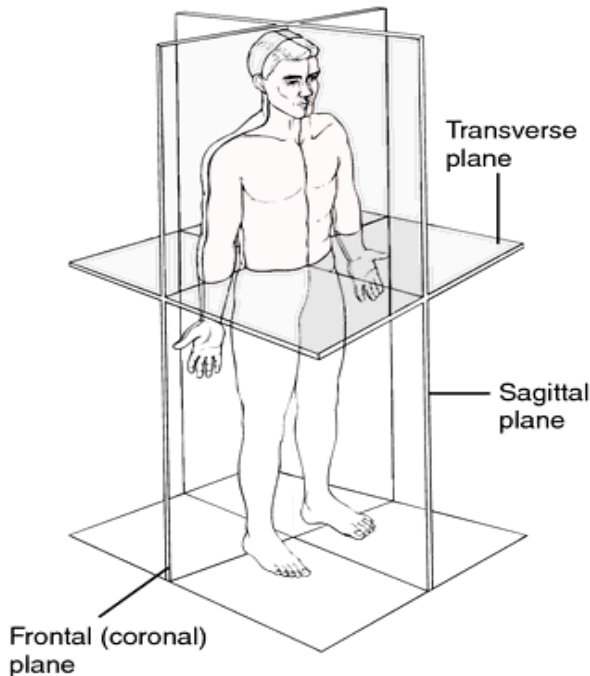


Figure 2. 8: Planes of section. Sagittal, coronal and transverse planes (<http://medical-dictionary.thefreedictionary.com/plane>).

2.2.2. Joint loading and gait analysis

For a complete understanding of the mechanics of total hip replacements, it is necessary to understand first of all the mechanical behaviour of the hip bone in static and dynamic conditions and the stresses generated in the normal acetabulum.

Within the limits of the anatomical boundary conditions, the hip bone has evolved into a very efficient structure, which is well able to carry large forces. Consisting mainly of low-density trabecular bone, which by itself is not strong enough to withstand alone such high loads, it is totally covered by a thin layer of cortical bone. In this way, it resembles a so-called ‘sandwich

construction', used in engineering to combine high strength and low weight (Dalstra & Huiskes, 1995). The space between the femoral head and the acetabulum, while the joint is articulated and loaded during walking, is known as hip incongruity (Afoke et al., 1984). The presence of incongruity, in the anatomical configuration of a normal hip joint, governs both the extent and the position of the weight-bearing areas in the joint and is partially responsible for the pressure level and stress distribution across the articular surfaces (Greenwald & O'Connor, 1971). The bone system consisted of cancellous bone, subchondrial bone and articular cartilage may allow, in normal conditions, the transition from incongruence to congruency resulting in a uniform and efficient distribution of the loads in the joint. Excessive deformation, especially when repetitive, might result in microfractures in the trabecular bone (Heaton-Adegbile, 2005).

The human gait can be defined as cyclic, bipedal locomotion that requires interaction and coordination among most of the major joints in the body, with the hip and the knee playing major roles. The contact forces on the hip vary during the gait cycle (Fig. 2.9).

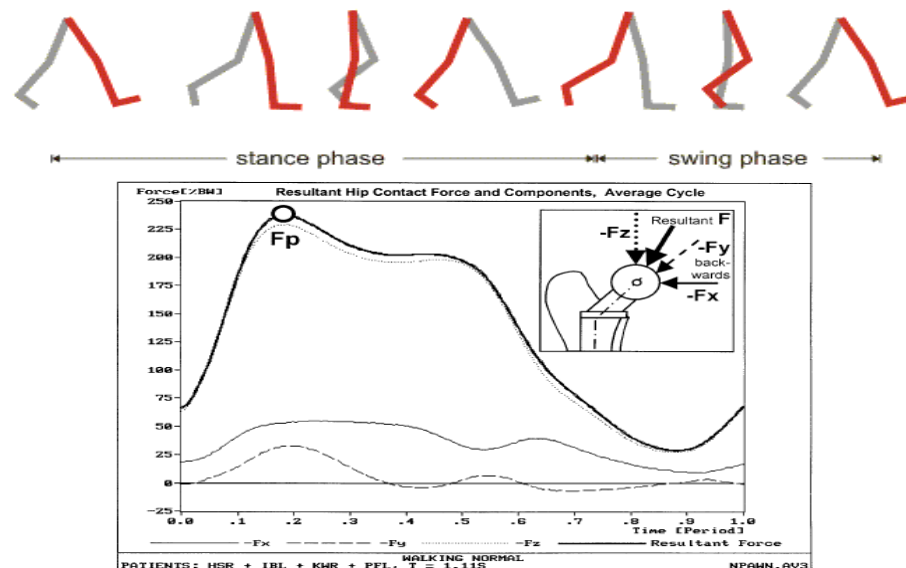


Figure 2. 9: The gait cycle and contact force in normal walking (Bergmann et al., 2001).

Initial contact occurs as the heel makes contact with the ground. The foot is then in flat contact with the ground supporting almost all the body weight before the heel rises. The toe then leaves the ground marking the end of the stance phase and entering the swing phase. During the swing phase the other foot wholly supports the body. The swing phase ends with heel contact, and the cycle then repeats (Vaughan et al., 1992).

Analysis of gait data relies on the accurate determination of events that occur during walking such as 'heel contact' (the time at which the heel first contacts the walking surface) and 'toe-off' (the time at which the foot is removed from the walking surface). Knowing such events allows for normalization of gait kinematics and/or kinetics over various periods, i.e. stance phase, swing phase, or the entire gait cycle (Hansen et al., 2002).

A number of studies have been conducted to examine the human gait under normal and pathological conditions (Miller & Verstraete, 1996; Kleissen et al., 1998; Hansen et al., 2002; Capozzo et al., 2005; Takeda et al., 2009) using different experimental techniques and methodologies. However, the information gathered on the gait is not always easy to translate in terms of hip joint reaction forces. The most comprehensive studies on the matter have been carried out by Bergmann and his associates with instrumented hip prostheses implanted in patients who underwent THR surgery (Bergmann et al., 2001; Heller et al., 2001; Morlock et al., 2001). Gait analysis was performed on each patient where contact pressure data and equivalent joint reaction force were evaluated in parallel with joint kinematics during a variety of daily activities such as normal walking, stair climbing, descending stairs, and rising from a chair. The highest joint reaction forces were found to be 870%BW during stumbling, but were generally in the range of 238-260%BW during walking, stair climbing, and descending stairs. The data also suggested that contact stresses and joint reaction forces

correlated well with foot-floor force measurements, although large inter-subject variation in contact stresses, joint reaction forces and joint kinematics were found. Data from the instrumented prostheses gave so far the most accurate values of the contact pressure of the hip joint. However, the procedure is invasive and the data retrieved from the instrumented prostheses only gave mechanical information relative to the contact site.

2.2.2. Pathological problems of the hip

A hip replacement is usually required when a patient experiences severe pain and disability accompanied by radiological changes of the hip; and where non-operative treatments have failed. The main conditions that necessitate a surgery according to the National Joint Registry (NJR Annual Report 2011, UK) are osteoarthritis, rheumatoid arthritis, avascular necrosis and fracture of the femur.

Osteoarthritis

Osteoarthritis (OA) is a degenerative joint disease that commonly affects middle aged and elderly people and is mainly caused by wear and destruction of the hyaline cartilage, which covers the ends of the bones in proximity of the joints. The incidence of this type of arthritis, uncommon in patients under 40, increases with age and is observed in 80% of men and 89% of women after the age of 75. Obesity and joint damage also increase the likelihood of early development of OA. This pathology can lead to bone on bone contact which may cause intense pain to the patient resulting in the complete destruction of the joint and consequently requiring surgery. Surgical treatment is usually necessary when patients are no longer able to carry out their activities of daily living or employment (Echternach, 1990).

Avascular Necrosis

Avascular necrosis (AVN) of the hip occurs when blood flow to the top portion of the femur is interrupted. The affected portion of the bone consists of head and neck. When it is deprived of blood, this part of the bone begins to 'die', breaking down and causing the cartilage on top of it to collapse. AVN of the femoral head is an increasingly common cause of musculoskeletal disability. Although patients are initially asymptomatic, AVN of the femoral head usually progresses into joint destruction, requiring THR, usually before the fifth decade. In fact, 50% of patients with AVN experience severe joint destruction as a result of deterioration and undergo a major surgical procedure for treatment within 3 years of diagnosis. Femoral head collapse usually occurs within 2 years after the development of hip pain (Aiello, 2011).

Arthritis

Rheumatoid arthritis (RA) is a chronic disease caused by inflammation of the synovium rather than wear of cartilage as with OA. This condition is characterised by exacerbations and remissions that can cause deterioration in the function of the joint, joint pain and joint damage and often affects multiple joints in the body. In women, who are affected two to three times of men, the peak incidence is usually between the fourth and sixth decades. Although medical treatment helps to suppress the associated synovitis, motion once lost from joint destruction is generally not regained. If the patient reaches a point where pain significantly interferes the activities of daily living and occupational requirements, surgery is usually necessary, i.e. THR (Echternach, 1990).

Osteoporosis (OP) is the most common of all metabolic bone disorders. It is characterised by low bone mass and microarchitectural deterioration of bone tissue, with a consequent increase in bone fragility and susceptibility to fractures. Because of the increasingly aging population, the number of persons affected by OP is also increasing. Complications related to osteoporosis can create social and economic burdens (Guglielmi et al., 2011). OP is generally associated with fractures, more commonly observed in women over the age of 65. The common fracture sites occur across the neck of the femur and the femoral trochanter. The blood supply to the bone above the fracture may be interrupted and the femoral head may then die, resulting in secondary deformity and subsequent arthritis (Wang, 2009).

2.3. Total hip replacements

2.3.1. Background

Over the last decades, total hip arthroplasty (THA) has become one of the most successful surgical operations. The main reason for a THA is usually related to hip pathologies that cause pain and loss of functionality in a natural joint. The modern surgical treatment was introduced by the pioneer Sir John Charnley, who first used a self polymerising acrylic bone cement to act as a bonding material between the bone and the implant. Charnley also introduced the concept of 'low friction arthroplasty', referred to the successful pairing of ultra-high molecular weight polyethylene (UHMWP) and metal for the ball-socket replacement system (Charnley, 1979). The evolution of surgical techniques, implant materials and implant designs have led to excellent long-term results in older and more inactive patients, with survival rates reaching 90% or more at 10 years (Varini, 2007). Today, a large

variety of solutions are available: from the use of UHMWP for the hemispherical socket, which is fixed onto the acetabulum, to the metallic or ceramic head mounted using a press-fit principle onto the metallic femoral stem. There are two main different fixation techniques to perform a THR currently: cemented implants and cementless implants. Fig. 2.10 shows a schematic of a cemented THR.

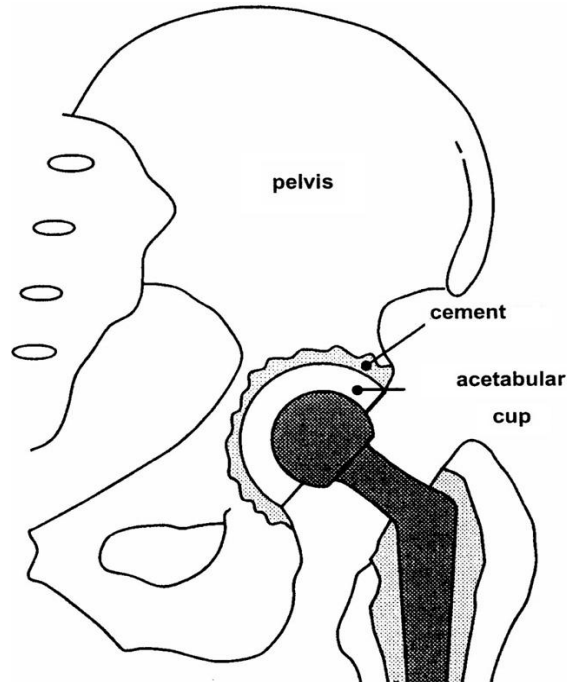


Figure 2. 10: Schematic cemented acetabular replacement in a total hip replacement (Stolk et al., 2004).

The main advantage of cemented hip arthroplasty is the primary stability ensured by the presence of a cement layer at the interface between the implant and the bone. This stability can be achieved soon post-operatively resulting in a short recovery period. The main problem with cemented hip implants is the deterioration of the bone/implant cement interface due to both mechanical and biological factors (Sundfeldt et al., 2006). Furthermore, the microcracks generated by mechanical cyclic loading can aggregate in large cracks

producing cement debris that may eventually lead to inflammatory reactions of living tissues (Schmalzried et al., 1992).

Cementless implants are based on a simple principle. Firstly “press-fit” guarantees that both the stem and the prosthetic cup are firmly anchored to the bone, in the immediate post-operative period (primary stability), in order to provide necessary time for the bone ingrowth into the metallic surfaces. This would allow long-term bonding and the mechanical stability of the prosthesis to the surrounding bony layers (secondary stability). Unfortunately, many factors can jeopardise the outcome of such an implant. The main complications of cementless prostheses are related to the incomplete osteointegration process. This may affect the mechanical stability causing high relative motions between the implant and the bone. The high interfacial motions further discourage bone ingrowth, and can lead to tissue inflammatory processes, which often results in fibrous tissue formation leading to implant gross loosening (Varini, 2007).

Regardless of the fixation technique the main aim of THA is to restore the function of the hip joint by reproducing the same load transfer and the same stress distributions in the healthy joint. The forces normally applied to the femur are transmitted via the cancellous bone of the pelvis to the acetabular subchondral bone (Currey, 1998). The resultant force can be several times in magnitude of the body weight (Bergmann et al., 2001). A strong and reliable bond at the interface between the bone and the implant, after THR, needs to transmit those forces from the prosthesis to the natural bone under in vivo service conditions.

2.3.2. Failure scenarios

It has been a matter of debate whether the failure of a THA is triggered by mechanical or biological events. The roots of a potential failure may have both mechanical and biological origins; and a failure is always a process and hardly ever an event (Huiskes, 1993). The combined effects of mechanical and biological elements in the failures of THAs have been reported by Huiskes (1993; 1993b), who suggested how long-term integrity of THA strongly depends on the lasting mechanical integrity of the bond between the implant and the bone. If that bond, regardless of its nature, is progressively damaged or destroyed, the loading acting on the hip joint would be affected and can produce increased micromotions between the implant and the bone. This results in inflammatory reactions, progressive interface bone resorption, fibrous-tissue interposition and eventually clinical loosening (Huiskes, 1993). Huiskes described the 'failure scenarios' based on failure mechanisms, which have become widely accepted. A short description is given here:

Accumulated damage

The accumulated-damage scenario is based on the gradual accumulation of mechanical damage in materials and interfaces due to repetitive dynamic loading. The local stress at the interface or in the cement mantle may exceed the material strength, which leads to initial cracks that can then gradually grow under the influence of the dynamic loading. This situation may further produce interface micromotion, leading to bone resorption and breakdown of the implanted component due to gross loosening. This scenario was described as the main reason for failures in the cemented THR by Stauffer et al. (1982), and supported by ex vivo investigations of retrieved samples (Jasty et al., 1991). The likelihood of failures

due to accumulated damage may be different for cemented and non cemented THRs, but the underlying principle is the same.

Particle reaction

This scenario describes the debonding of the bone-implant interface as a result of macrophages activation at the interface caused by small particles and wear products produced by the implant during service. These products, such as polyethylene particles from the acetabular implant at the bearing surface, PMMA cement abraded from the bone-cement interface or even metal debris migrated into the interface from the periphery of the acetabulum, can trigger inflammatory responses leading to local bone resorption and interfacial debonding (Schmalzried et al., 1992). Eventually this process produces relative interface motions that may lead to gross loosening of the implant.

Stress shielding

The introduction of artificial implants leads to bone remodelling as a result of the new loading conditions due to the introduction of the implanted components. The hip joint force, which was formerly carried only by the bone, is now shared between the implant and the bone depending on the stiffness of the two materials. The stiffer implant takes a higher share of the total load 'shielding' the less rigid bone. This usually induces bone remodelling in the regions used to experience certain loads and now stress-shielded by the introduction of a new and stiffer material. This scenario suggests that stress shielding due to bone remodelling after replacement can lead to bone loss. The degree of the bone loss/stress shielding in the bone depends on a number of patient-related and implant-related factors, such as age, sex, medication, time of primary surgery, activity level, weight, bone quality, bone properties,

implant stiffness, shape and elastic modulus, interfacial bond characteristics and hip joint force (Mow & Hayes, 1991). During the revision procedure stress shielding regions can be easily recognized as massive bony defects.

Destructive wear

The destructive-wear scenario refers to the destruction of the bearing surface of implanted components due to wear, to a point where mechanical integrity can no longer be maintained (Saikko, 1993). The small micromotions and implant migration produced in service may contribute to create a malfunctioning joint coupling leading to the creation of wear and abrasion.

2.3.3. Aseptic loosening

Aseptic loosening, defined as the late failure of an implant in the absence of infection, is the most common cause of revision of major arthroplasties and is responsible for more than two-thirds of hip revisions (Sundfeldt et al., 2006). The loosening process can be captured on early radiographs as a thin radiolucent line, which later thickens causing consequent bone loss and resulting in the eventual failure of THR.

DeLee & Charnley (1976) examined the frequency of radiological demarcation of the bone-cement junction in the acetabulum after total hip replacement in 141 Charnley low-friction arthroplasty followed for an average of 10.1 years. Standard anterior-posterior radiographs were used to identify the radiolucent traces at the bone-cement interface. For quantification purposes, the total area of interest was divided in three zones (Fig. 2.11) which are currently known as DeLee & Charnley zones. The results showed a demarcation of various degrees in

69% of the cases, whilst only 9.2% of the series showed evidence of progressive migration of the socket.

Aseptic loosening can be the result of inadequate initial fixation, mechanical loss of fixation over time, or biologic loss of fixation caused by particle-induced osteolysis around the implant. The causes of particle accumulation vary from implant interface wear, micromotion occurred in response to corrosion, oxidative reactions, and minor pathogen contaminations (Abu-Amer et al., 2007).

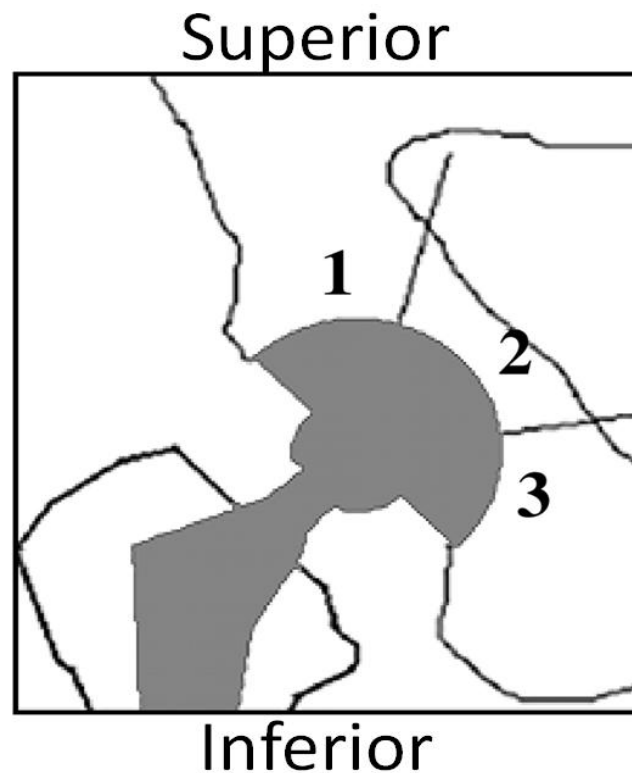


Figure 2. 11: DeLee & Charnley zones identification and distribution relative to the anterior-posterior schematic view of THR (Dumbleton & Manley, 2004).

Jasty et al. (1991) proposed that mechanical factors such as bone cement properties and mantle thickness; implant-cement and bone-cement interface; or the cementing techniques for cemented components are primarily responsible for the loosening process. On the other hand it is believed that biological reactions to the implant materials can lead to fibrous tissue formation between trabecular bone and cement, being mainly responsible for the loss of integrity and functionality of the fixation (Schwarz et al., 2000).

Fornasier & Cameron (1976) observed diffusion of tissue fluid between the prosthesis and the cement in well-fixed THR retrievals. Similarly, Schmalzried et al. (1992) also found that joint fluid penetrated extensively along the bone-implant interface. The effects of moisture and cyclic mechanical stresses can also lead to degradation of cement and failures at the bone-cement and cement-prosthesis interfaces (Fischer et al., 2001; Schmitt et al., 2004). Furthermore, small PMMA particles and other wear products from polyethylene (PE) cup or metal debris may be generated at the interfaces by micromotion or cement fracture, opening a transportation channel and being injected into the hip joint space by the high pressure joint fluid (Aspenberg et al., 1998). Wear debris activate macrophages, which stimulate osteoclasts and initiate the consequent bone resorption. For this reason sealing off the interfaces in cemented or uncemented implants is probably vital for a better outcome of the THA. However, sealing must be established very early, otherwise an increasing amount of debris will be pumped into the interface by the joint fluid drastically reducing the contact between bone and cement, or bone and implant eventually leading to the failure of the implant (Sundfeldt et al., 2006).

Cement porosities generated during the polymerisation process might be a source of fracture with the consequent production of cement particles, thus air bubbles should be reduced as

much as possible (Lewis, 1997). In this case cementing techniques, such as vacuum mixing, cement pressurization and cement pre-cooling, can be most influential in reducing the possibility of cement particles.

More realistically multiple mechanical and biological factors may jointly contribute to the implant loosening (Huiskes, 1993, Sundfeldt et al., 2006). Although the eventual failure of THA may appear to be biological, such as bone resorption and deterioration of the bone-cement interface, the initiation of failure may well be mechanical (Jasty et al., 1991; Huiskes, 1993). For example, stress shielding is a mechanical phenomenon, but its relationship with bone remodelling is a biological one; inflammatory reactions to particles are biological processes, but they are influenced by relative motions and wear processes, both mechanical phenomena.

Although great efforts have been made to understand aseptic loosening, the underlying mechanisms are still not fully understood. Fundamental studies are required in the design and analysis of THR, fixation methods, surgical techniques and in vitro testing of implanted replacements.

2.4. Bone-cement fixation

2.4.1. Bone cement properties

Bone cement is a two-component material. The basic constituents are the powder (initiator and radio pacifier) and the liquid monomer (polymer and accelerator). The compositions of various commercial brands are variants within a certain range. An inhibitor or stabilizer (hydroquinone) is added to the liquid component to ensure that the monomer does not polymerise spontaneously. BaSO_4 or ZrO_2 are commonly used as a radio pacifying agent in

the powder, but an experimental composition with a methacrylate-type iodide containing copolymer (ICCP) showed much better integration into the cement matrix (Lewis, 2003). For operations where an infection is suspected, an antibiotic is added to the powder. This wide-spectrum antibiotic is either added by the manufacturer or by the surgeon.

After mixture of the two components, the initiator is decomposed by the accelerator in molecules, so-called radicals that trigger off the polymerisation (Huiskes, 1979). Fig. 2.12 shows the equation of the polymerisation reaction. The characteristics of the polymerisation process can be defined with reference to a temperature-time curve, as shown in Fig. 2.13 (Huiskes, 1979).

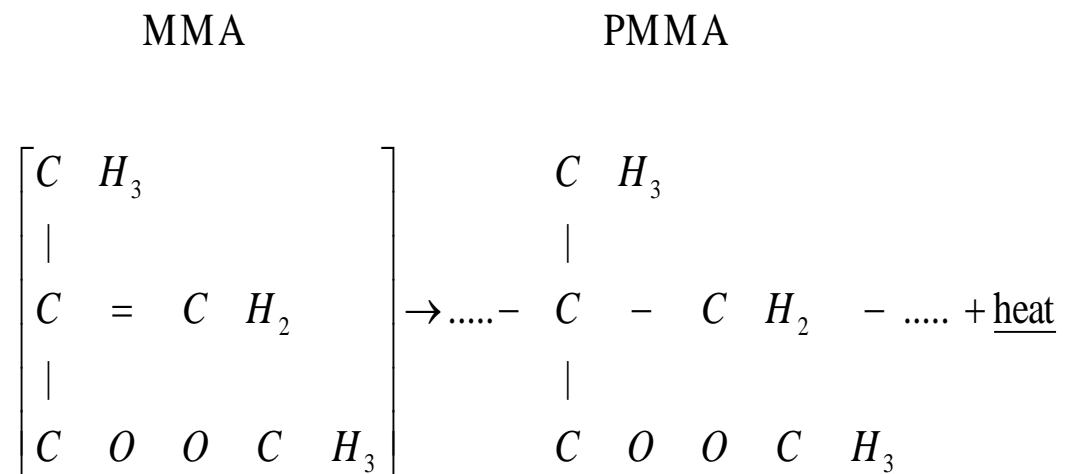


Figure 2. 12: Equation describing the polymerisation reaction. Mono-methyl-methacrylate molecules are forming into polymer chains (Huiskes, 1979).

The bone cement mechanical properties can vary with a number of factors such as basic composition, antibiotic presence, storage temperature and soaking conditions (Lewis, 1997; 2003; Koole et al., 1999; Nottrott et al., 2007).

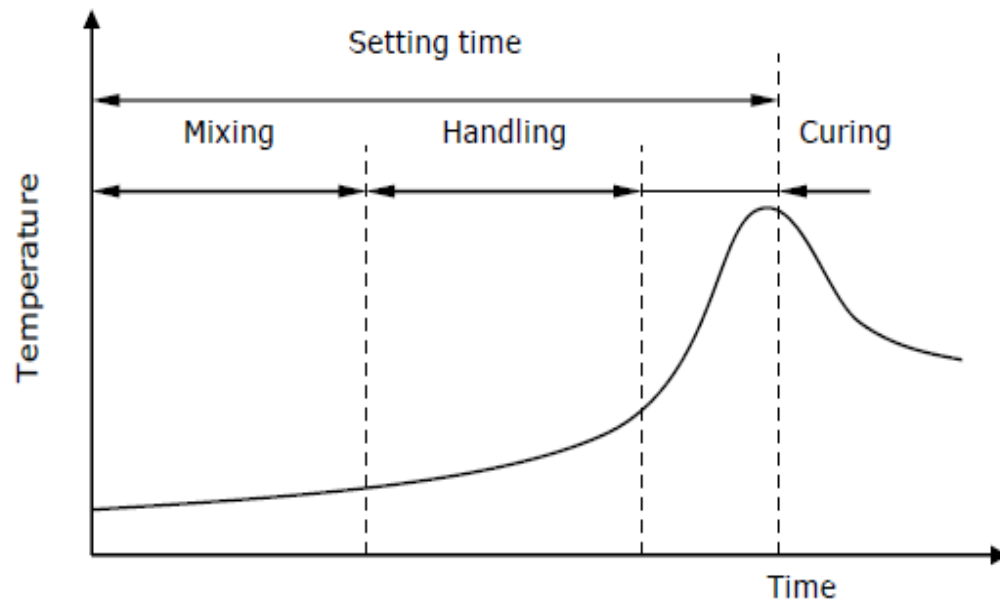


Figure 2. 13: Temperature-time curve of the polymerisation process (Huiskes, 1979).

Table 2.3 shows the range of properties for six commercial formulations of bone cement for orthopaedic applications (Lewis, 1997).

Table 2. 3: Mechanical properties of selected cement brands (Lewis, 1997).

Properties	Range of values
Ultimate Tensile Strength	23.6-47 MPa
Tensile Modulus	1583-4120 MPa
Ultimate Compressive Strength	72.6-117 MPa
Compressive Modulus	1940-3180 MPa
Shear Strength	32-69 MPa

2.4.2. Histology of the bone-cement interface

Charnley (1960, 1964) examined the histology of the bone-cement interface by sectioning of post-mortem femurs from patients with clinically successful implants. In load bearing regions new bone tissues, evidence of bone remodelling, formed on the ends of the trabeculae. In non-load bearing regions, a fibrous tissue layer may develop between the bone and the cement, as shown in Figure 2.14. In the case of successful replacements, this fibrous layer was thin and non-continuous. When a thick fibrous tissue layer is visible as lucency on a radiograph, this represents a main indication of loosening.

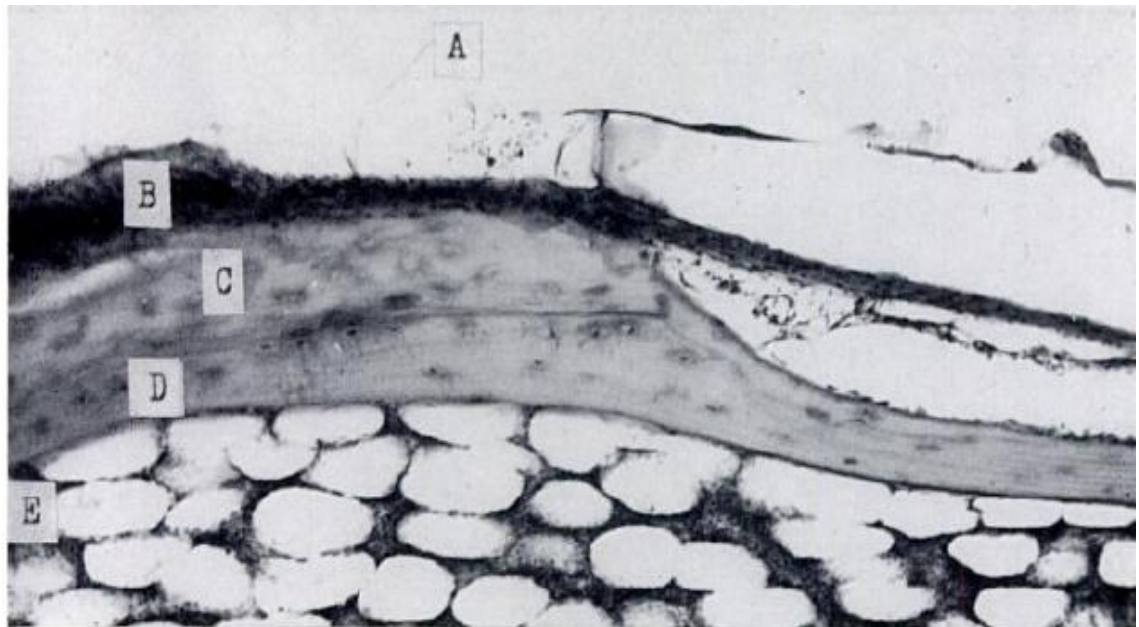


Figure 2. 14: Histology of the bone-cement bond. (A) - cement, (B) - fibrous tissue, (C) - dead cancellous bone and (D) - new lamellar bone deposited on dead bone and closing the trabecular space, (E) - fatty marrow with normal appearance (Charnley, 1964).

The histology of the interface changes with time postoperatively. Vernon-Roberts (1984) characterised the histology of the bone-cement interface in stable hip prosthesis up to 20 years after implantation from a number of macroscopic and microscopic studies. The stability was defined by four phases of the tissue reaction:

- Up to three weeks after implantation a layer of dead bone up to 5mm in thickness was found in apposition to the cement surface, with the presence of osteoclasts laying down new bone.
- The reparatory phase began at week four, lasting for a minimum of six months up to two years. Microscopic analysis revealed a thin layer of connective tissue approximately 1.5mm thick between bone and cement, where active remodelling of tissue occurred.
- Stabilisation occurred between 6 months to 2 years postoperatively, with all necrotic and repaired bone being replaced by living lamellar bone. The architectural modelling was by then completed and all the remaining fibrous tissue disappeared.
- At a minimum of two years, the prosthesis may be considered stable. Lamellar trabecular bone appeared in close contact with cement surrounding the prosthesis, although a thin layer of fibrous tissue was normally present between the bone and cement surface.

Jasty et al. (1990) examined thirteen post mortem well functioning femoral prostheses that were obtained at autopsy from patients in whom a cemented total hip replacement had been implanted from 40 months to 17.5 years. All specimens exhibited an intact bone-cement interface with interdigitation of the cement into the trabecular bone. Evidence of extensive intramedullary trabecular bone remodeling was observed in all of the femora that were retrieved in the time frame of 5 to 17.5 years after operation. In the samples retrieved at less than 5 years, cement had intruded into trabecular bone. The trabecular pattern in the proximal metaphysis and in the proximal part of the diaphysis of the specimens, retrieved at about five to 17.5 years, had been extensively altered and was characterised by thick

trabeculae running circumferentially around the cement-mantle and thinner trabeculae surrounding these thick trabeculae, which gradually merged with the original outer cortex.

The above histological studies indicated that a well functioning prosthesis exhibits an intact bone-cement interface with direct contact between cement and bone and bone apposition to the cement surface. However, the histology of the bone-cement interface in a loosened implant is different. Schmalzried et al. (1992) examined fourteen hemi-pelvic bones with cemented polyethylene acetabular components obtained at autopsy, with an implantation time varied from 58 to 210 months. Evidence of bone resorption and membrane formation at the bone-cement interface was clearly visible, even in the most well-fixed implants, resulting in the loss of mechanical stability. Histological analysis revealed that progressive bone resorption, and consequent membrane interposition between bone and cement, is fuelled by small particles of polyethylene migrating along the bone-cement interface as a result of macrophage inflammatory response of the tissue to the particulate. It was concluded that the mechanism of aseptic loosening for cemented acetabular components is biological in nature and not mechanical. However, the mechanical contribution to the failure of cemented acetabular replacement has never been reported and this will be addressed in Chapter 3.

2.4.3. Properties of the bone-cement interface

Since bone cement does not have any adhesive qualities, the strength of the bone-cement interface strongly depends on the mechanical interlock between cement and cancellous bone. An improved cement interdigitation into the cancellous bone is considered beneficial to increase the strength of the bone-cement interface fixation, as the greater the penetration into the trabecular structure (up to 3-5mm), the stronger the mechanical interlock (Krause et al., 1982; Bugbee et al., 1992; Majkowski et al., 1993; Mann et al., 1997).

To achieve a good degree of cement penetration, high cementation pressures are needed (Juliussen et al., 1995) and adequate preparation of the cancellous bone bed is essential (Krause et al., 1982). Oates et al. (1995) reported the in vivo effect of cement pressurisation on the interface strength of sixteen goats that underwent total hip arthroplasty. Pressure was maintained using a distal plug and a pneumatic gun apparatus in half the animals and finger packing in the other half. The biomechanical results demonstrated that pressurised interfaces were stronger than those produced by digital packing; although there was a limit beyond which increasing pressure had no effect. In addition to pressurisation, cleaning the surface of bone also aids cement penetration. An accurate lavage before the cement application removes loose cancellous bone, blood and fat from the surface of the bone, enhancing the penetration of the cement into the trabecular spaces and improving the contact between bone and cement (Krause et al., 1982). Krause et al. (1982) reported how pressurised interfaces with lavage exhibited greater strength than as cut, unpressurised interfaces. In vitro tensile and shear tests of bone-cement interfaces were carried out to determine the mechanical strength of the interface with respect to preparation of bone surface and cementing technique. Tensile strength varied from $2.72(\pm 1.42)$ MPa for as cut specimens to $5.30(\pm 2.13)$ MPa for specimens subjected to high intensity lavage. Same findings, with lower improvement for high intensity lavage, were obtained for shear strength ranged from $7.92(\pm 2.54)$ MPa to $8.50(\pm 4.28)$ MPa. It was concluded that, although cement penetration depths above 1-2mm have been shown to increase the cement-bone interface strength, it seems that an upper penetration limit exists and for penetration depths above 5mm the strength of the interface does not increase significantly.

The viscosity of cement changes with time after mixing until the cement cures, in different modalities depending on the type of cement. The viscosity at time of insertion can influence the degree of cement penetration. Stone et al. (1996) examined the effect of viscosity on the push out strength of bone-cement samples. Femora were injected with cement at intervals of 4.5 minutes and 6.5 minutes (after mixing) respectively to simulate insertion at low and high viscosity. The results showed that cement injected with lower viscosity exhibited greater failure strength. However, these tests did not simulate intramedullary bleeding which can also influence the amount of cement interdigitation (Majkowski et al., 1994). Miller et al. (2007) included bleeding in their investigation and examined the difference in tensile strength between low viscosity and standard viscosity cements. The cements were injected at 3.7 minutes and 4 minutes (after mixing) respectively into human femurs, and in this case the interface made of standard viscosity cement was 21% stronger than that using the low viscosity cement. They concluded that low viscosity cement was less effective at displacing bone marrow resulting in less bone apposition and therefore lower interface strength. Similar findings were also reported by Race et al. (2006).

Although numerous studies have been carried out on bone-cement interfacial strength, using experimental (Krause et al., 1982), analytical (Clech et al., 1985) or numerical (Tong et al., 2007) methods, the most complete studies have been carried out by Mann and his associates (Janssen et al., 2008; Mann et al., 1997, 1999, 2001; Kim et al., 2004; Mann et al., 2009). Their work includes mechanical testing of cadaveric bone-cement interface samples under tensile (Mann et al., 1997), shear (Mann et al., 1999), mixed mode (Mann et al., 2001) and creep-fatigue (Kim et al., 2004). Much valuable information has been generated from these well-designed studies, although some of the fundamental aspects of the topic remain to be

further explored. For example, what is the role of cement interdigitation in defining the bone-cement interfacial strength? Several methods have been used to evaluate interdigitation. The depth of the cement pedicles (Krause et al., 1982) or the quality of cancellous bone (Askew et al, 1984) was used as a measure of cement penetration. Mann et al. (1997) used quantitative computed tomography (QCT) to measure interdigitation by a product of the amount of interdigitation and equivalent mineral density of the interdigitated bone; whilst Maher & McCormack (1999) used a random undulating parameter to measure the depths of the cement penetration at a regular interval.

Published works do not appear to suggest a significant correlation between the bone-cement interfacial strength and the depth of cement penetration ($r^2=0.22$ (Krause et al., 1982)), although the methods of quantification of cement penetration tend to vary. The quantity of bone interdigitated with the cement has been used to improve the correlation (Mann et al., 1997, 1999, 2001; Kim et al., 2004), albeit with only moderate success ($r^2 \approx 0.5$ for tensile; $r^2 \approx 0.5-0.7$ for mixed mode cases). The main failure mechanisms were identified (Mann et al., 1997, 1999), although not correlated with the loading angle. In general, there is a large variation in the results between specimens tested under similar loading conditions, possibly due to the variation in cadaver bone tissues, and the difference in the amount of cement infiltration. Due to the substantial scatter, the penetration depth alone does not appear to fully describe the behaviour of the interface. Important indications in this sense are provided by the type of loading condition to which the interface is subjected. In this sense the results reported by Mann and associates (1997, 1999, 2001) showed a general increase in apparent strength with the loading angle from $1.35(\pm 0.65)$ MPa in tension (0°) to $2.25(\pm 1.49)$ MPa in shear (90°), with the only exception for the value of $3.00(\pm 1.70)$ MPa obtained for samples

tested under mixed-mode at 67.5°. It was concluded that the strength of the bone-cement interface was moderately correlated with the loading angle ($r^2 \approx 0.6$) being greater in shear than in tension. Wang et al. (2010) obtained similar results that will be discussed in Chapter 4.

At a microstructural level, failure of the bone-cement interface depends by both the type of interlock and the degree of cement penetration. The penetration of cement into the trabecular bone forms a composite structure. As such, failure can occur through a number of mechanisms. Miller et al. (2007) classified the failure of bone-cement interface into six categories, as shown in Figure 2.15.

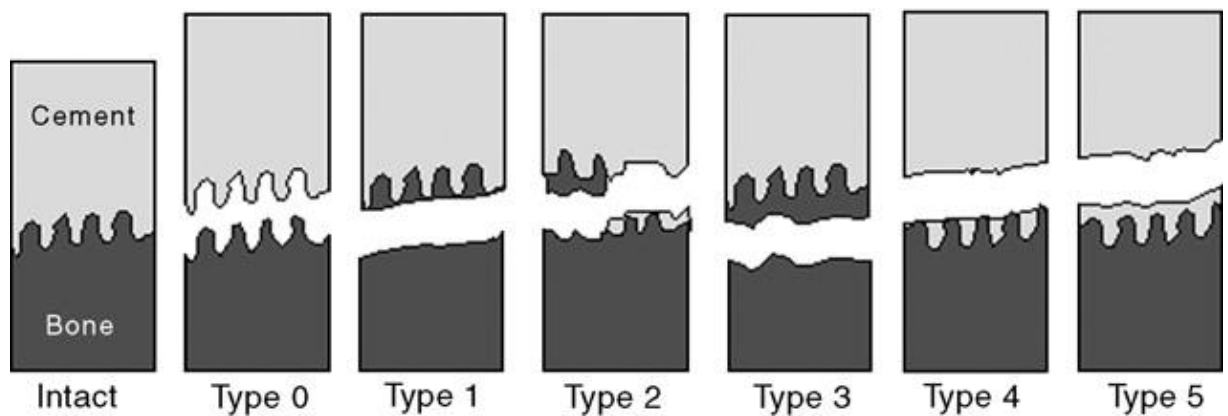


Figure 2. 15: Failure type classification of bone-cement interface specimens (Miller et al., 2007).

With the recent development of μ CT imaging techniques and the advent of digital correlation analysis a better understanding of the micromechanical behaviour of the bone-cement interface has been achieved. Mann and co-workers reported the micromechanical behaviour of the bone-cement interface under tension-compression (Mann et al., 2008; Miller et al., 2009), shear fatigue (Mann et al., 2009) and multi-axial (Miller et al., 2011) loading conditions. The main aim of these studies was to quantify the micromechanics of the bone-

cement interface of laboratory-prepared and post-mortem retrieved cemented total hip replacements when subjected to the selected loading conditions. μ CT together with digital image correlation techniques (DIC) were utilised to determine the relative micromotions and local deformation at the interface. The results showed that the actual contact interface was responsible for the majority of motion of bone-cement structures when subjected to tensile or compressive loading. This interface was more compliant in tension than in compression, with a fraction of total displacement that was 0.83 (0.19) for tension and 0.64 (0.28) for compression respectively (Mann et al., 2008). The mechanical response of bone-cement post-mortem retrieved specimens was characterized by a highly compliant interface in both tensile and compressive loading directions. The retrievals exhibited a very low tensile strength ($0.21(\pm 0.32)$ MPa) when compared to laboratory-prepared specimens ($3.00(\pm 1.4)$ MPa). The relatively weak and compliant interface from the post-mortem retrievals was explained by the paucity of direct apposition between the cement and bone, and the lack of bone-cement interlock (Miller et al., 2010). Under shear fatigue the damage response of the interface was characterized by creep damage that localized to the contact interface between bone and cement. The overall creep damage was associated with stiffening of the structure ($r^2=0.68$), and more microcrack damage occurred to the cement than to the bone with failure that was confined at the contact interface (Mann et al., 2009). Under non destructive multi-axial loading conditions it was found that the magnitude of interface micromotions from an externally applied load did not depend on the angle at which that load is applied (Miller et al., 2011), not entirely consistent with the findings of Mann et al. (2001). All the above reported studies are limited to 2D analysis of the external surfaces of the samples. This will be further discussed in Chapter 4.

2.4.4. The bone-cement fixation in acetabular reconstructs

Although much efforts have been made towards the understanding of aseptic loosening of cemented THRs, most studies are related to the femoral components (Maloney et al., 1989; Topoleski, 1990; Jasty et al., 1991; McCormack & Prendergast, 1996, 1999) and only few works are reported on acetabular components (Heaton-Adegbile et al., 2006; Zant et al., 2007, 2008; Tong et al., 2008; Wang et al., 2009), despite that the long term failure rate is reported to be three times in acetabular components than that in femoral components (Schulte et al., 1993).

The fixation of acetabular replacements has been studied by Heaton-Adegbile et al. (2006), Zant et al. (2008), Tong et al. (2008) and Wang et al. (2009). The findings from the above studies suggested that the interfacial strength of bone-cement interface depends on the bone-cement interdigitation, which is related to a number of factors including cement defects (pores, voids and other impurities), bone quality of the patients and surgical factors (reaming, cleaning, cement viscosity, pressurization).

In vitro fatigue testing of 3D cemented acetabular reconstructs was reported by Heaton-Adegbile et al. (2006). The damage initiation and evolution in the cement fixation of a composite sawbone hemi-pelvic specimen were examined by a combination of conventional fatigue testing and μ CT analysis, the latter used for the first time to monitor the fatigue damage in acetabular reconstructs. Extensive interfacial fracture at the bone-cement interface was observed in proximity of the dome region, as also confirmed by microscopic studies post testing. The sawbone specimens were fully fixed to the experimental setup on the servo-hydraulic testing machine at iliac crest and ischial tuberosity, applying a uniform pressure (Fig. 2.16). The peak hip contact force during normal walking (Bergmann et al.,

2001) was used to load the sample and the body weight set at 70Kg. The load ranged from -0.2 to -1.6kN at frequency of 5Hz in order to reduce the test duration. The specimens were CT scanned at regular intervals. No defect was found at 10 million cycles, and the evidence of initial fractures was detected at 15 million cycles. An extensive demarcation (1-3mm) at the bone-cement interface was identified at 20 million cycles.



Figure 2. 16: The composite sawbone specimen under in vitro fatigue test (Heaton-Adegbile et al., 2006).

Experimental results from a new hip simulator study were reported by Zant et al. (2008). The hip simulator (Fig. 2.17) was able to simulate the complex resultant hip contact force, with varying direction and magnitude at selected load levels ranging from a body weight of 75kg to 125kg, under typical physiological loading conditions, including normal walking and stair climbing data from Bergmann et al. (2001).



Figure 2. 17: The Portsmouth four-station hip simulator with a specimen loaded in one of the test cells (Zant et al., 2008).

The femoral head attached to a swing arm is able to oscillate about the Z-axis, while the rotation of the base plate is about the Y-axis. The force vector was transmitted through the femoral head to the acetabular cup of the reconstruct, as shown in Fig. 2.18.

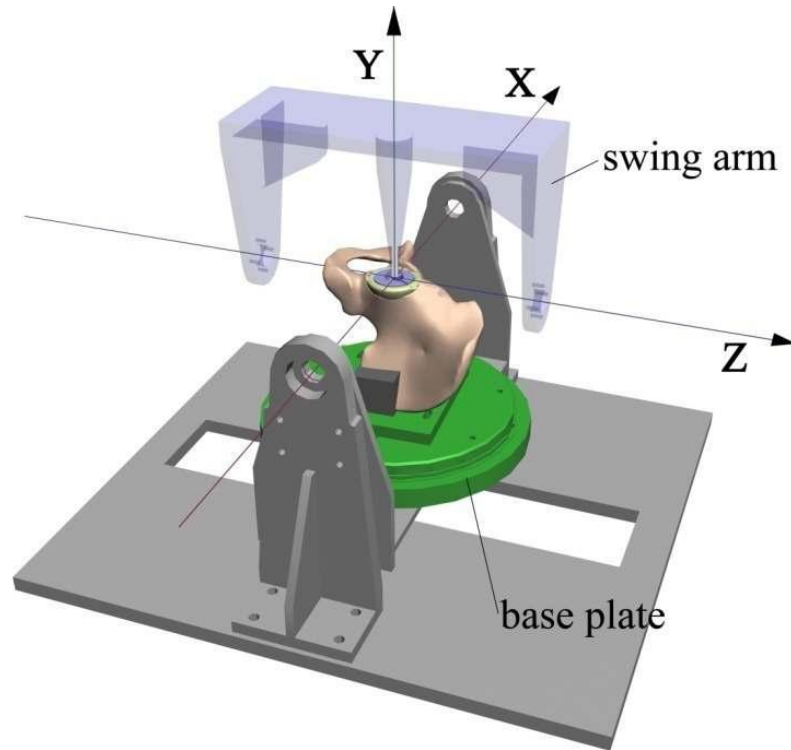


Figure 2. 18: Schematic of the Portsmouth hip simulator with the coordinate system (Zant et al., 2008).

The three-dimensional time-dependent force vector in a cup coordinate system was achieved by synchronising the magnitude of the force with two rotational displacements associated with cup and femoral head. The ranges of the motions allowable were $\pm 35^\circ$, $\pm 360^\circ$ and $\pm 60^\circ$ for rotations on YZ, XZ and XY planes, respectively. The ranges required for normal walking (XZ: 88° ; XY: 35°), ascending stairs (XZ: 141° ; XY: 29°) and descending stairs (XZ: 193° ; XY: 44°) were thus well within the capacity of the apparatus. The machine included four stations which were individually controlled, allowing different selected load profiles to operate independently and simultaneously. The stations were carefully designed to hold hemi-pelvic bones with an implanted acetabular cup. The cup was in articulation with a spherical femoral head through which the load was transmitted. The hemi-pelvic bone specimen was mounted

upside down in order to facilitate lubrication during testing. Fig. 2.19 shows the comparison of the hip contact force as a command signal and the response of the machine for normal walking, ascending and descending stairs. The hip simulator seems to reproduce the physiological loading profiles well, with a relative error of 3.6% for normal walking, 1.5% for ascending and 6.2% for descending stairs, at peak hip contact force.

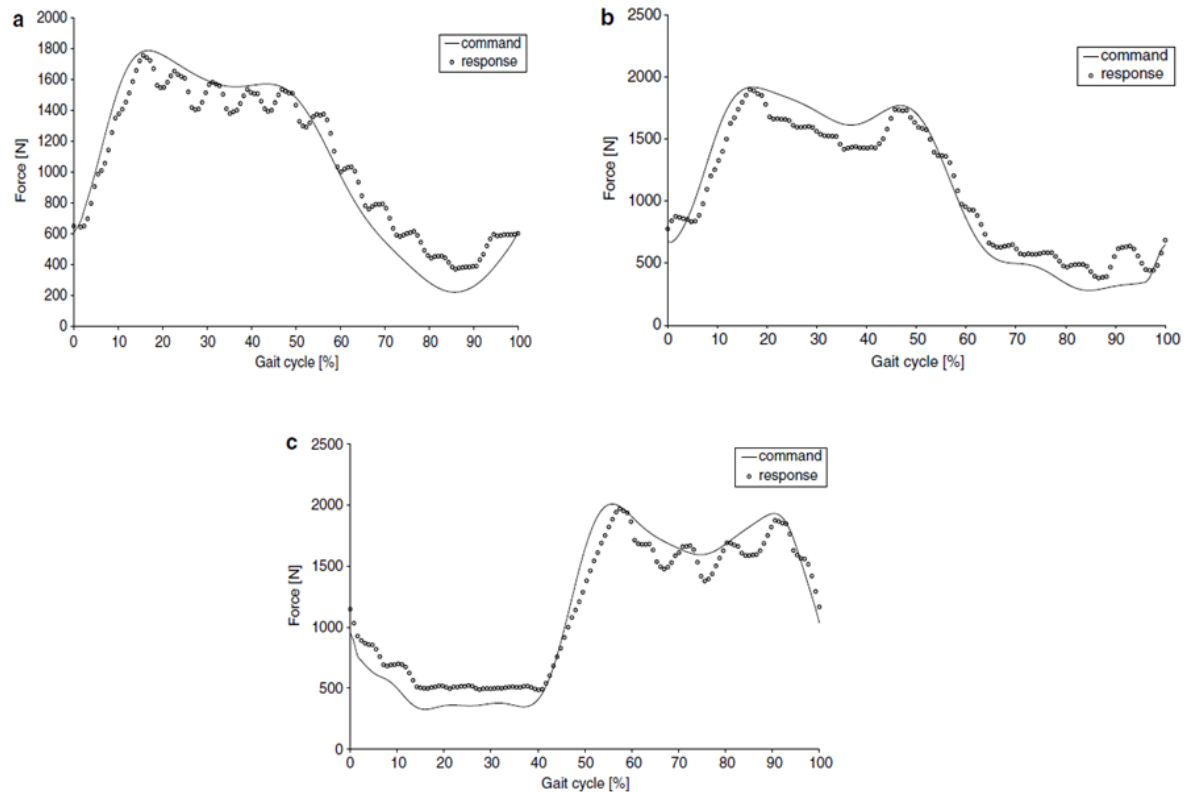


Figure 2. 19: Comparison of the hip contact forces as demand signal (line) and response of the machine (symbol) for normal walking (a), ascending stairs (b), and descending stairs (c) (Zant et al., 2008).

Zant et al. (2008) & Tong et al. (2008) reported some preliminary results obtained from three bovine acetabular reconstructs tested up to 17 million cycles using a new hip simulator. Damage was monitored using CT scanning at regular intervals, and the samples were then sectioned and analysed using conventional microscopy. Debonding at the bone-cement

interface was observed in all samples, irrespective of the loading conditions. Local debonding in the posterior-superior quadrant was revealed in one sample tested under normal walking for 15 million cycles; while multiple debondings were found at the bone-cement interface for another sample experienced 17 million cycles under normal walking loading condition. Extensive debonding at the bone-cement interface was identified by CT scanning in a sample tested under the most severe condition (Fig. 2.20a), descending stairs, for merely 2.2 million cycles, where clear demarcation between the cement and the bone can be observed in the sectioned sample (Fig. 2.20b).

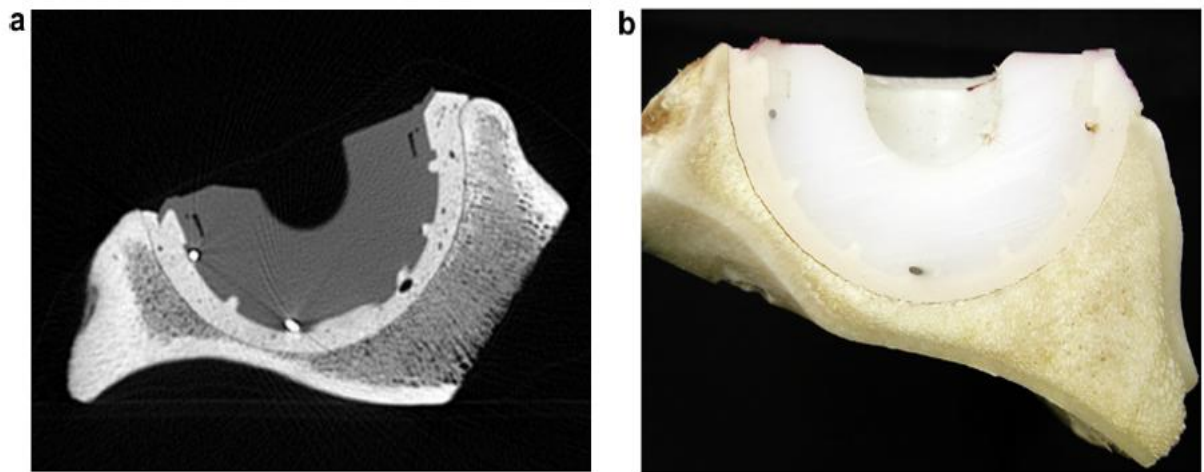


Figure 2. 20: Debonding at the bone-cement interface of a sample tested under descending stairs loading condition. (a) A CT scan image showing clear demarcation at the bone-cement interface; (b) almost complete separation of the bone-cement interface (Tong et al., 2008).

Wang et al. (2009) reported the effect of a combined load profile representative of patient routine activities on the endurance of cement fixation. The hip simulator developed by Zant et al. (2008) was utilised to study the long-term mechanical response of acetabular reconstructs under the combined loading blocks, as well as stair climbing and normal walking

loading conditions. The damage development in the fixation was periodically monitored by means of a μ CT scanner during testing. A body weight of 125kg was adopted to represent an upper bound load case and to reduce the test duration. All samples were CT scanned prior to the experiments and at half million cycle intervals throughout the experiments. The results showed that bone–cement interfacial debonding was identified as the predominant failure mechanism and that descending stairs was the most damaging loading condition (test terminated at ~3 million cycles), compared with the combined and the normal walking loading conditions.

The effect of environment on the survivor rate of a cemented acetabular reconstruct was attempted for the first time by Wang (2009). A preliminary environmental chamber was used to accommodate simulated body fluid (SBF) in the hip simulator to carry out pilot fatigue tests under wet conditions. The SBF was produced based on the formulation reported by Kokubo & Takadama (2006) and given in Table 2.4. A transparent plastic box was utilised to provide a sealed space for the specimen immersed in SBF and silicone gel was evenly spread over the gap between the box bottom and the rotation plate to avoid fluid leaking (Fig. 2.21).

Table 2. 4: Order, amounts, weighing containers, purities and formula weights of reagents for preparing 1000ml of SBF (Kokubo & Takadama, 2006).

Order	Reagent	Amount	Container	Purity (%)	Formula weight
1	NaCl	8.035 g	Weighing paper	99.5	58.4430
2	NaHCO ₃	0.355 g	Weighing paper	99.5	84.0068
3	KCl	0.225 g	Weighing bottle	99.5	74.5515
4	K ₂ HPO ₄ · 3H ₂ O	0.231 g	Weighing bottle	99.0	228.2220
5	MgCl ₂ · 6H ₂ O	0.311 g	Weighing bottle	98.0	203.3034
6	1.0M-HCl	39 ml	Graduated cylinder	—	—
7	CaCl ₂	0.292 g	Weighing bottle	95.0	110.9848
8	Na ₂ SO ₄	0.072 g	Weighing bottle	99.0	142.0428
9	Tris	6.118 g	Weighing paper	99.0	121.1356
10	1.0M-HCl	0–5 ml	Syringe	—	—

The fluid was circulated through a pump and the temperature was controlled at 37°C. The test was carried out for 18,000 cycles under a BW of 125Kg.

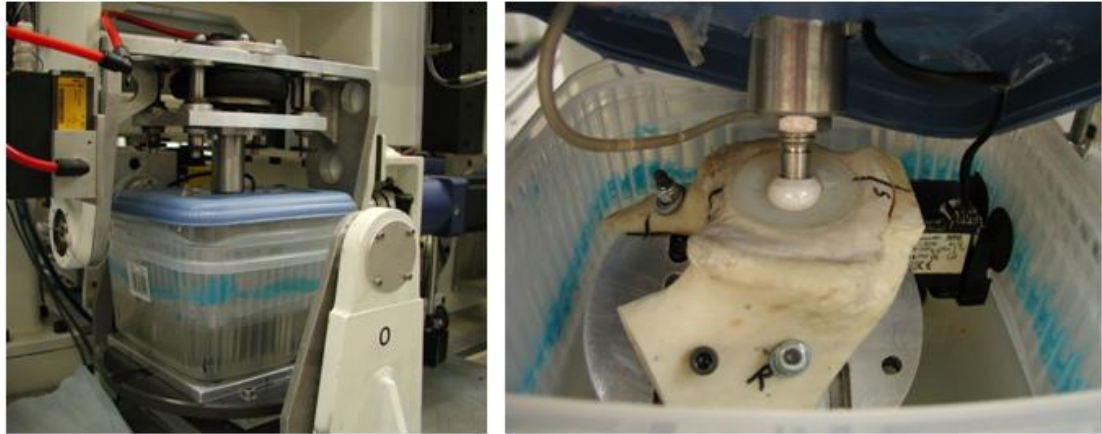


Figure 2. 21: A simple environmental chamber with SBF (Wang, 2009).

A full bone-cement interfacial debonding was clearly identified as a continuous demarcation at the bone-cement interface along the periphery of the acetabulum. Because a gross interfacial debonding was observed after only 18,000 cycles, no CT investigation was made hence the location of debonding initiation was not recorded.

2.5. Micro-focus computed imaging and relative applications

2.5.1. Background

Today X-ray computed tomography (CT) finds applications in different fields. Ketchem & Carlson (2001) divides CT scanners into four categories: conventional with a scale of resolution \sim mm; high-resolution with a scale of resolution \sim 100 μ m; ultra-high-resolution with a scale of resolution \sim 10 μ m; micro tomography (μ CT) with a scale of resolution \sim μ m. Medical CT scanners are generally in the conventional category, but some span the high resolution range. μ CT was pioneered in the late 1980's and is based on the same basic

principles as the common computed tomography (Sasov, 1987; Feldkamp et al., 1989). Generally speaking, the system consists of a micro-focus tube which generates a cone-beam of X-rays, a rotating specimen holder on which the object is mounted, and a detector system which acquires the images. One of the main differences to medical CT is that during the CT scanning, the source-detector geometry is fixed, while images are taken of the rotating specimen. Several systems are currently available on the market. μ CT systems used for in vitro imaging of small specimens and for in vivo imaging of laboratory animals (Sasov & Van Dyck, 1998; Ruegsegger et al., 1996; Sasov & Dewaele, 2002; David et al., 2003) are usually called Desktop CTs. For the first systems developed, examinations at a pixel size of $30\mu\text{m}$ were reported, with cross-sections of about $4\text{mm} \times 4\text{mm}$ (Sasov, 1987b). The first μ CT examinations of bone specimens (trabecular bone cubes, 8mm side) were performed at a resolution of $50\mu\text{m}/\text{pixel}$ (Kuhn et al., 1990). The main disadvantage of the Desktop systems is that the cabinets for material testing are small and aimed at hosting samples cut out from larger objects.

More recently, high resolution industrial μ CT systems were put on the market. These machines are able to combine a spatial resolution that can achieve cubic voxel sizes as small as $5\mu\text{m}$ (Ritman, 2004) with a larger chamber volume, allowing testing of a wide range of materials and dimensions. For in vitro imaging of trabecular bone biopsies, specimens having an external size of 8-10mm are normally examined, with a nominal spatial resolution of 14 to $30\mu\text{m}/\text{pixel}$ (Thomsen et al., 2005; Nagele et al., 2004; Hildebrand et al., 1999). Bigger acetabular reconstructs are usually of dimensions in the range of 20-25cm and imaged at $\sim 75\mu\text{m}$ isotropic resolution (Wang et al., 2009). Furthermore, the large dimensions of the CT chamber allow the use of specific testing devices for in-situ mechanical characterisation.

2.5.2. Basic principles

CT imaging offers a non destructive method of three-dimensional visualisation and characterisation of an object through the volumetric mapping of its X-ray attenuation (which is closely related to density). At the fundamental level, CT involves capturing a series of multiple X-ray views of an object at a range of angular orientations. X-rays are emitted from a source, pass through the object and the amount of attenuation (through scattering and absorption) of the X-ray signal is measured by a detector. Attenuation of monoenergetic X-rays is described by the Beer-Lambert law (ASTM, 2000).

$$I = I_0 e^{-\mu x} \quad (2.1)$$

Where:

- I_0 = the initial X-ray intensity
- I = the intensity of the X-ray after it has traversed a thickness of material, x
- μ = linear absorption coefficient for the material scanned

When this equation is applied to a non-homogenous material, the equation becomes:

$$I = I_0 e^{-\int \mu(s) ds} \quad (2.2)$$

Where the line integral is taken along the direction of propagation and $\mu(s)$ is the linear absorption coefficient as a function of distance along the ray path. As the attenuation coefficient is also generally a function of X-ray energy, the complete solution requires solution of the equation over the range of the effective X-ray spectrum.

$$\int \mu(s) ds = -\ln\left(\frac{I}{I_0}\right) \quad (2.3)$$

To reconstruct a three dimensional image, a series of radiographs taken at different angular orientations is processed so that the local value of attenuation is determined at every point of the sample volume.

Artefacts and partial volume effects can appear during tomography, obscuring details of interest and introducing errors for quantitative analysis (Ketcham & Carlson, 2001; Barrett & Keat, 2004). These errors include:

Beam hardening: As a polychromatic beam passes through an object, the X-rays with lower energy are more easily attenuated than the higher energy X-rays, for this reason a beam will preferentially lose the lower end of its spectrum. This increase in mean energy is defined as beam hardening. This is the most common artefact encountered in CT imaging and causes an object to appear brighter at its edges than at the centre. Filtration and software can be used to eliminate the effects of beam hardening.

Aliasing: This appears as streaks in the end images at the corner of objects and is a result of steep intervals in projections.

Detector saturation/photon starvation: The detector signal needs to be proportional to the photon flux. If the detector is saturated, or the signal is attenuated too much, then streaks may appear on the resulting image.

Ring artefacts: These very common artefacts appear as full or partial circles centred on the rotational axis. They occur in third generation scanners (rotating X-ray tube and detector assembly) as a result of a shift in output from individual or sets of detectors which cause the corresponding rays to have anomalous views.

Partial volume effect: This is the blurring of material boundaries as a result of the voxels at the boundaries consisting of an aggregate value of X-ray attenuation of the two materials.

2.5.3. Imaging and determination of morphometrical parameters of trabecular bone

Trabecular bone can be studied at different hierarchical levels, from the ultrastructure of collagen and mineral to macroscopic density (Wang et al., 2002; Hoffler et al., 2000; Aerssens et al., 1997; Ding et al., 2002). The architecture of trabecular bone is studied at the scale of individual trabeculae, at a resolution in the range from of 20 μ m to 50 μ m (Cowin, 2001). The spatial arrangement of the trabecular structure in trabecular bone is not random, as some regions are very dense, whereas others have only sparse trabeculae. The mean orientation and degree of anisotropy are also variables that change between anatomical site and between subjects (Ohman, 2011). Furthermore, in some regions the trabeculae are coarse and in others they are fine (Fig. 2.22).

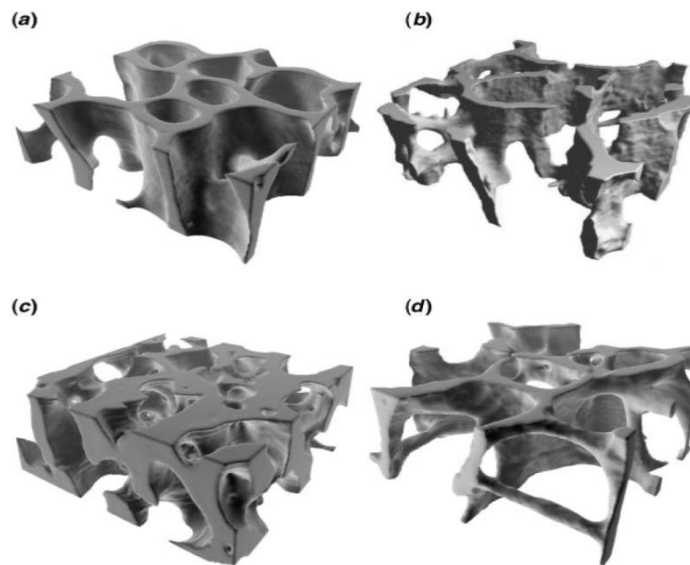


Figure 2. 22: The different trabecular structure of: (a) bovine proximal tibial, (b) human proximal tibial, (c) human femoral neck and (d) human vertebral trabecular bone (Keaveny et al., 2001).

A typical procedure for μ CT imaging and quantification of trabecular bone structure consists of three steps:

- Acquisition of the projection data;
- Reconstruction;
- Segmentation of the images and calculation of the morphometric parameters.

The bone specimen is placed on the rotation stage in the μ CT scanner and a series of frontal images of trabecular bone samples are acquired using selected settings (kV, μ A, filtering for beam hardening minimization, exposure time, rotation step). From the frontal projection data it is possible to obtain a reconstruction of the cross-section images of the specimen using dedicated software, which is based on a cone beam algorithm (Feldkamp et al., 1984).

For the calculation of the morphometric parameters, the grey-level image has to be segmented into a bone and non-bone phase. This process, also called segmentation, thresholding or binarization, is of vital importance in the assessment of the structural properties of trabecular bone (Hara et al., 2002). Different segmentation methods are presented in literature, such as global and local methods (Khun et al., 1990; Muller et al., 1994; Ding et al., 1999; Hildebrand et al., 1999; Hara et al., 2002; Waarsing et al., 2004; Kim et al., 2007; Perilli et al., 2007). Global methods are histogram-driven methods, where the grey-level histogram of an image can be represented as a bimodal curve (Fig. 2.23). A threshold could then be set in a fixed position between two consecutive peaks, with all the grey-levels above the threshold value identified as bone and those below the threshold value non-bone. The value that is used as a threshold is selected either visually, by analysing the histogram of CT numbers or by forcing the resulting binary data set to have the same volume

fraction of the original bone sample as determined experimentally (Ding et al., 1999). Local threshold methods are defined when a different threshold is used for different regions in the image and are based on the gradients in the grey values encountered at the bone/non-bone interface. Examples of this method can be found in the works of Kuhn et al. (1990) and Waarsing et al. (2004). These methods principally aim at overcoming problems in the segmentation of images with a limited resolution (i.e. partial volume effect) or low signal-to-noise ratio (i.e. in-vivo imaging). Once the images are segmented, the morphometric parameters can be calculated.

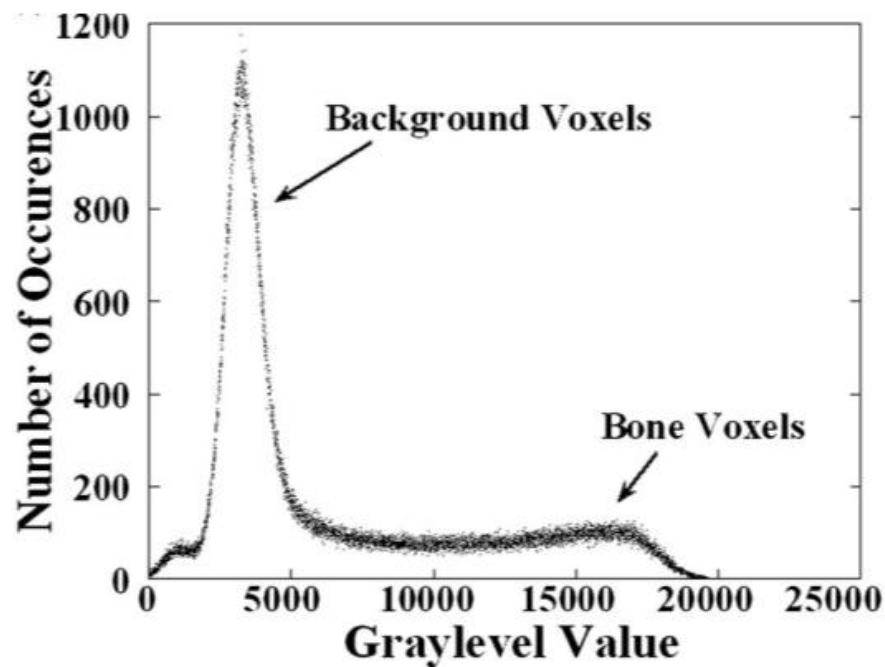


Figure 2. 23: Histogram of a typical 16-bit 3D CT image of bovine trabecular bone. The left peak represents background voxels, whereas the right peak represents bone voxels (Kim et al., 2007).

Bone Volume Fraction (BV/TV)

Bone volume fraction (BV/TV) is defined as the sum of all voxels marked as bone divided by the sum of all voxels (i.e. the total volume of interest). BV/TV is calculated directly from the image or the volume in examination.

With the development of CT-imaging techniques, it became possible to develop new image tools that took full advantage of all three dimensions (Hildebrand & Ruegsegger, 1997, Odgaard, 1997). Such procedures are given by the direct surface and volume determination, allowing a direct calculation of bone morphological parameters such as bone trabecular thickness (Tb.Th) and trabecular separation (Tb.Sp).

Trabecular Thickness (Tb.Th)

Model independent thickness (Hildebrand & Ruegsegger, 1997) is based on the estimation of volume-based local thicknesses, calculated as the diameter of the largest sphere which can locally be fitted completely inside the structure. From these local thicknesses, a volume-weighted mean thickness of the structure is calculated.

Trabecular Separation (Tb.Sp)

The Tb.Sp (Hildebrand & Ruegsegger, 1997) is calculated with the same procedure used for the Tb.Th, but this time the voxels representing non-bone parts are filled with maximal spheres. The separation is thus the thickness of the cavities.

As well as the mechanical properties, the morphological parameters for cancellous bone can vary with a number of conditions such as anatomical location, species and disease state as

reported by Keaveny et al. (2001). A range of values reported in literature for bovine, human healthy and osteoporotic bone is shown in Table 2.5.

The same morphological investigations can be performed for all foam materials with the following modifications:

- $BV/TV = FV/TV$
- $Tb.Th$ = Strut diameter
- $Tb.Sp$ = Cell spacing

Table 2. 5: Selected morphological parameters reported in literature for (1) healthy, (2) osteoporotic and (3) bovine trabecular bone.

Healthy bone (1)		Osteoporotic bone (2)	
Properties	Range of values	Properties	Range of values
BV/TV (%)	11.8-27	BV/TV (%)	5.8-18
Tb.Sp (mm)	0.4-1.11	Tb.Sp (mm)	0.871-2.184
Tb.Th (mm)	0.095-0.213	Tb.Th (mm)	0.09-0.236

Bovine bone (3)	
Properties	Range of values
BV/TV	24.4-30.4
Tb.Sp	0.589-0.613
Tb.Th	0.177-0.186

(1) : Parfitt et al., 1983; Silva & Gibson, 1997; Fazzalari & Parkinson, 1998; Majumdar et al., 1998; Homminga et al., 2002; Ohman et al., 2007; Renders et al., 2007; Busse et al., 2009.

(2) : Silva & Gibson, 1997; Cendre et al., 1999; Homminga et al., 2002; Busse et al., 2009; Ding et al., 2009.

(3) : Lim & Hong 2000; Verhulp et al. 2008; Garrison et al., 2009; Guillen et al., 2011.

2.5.4. Image-guided failure assessment (IGFA)

μ CT imaging combined with mechanical testing has been recently used to examine the damage behaviour of several engineered and biological cellular solids (Nazarian & Muller, 2004; Nazarian et al., 2005; Nagaraja et al., 2005). These techniques are relatively new and research in this area is quite limited. Nazarian & Muller (2004) were the first to introduce this particular experimental protocol. They developed a novel micromechanical testing system, composed of a microcompression device and a material testing and data acquisition system, to perform step-wise testing of trabecular bone and aluminum foam specimens based on image-guided failure assessment (IGFA) analysis. The specimen was compressed and imaged in a time-lapsed fashion, allowing the visual assessment of fracture initiation and progression not only in the elastic region but also beyond yield in the plastic region (Nazarian & Muller, 2004; Nazarian et al., 2005).

Measurements corresponding to each compression step were stored in 3D image arrays with isotropic voxel sizes of 34 μ m, and a three-dimensional Gaussian filter with a filter used to suppress the noise in the volumes. In the next step, the cellular solid images were segmented from background using a global thresholding procedure. The 3D images of each compression step were combined into an animation, since 3D animations of testing experiments contribute significantly to the understanding of sample failure under load. For this purpose, the 3D images first had to be aligned with respect to the bottom end-plate fixed during the experiment. An algorithm was used to find the last plane of this end-plate in each 3D image enabling an alignment of the images along the perpendicular axis. A subsequent 2D correlation procedure was used in the first five sample planes in order to perform the alignment. These aligned images were then visualized under the same conditions

(orientation, light settings) and the resulting images were finally turned into an animation (Nazarian & Muller, 2004; Nazarian et al., 2005).

Fig. 2.24 shows time-lapsed images obtained from the step-wise microcompression of a representative open-cell aluminum foam (Fig. 2.24a) and a whale trabecular bone cylinder (Fig. 2.24b), respectively.

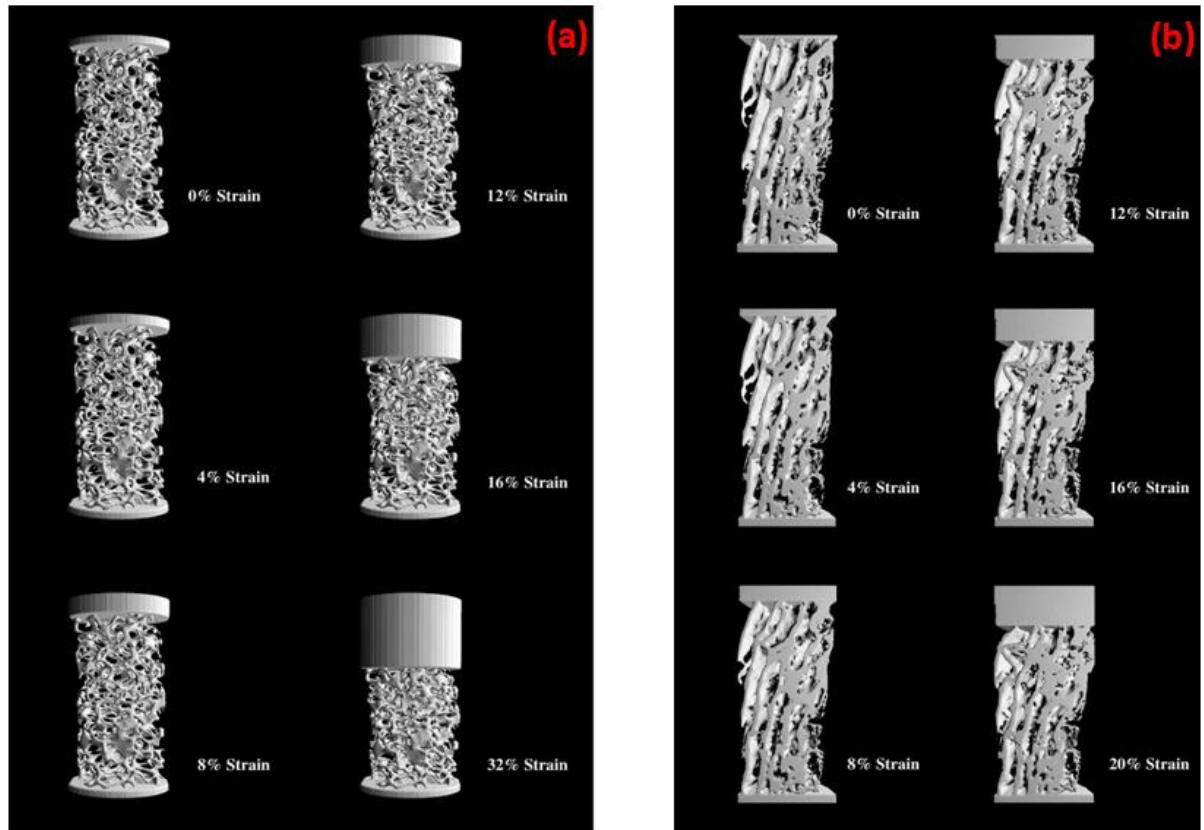


Figure 2. 24: IGFA of representative open-cell aluminium foam (a) and whale bone core (b) (Nazarian & Muller, 2004).

For both types of materials localised damage was observed to accumulate and propagate. In aluminum specimens (Fig. 2.24a) a general trend of buckling was observed within the structure as a consequence of the imposed strain, where the individual trabeculae tended to bend and buckle. Failure was observed somewhat differently in whale trabecular bone

specimens (Fig. 2.24b). Over the imposed strain range of 20%, the top third of the specimen crumbled significantly with the individual trabecular elements experienced high levels of deformation, whereas the lower part of the specimen remained relatively intact. It can be observed that bone failed in a brittle manner due to the presence of minerals but then showed ductile post-failure behaviour. It seems that the collagen network within the trabeculae can keep the mineral phase together over a long range of strain (Nazarian & Muller, 2004). Additionally, in all time-lapsed images failure progression of individual elements could be followed in both the aluminum and bone samples, as shown in Fig. 2.25 (Nazarian & Muller, 2004).

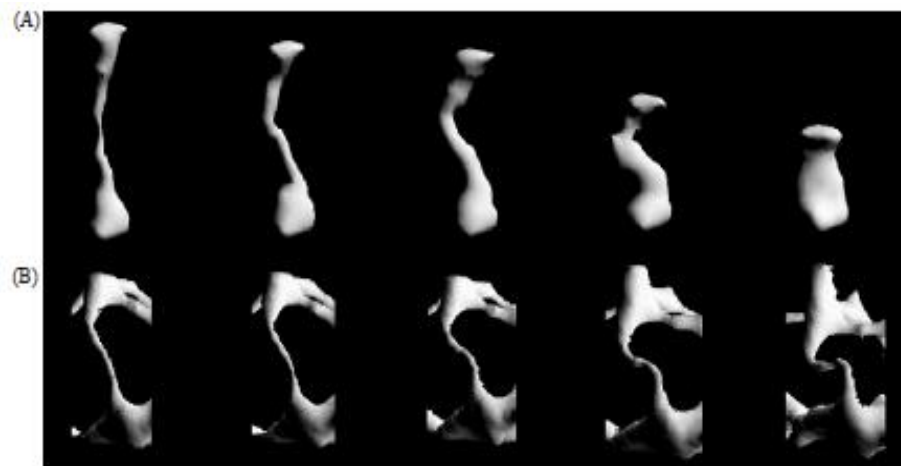


Figure 2. 25: IGFA of local failure: (a) rod-like element undergoing large local strain; and (b) although inter-nodal strain can be small, large deformation can occur (Nazarian & Muller, 2004).

The IGFA technique has been so far used for cancellous bone and cellular materials. The first implementation, to date, of such technique for bone-cement composites will be reported and discussed in Chapter 4.

Chapter 3

Damage evolution in
cemented acetabular
reconstructs

3.1. Introduction

The main findings discussed in this chapter have been reported in the following paper:

- Tozzi, G., Lupton, C., Heaton-Adegbile, P., Tong, J., 2012. Damage evolution in acetabular reconstructs under physiological testing in a saline environment. J. Biomech. 45, 405-408.

In this chapter the effects of saline environment on the survival rate of two cemented acetabular reconstructs were studied, following some preliminary work of Wang (2009). A new modular environmental chamber was developed and used on the Portsmouth hip simulator. μ CT imaging was carried out at regular intervals of the experiments, to systematically monitor the damage till gross loosening occurred. The results were compared with those obtained in dry conditions.

3.2. Methods

3.2.1. Specimen preparation

Fresh bovine bones were retrieved from a local abattoir from cows of ages between 12 to 20 months. Prior to implantation, the bones were treated and all fatty tissues were removed to ensure the cleanliness of the tissues for long duration testing as well as for effective interdigitation of the cement. Sequential reaming of acetabulum was performed, according to standard orthopaedic procedures, at a step of 2mm until the desired socket size was obtained. During the reaming operation the specimen was kept in a stable position by means of three clamps at the iliac, ischial tuberosity and pubis. Standard cementing technique was utilised to prepare and apply a mix of CMW1 (DePuy CMW, UK) cement to secure a polyethylene cup (Charnley Ogee DePuy Int, UK) in the standard position. An average thickness of the cement mantle was 2-3mm, and the cup size used to suit the size of the bones was 53/22. Two hemi-pelvic reconstructs (right and left) were used in this study (Fig. 3.1).

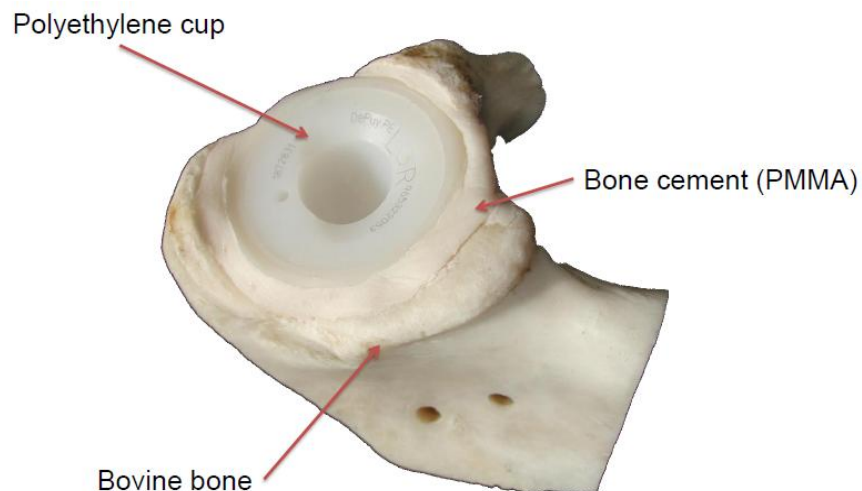


Figure 3. 1: The acetabular reconstruct implant used in this study.

3.2.2. Environmental chamber design and implementation

A new modular chamber was designed in this study using Pro/Engineer wildfire 4.0 software (PTC, USA). The basic idea was to develop a fully waterproof chamber, able to be closed around the specimen already fixed in the optimal desired position on the base rotation plate. Another important issue was to design the components in order to be assembled and sealed readily without the use of silicone gel. Figure 3.2 shows the 3D model of the chamber with all its individual components and assembled. The chamber consists of five components: a circular base plate (Fig. 3.2a), two hemi-cylindrical parts (Fig. 3.2b-c), and two transparent top disks (Fig. 3.2d-e). The components, with canals on the edge to accommodate rubber rings for sealing purpose, are assembled using bolts. Four blind holes in the central part of the base plate allow the fixation of the sample on the base plate. The base plate of the chamber is connected with the simulator plate through the four circumferential holes.

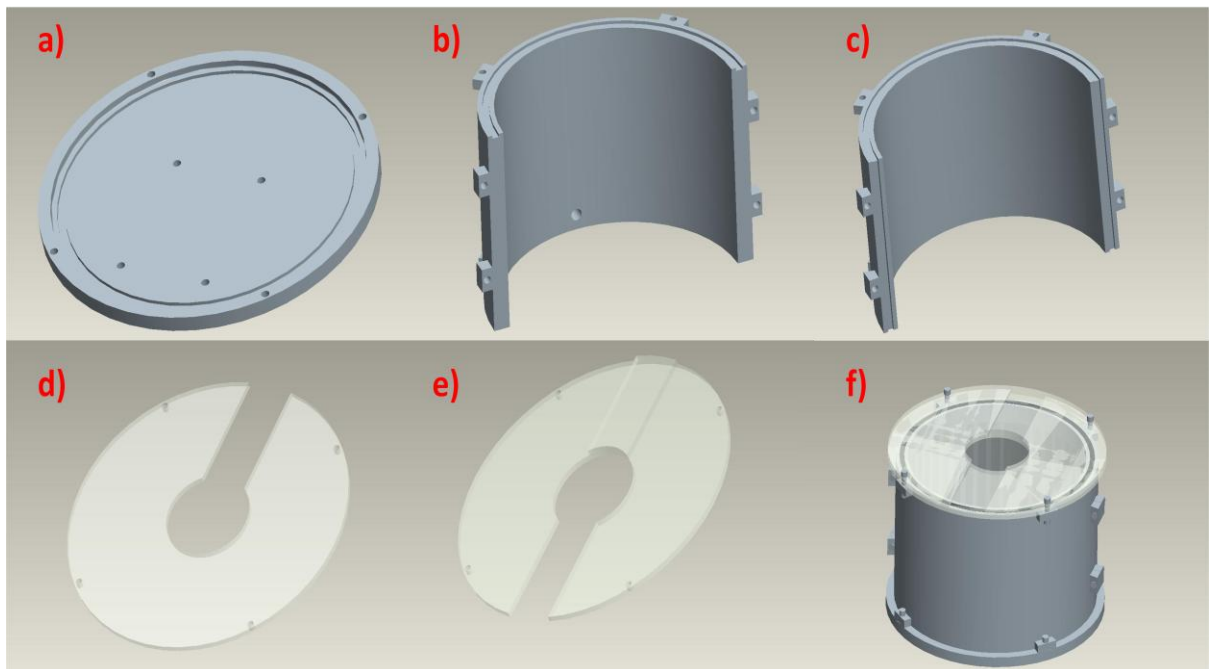


Figure 3. 2: The new environmental chamber with its five modular components (a-f) and assembled (f).

The modular components of a chamber prototype were then manufactured (Hillcrest Machinery & Engineering, UK) and installed on the hip simulator for a pilot testing of acetabular reconstruct under physiological loading conditions in a wet environment (Fig. 3.3). After some preliminary tests minor modifications were introduced to the original drawings and a final version of the chamber (n=3) was manufactured and used on the hip simulator.



Figure 3. 3: The prototype environmental chamber installed on the hip simulator.

3.2.3. μ CT system

The μ CT used in this study is a XT H 225 real-time X-ray inspection system (X-Tek Systems Ltd), which has both programmable and manual functions. The system has a 225kV X-ray source, feature recognition down to 4 μ m and maximum volume measurement of \varnothing 250mm and 600mm height. The XT H 225 has 3 linear axes and 2 rotational axes allowing a view of the sample at any angle. The software used for CT 3D reconstructions is VG Studio Max 2.0 (Volume Grapics, GmbH). Fig. 3.4 shows the machine description.

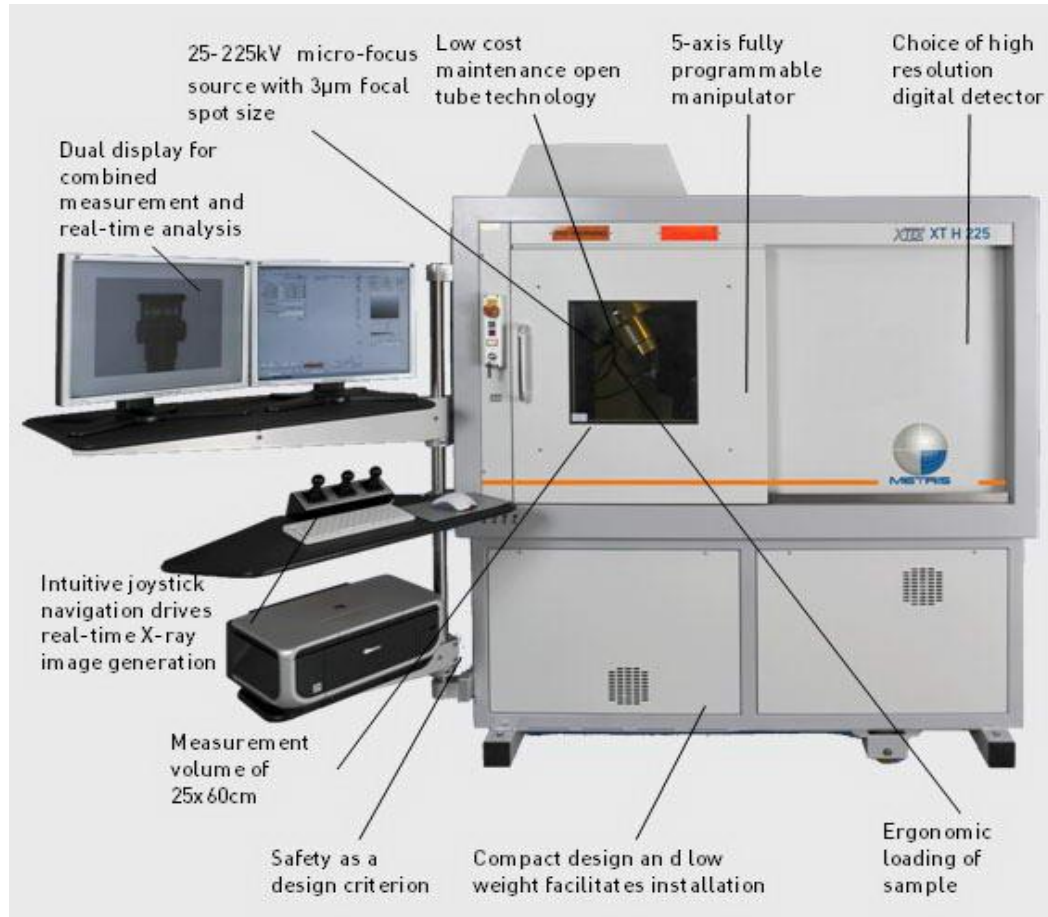


Figure 3. 4: The XT H 225 µCT system description (www.nikonmetrology.com).

3.2.4. Experimental procedures

The novel modular environmental chamber was installed on the hip simulator to allow testing of the implants in wet condition. Prior to testing a three-point fixation was used to stabilise the samples on the base plate and to achieve an optimal alignment with the femoral head of the simulator (Fig. 3.5a). The chamber was then assembled around the specimen (Fig. 3.5b), into which a saline solution (0.9% NaCl) was filled, as shown in Fig.3.5c. The temperature was kept constantly at 37°C by a control system consisting of heating bar, temperature sensor, and a central control box. A pump was utilised to inject the saline

solution directly onto the top surface of the specimen and to allow fluid circulation during the test, therefore preventing abnormal high temperatures at the interfaces.

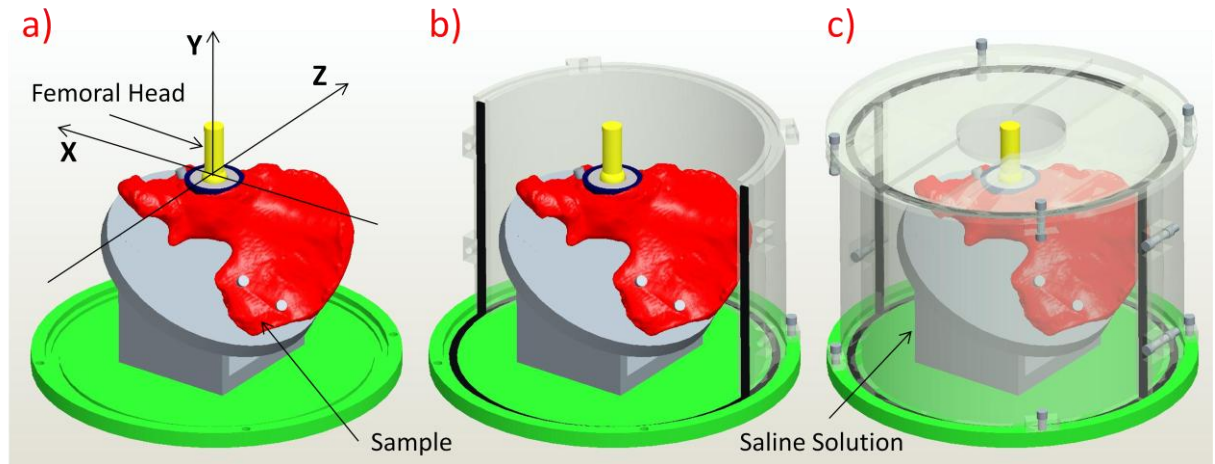


Figure 3. 5: Schematic representation of the experimental setup. The sample was first fixed onto the base plate (a), the chamber was then assembled around the specimen (b), into which a saline solution (0.9% NaCl) was filled (c). The solution was kept at 37°C and circulated during testing (Tozzi et al., 2012).

A combined loading block (Fig. 3.6) composed of four routine activities (normal walking, stairs down, standing 2-1-2 legs and stairs up) was adopted by Wang et al. (2009) from the data of Bergmann et al. (2001). A body weight of 125kg was chosen to represent an upper bound load case and to reduce the test duration. The resultant force vector was reproduced with an average frequency of 0.6Hz for a combined block.

The effect of saline environment on the cement fixation was assessed by periodically removing the sample from the hip simulator, and examining it by means of μ CT. The samples were scanned before testing and at every 100,000 cycles till gross loosening occurred. The scanner was set to a voltage of 125kV and a current of 25 μ A. With an isotropic voxel size of 75 μ m the image acquisition process was performed at a rotational step of 0.19° over 360° for approximately 90min and 3D reconstruction of the samples was obtained.

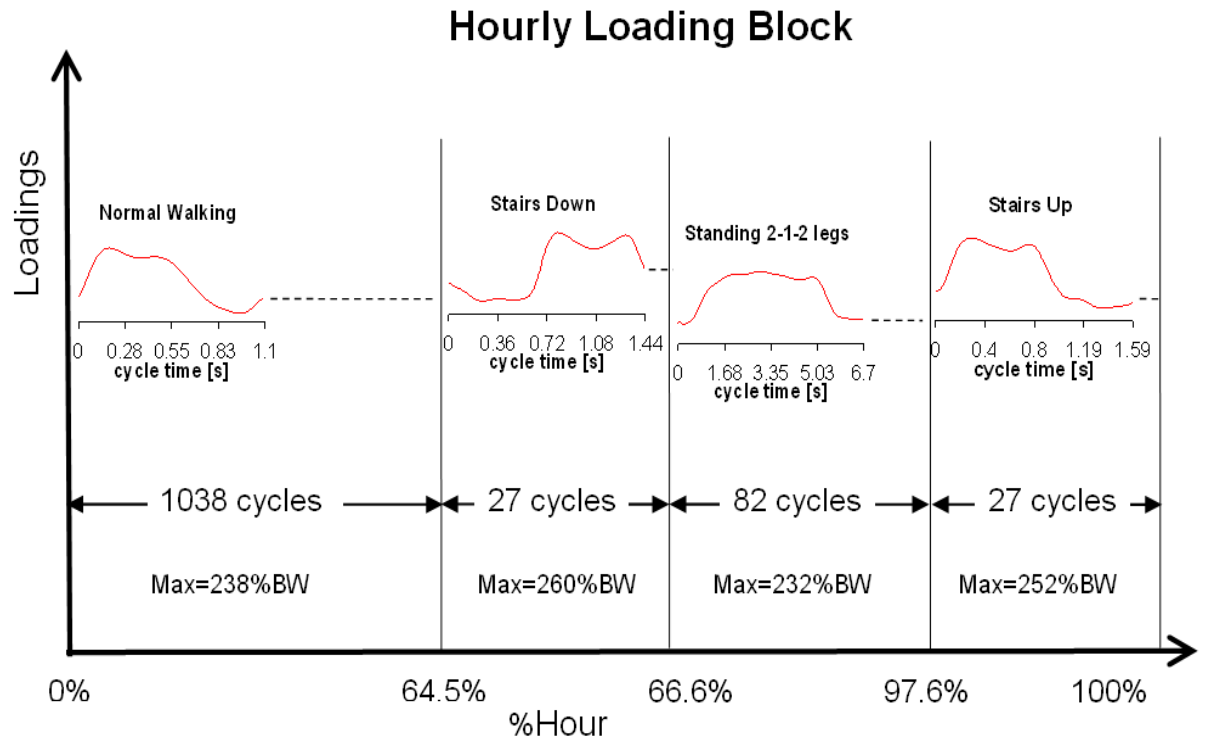


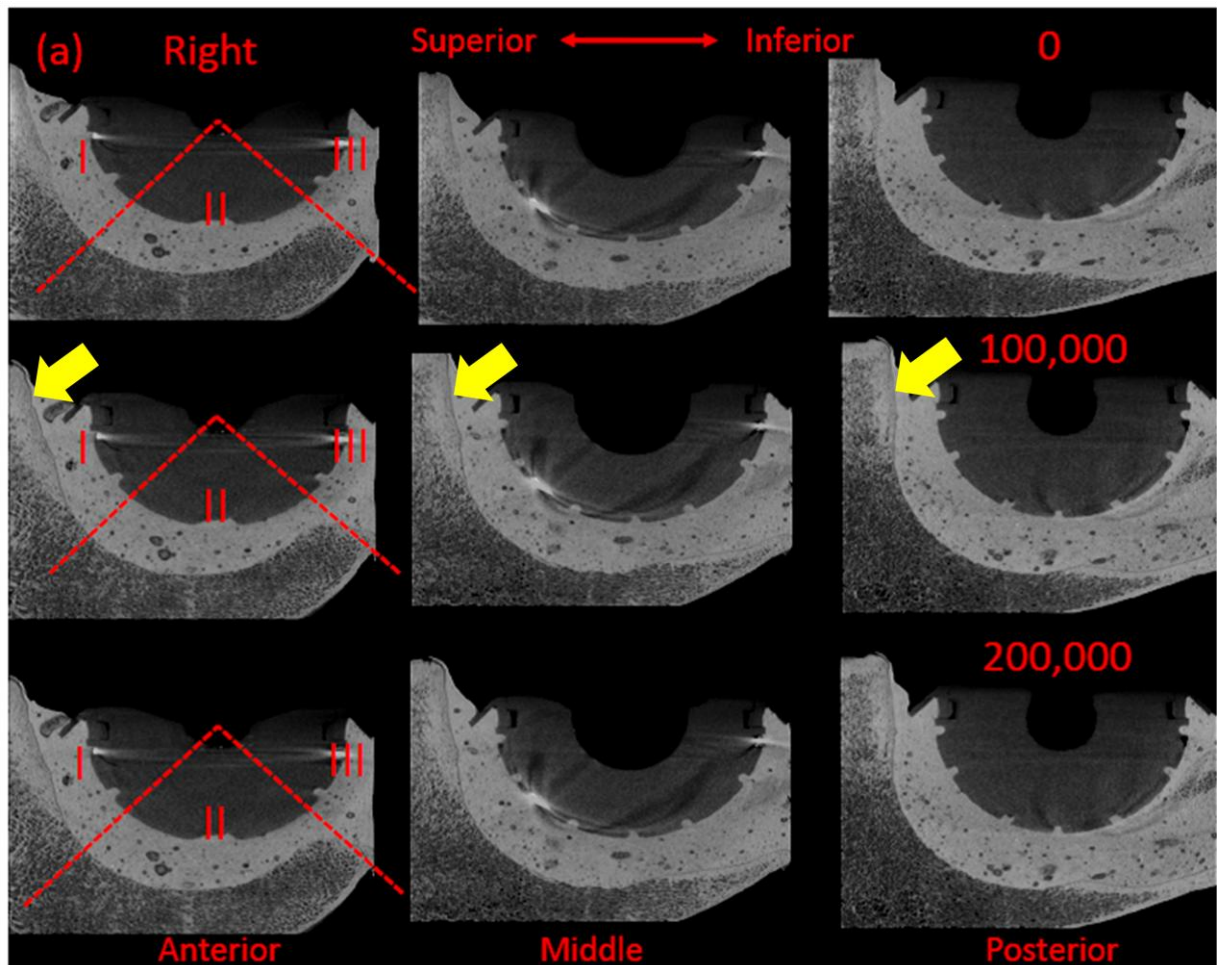
Figure 3. 6: The combined loading block built from the four physiological activities as an hourly loading block that was used in the current testing (Wang et al., 2009).

Two dimensional projections in transverse, sagittal and coronal planes were extracted from the 3D data for damage analysis evaluation in the zones described by DeLee & Charnley (1976).

3.3. Results

The μ CT scan images prior to testing and during testing (at 100,000 and 200,000 cycles) are presented in Figure 3.7 ((a) for the right implant, (b) for the left implant), where DeLee & Charnley zones (DeLee & Charnley, 1976) are also indicated.

Prior to testing, both hips examined showed a bone-cement interface that appeared to be fixed and intact. At 100,000 cycles, some evidence of demarcation may be observed starting from the bone-cement interface in the superior quadrant or from DeLee & Charnley zone I (yellow arrows) for both samples and in particular for the left one, as shown in Fig. 3.7.



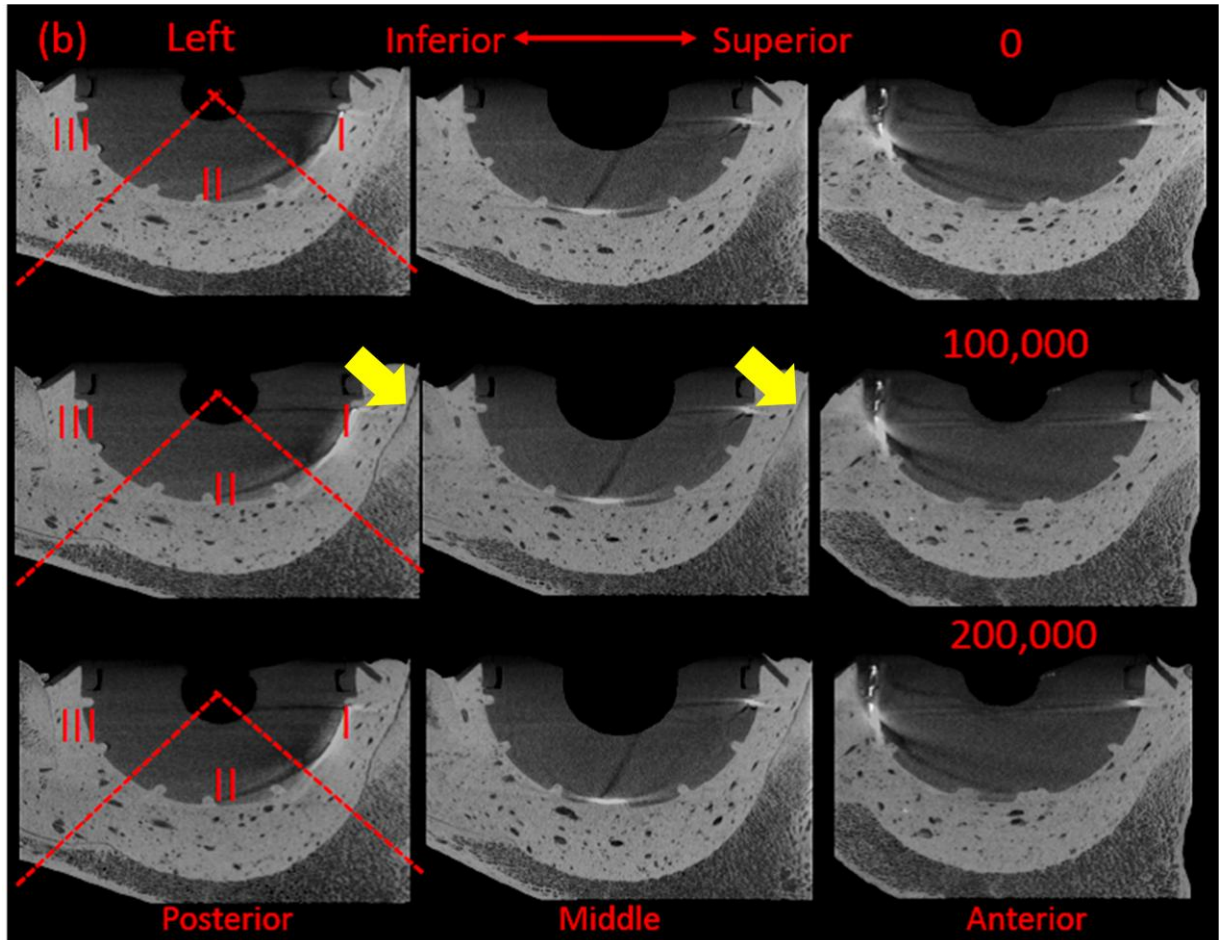


Figure 3. 7: μ CT sectional reconstructions recorded at selected loading intervals for (a) right and (b) left implanted hips. The images are shown with the DeLee & Charnley zones identified. For both samples, the bone-cement interface appeared to be intact prior to testing. Interfacial debonding between the bone and the cement seemed to initiate in zone I in the superior quadrant, as indicated by the arrows (Tozzi et al., 2012).

It seemed that debonding started from the edges of the bone-cement interface in the superior rim of the reconstructs (zone I) and developed towards zone II, although at 200,000 cycles debonding had yet to reach the dome region in either cases. Damage progression from 100,000 to 200,000 cycles seems to be small, although the initiation of debonding may have started well before 100,000.

3.4. Discussion

The results from preliminary testing of acetabular reconstructs in a saline environment and otherwise under the same combined loading blocks suggest that, although progressive debonding at the bone-cement interface defines failure in both wet and dry conditions, some differences may be pointed out. Firstly, reduced survival lives, i.e. about 100,000 to 200,000 cycles, were observed for the two reconstructs tested in wet conditions, as opposed to 3 to 4 million cycles under the same combined load block in dry conditions (Wang et al, 2009). Secondly, the initiation of the debonding seems to occur in DeLee & Charnley zone I, consistently with clinical observations (Garcia-Cimbrelo & Munuera, 1992), as opposed to zone II observed in dry conditions (Zant, 2006). Figure 3.8 illustrates these differences. Given that the implantations were carried out consistently (despite some differences in the average cement mantle thickness achieved during implantation), and the mechanical loading conditions remained the same, the substantial damage from an earlier stage and the change of the damage initiation pattern may be indicative of the role of environment in promoting loosening in this type of replacements.

Although the precise failure mechanisms at the bone-cement interface are to be established, the effects of moisture on the mechanical performance and fatigue behaviour of bone cements and bone-implant interfaces are known to be of importance. Both storage and testing environment, such as temperature and moisture, affect the mechanical properties of bone cement (Baleani et al., 2001; Nottrott et al., 2007). An increased strength and modulus of bone cement tested under compression in the first period of aging (up to 21 days) in wet conditions at 37°C was observed (Baleani et al., 2001; Nottrott et al., 2007). However, according to Nottrott et al. (2007), after only one week of soaking the mechanical properties

were found to decrease gradually due to the effect of water uptake, which has a plasticising effect but also acts as a crazing initiator, implying weakening and possibly affecting the cyclic stress distribution in the bone-cement-prosthesis complex. The degradation effects of moisture on adhesives in resin-dentin bonded systems were shown to increase with soaking time (Hashimoto et al., 2003; Reis et al., 2004). In the present study, a combination of micromotion generated at the bone-cement interface due to mechanical loading, cement weakening due to water uptake and moisture infiltration at the interface may have contributed to the early and progressive debonding of the fixation.

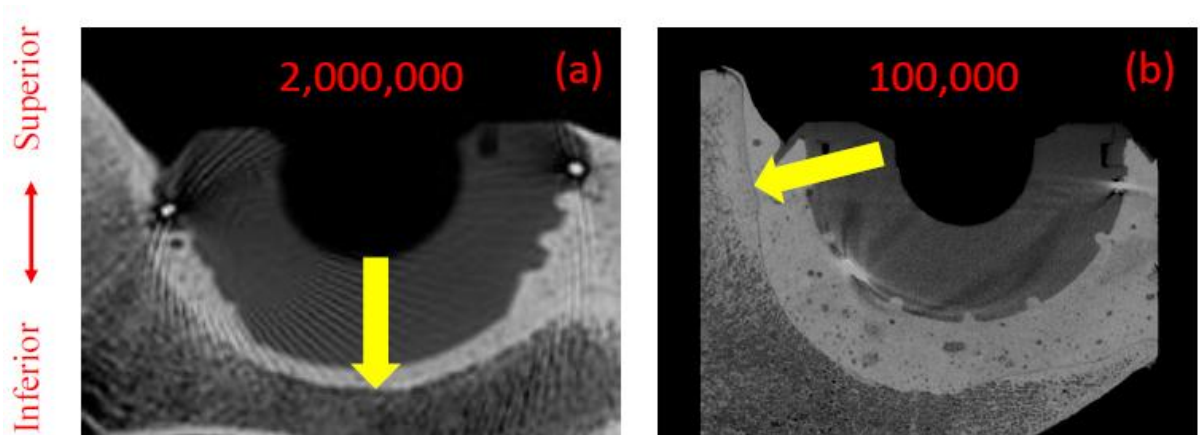


Figure 3. 8: μ CT images from a mid-section for (a) an implanted hip tested under normal walking in dry conditions (Zant, 2006), where debonding developed near the dome region (DeLee & Charnley zone II) at about 2,000,000 cycles; (b) an implanted hip tested in wet conditions (current), where a demarcation may be observed in DeLee & Charnley zone I at just 100,000 cycles, as indicated by the arrows (Tozzi et al., 2012).

Admittedly, this is a preliminary study with limited samples tested. However, the current results are consistent with the early results of Wang (2009), where debonding from DeLee & Chernley zone I & III was also observed at a much reduced number of cycles using SBF in a prototype environmental chamber. Another significant limitation is that the scanings were carried out on the specimen unloaded, where interface gaps may be closed or partially

closed. In addition, small interface gaps, such as those 1 to 2 voxels (75-150 μ m), would unlikely to be detected due to partial volume effects in the thresholding procedure. Biological aspects as well as long-term response of the bone to the loading environment in vivo were not considered. Factors such as residual fatty tissues or incomplete reaming may also contribute to the premature failure of an implant, as in the previous experiments (Wang et al., 2009) as well as in clinical cases, although the same test protocol was followed in both dry and wet tests.

3.5. Conclusions

Preliminary results from the endurance testing of cemented acetabular reconstructs in a saline solution seem to indicate that environment, when combined with physiological loading conditions, plays an important role in reducing the survival rate of cemented acetabular reconstructs. Although debonding at the bone-cement interface has been found as the predominant failure mechanism in both wet and dry conditions, damage in wet condition appears to occur much earlier and from DeLee & Charnley zone I, where the combined action of mechanical loading and fluid environment seems to be most effective.

Chapter 4

Bone-cement interfacial behaviour

4.1. Introduction

The main findings discussed in this chapter have been reported in the following papers:

- Wang, J.-Y., Tozzi, G., Chen, J., Contal, F., Lupton, C., Tong, J., 2010. Bone–cement interfacial behaviour under mixed mode loading conditions. *J. Mech. Behav. Biomed. Mater.* 3, 392–398.
- Tozzi, G., Zhang, Q.-H., Tong, J., 2012b. 3D real-time micromechanical compressive behaviour of bone-cement interface: Experimental and finite element studies. *J. Biomech.* 45, 356-363.

In this chapter, bovine cancellous bones were used to interface with acrylic bone cement and bone-cement interfacial behaviour was investigated at both apparent and micromechanical levels using different loading conditions and experimental techniques.

Bone-cement interface coupons were mechanically tested under mixed-mode conditions and the failed samples were examined using μ CT to identify the failure mechanisms associated with the loading angle. The main motivation of the study was to explore the relationship between the measurable parameters, such as the fracture strength and the failure mode, and the operational parameters, such as loading mode and cement penetration, so that the key parameters that dictate the interfacial behaviour may be identified.

Furthermore, compressive micromechanical behaviour of bone-cement interface specimens was investigated where the primary goals were: (1) To use a novel custom-made micromechanical loading stage and time-lapsed μ CT imaging to assess the local deformation and damage development in the bone-cement specimens under selected levels of uniaxial

compression; and (2) to evaluate the effects of cement penetration and bone morphology on the compressive response of the interface.

4.2. Methods

4.2.1. Specimen preparation

Bovine trabecular bones were harvested from fresh iliac crest and used to interdigitate with acrylic bone cement (CMW1, DePuy CMW & Simplex P, Stryker, UK) to create bone-cement interface samples. The rectangular bone coupons were machined to size using a low speed diamond saw with constant water irrigation and cleaned to remove fatty tissues. Each coupon was then placed into the lower half of a custom-made mould and bone-cement interface strips were prepared by applying a constant pressure to the cement ranging from finger packing (for the samples tested under mixed-mode conditions) to 60kPa (for the samples tested under step-wise compression), following a procedure of Graham et al. (2003). Bone-cement specimens were finally cut from the strips (Fig. 4.1).

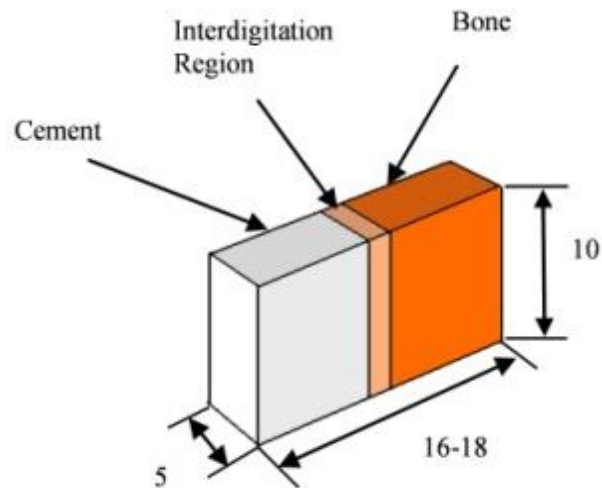


Figure 4. 1: The bone-cement interface sample used in this study (all dimensions are in mm).

4.2.2. Loading stage (LS) system

A novel custom-made (Deben Ltd, UK) micromechanical loading stage (LS), equipped with a 3kN miniature load cell, was used to load bone-cement interface composites under step-wise compression (Paragraph 4.4) within the μ CT (XT H 225 X-Tek, as described in Chapter 3) chamber (Fig. 4.2a). The loading stage is a modular system consisting of a carbon fibre cylinder and a modular actuator with a load cell. Mini modular clamps and platens allow the stage to be adaptable for different micromechanical testing. The carbon fibre cylinder structurally supports the actuator and with its low attenuation coefficient provides low interference effect on the radiographs during the acquisition (Baidya et al., 2001). Data acquisition software (Deben Microtest V6.0.21) was used for computer control of the LS and for real-time display and recording of the force-extension curves. A schematic of the LS is shown in Fig. 4.2b.

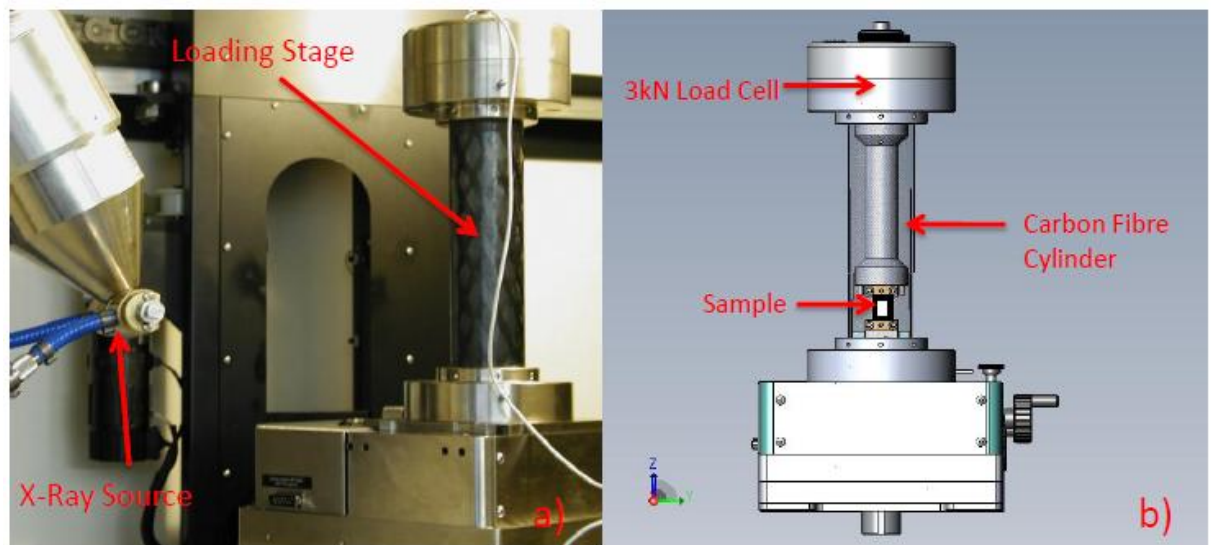


Figure 4. 2: (a) The custom-made micromechanical loading stage (LS) within the μ CT chamber; (b) a schematic of the LS, which was used to apply the load in stepped compression (Tozzi et al., 2012b).

4.3. Bone-cement interface under mixed-mode loading conditions

4.3.1. Experimental procedures

A loading device was designed (Fig. 4.3) to allow tensile ($\theta = 0^\circ$), shear ($\theta = 90^\circ$) and selected mixed mode ($\theta = 22.5^\circ$, 45° and 67.5°) loads to be applied. The specimens were fixed in custom grips at the prescribed loading angles and loaded to complete failure under displacement control at a rate of 0.01mm/sec (ESH 10kN Servo-hydraulic testing machine). The relative opening displacements were measured using a clip gauge, while the sliding displacements were measured using a video extensometer (Messphysik). The vector sums of the total displacements were calculated and used to plot load versus displacement curves. A servo-hydraulic testing machine was used for the experiments and all tests were conducted in air at room temperature. A total of 40 samples were tested at 5 selected loading angles (0° , 22.5° , 45° , 67.5° and 90°); 8 tests were carried out at each loading angle. The data were statistically analysed using ANOVA.

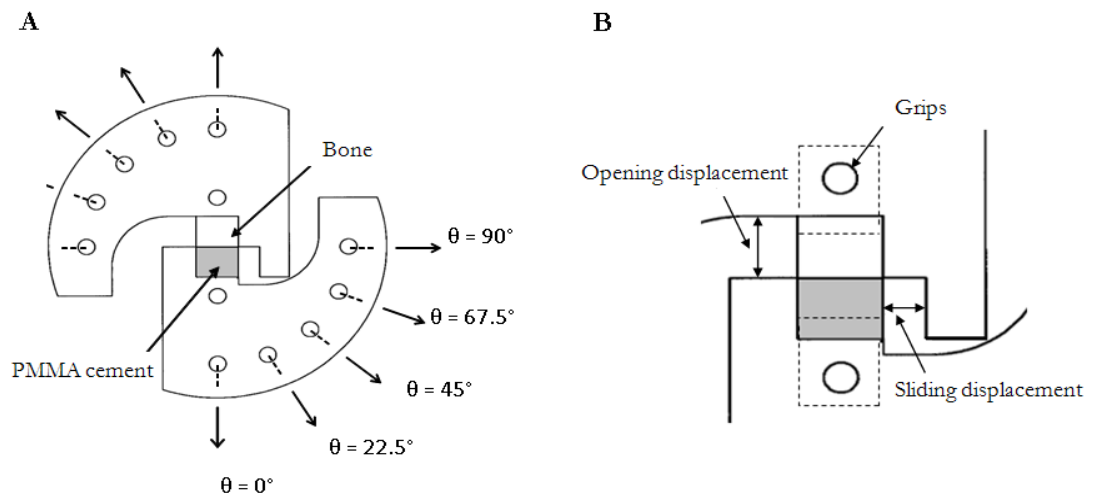


Figure 4. 3: (a) Schematic of the loading arrangement for tensile (0°), shear (90°) and mixed mode (22.5° , 45° , 67.5°) loading conditions. (b) The opening and the sliding displacements as measured using a clip gauge and a video extensometer, respectively (Wang et al., 2010).

Specimens post mechanical testing were examined using microscopy and μ CT (XT H 225 X-Tek, Ltd). The specimens were held vertically by a specifically designed clamp, and the scanning was performed at a rotational step of 0.19° , voltage of 55kV, current of 145 μ A and isotropic voxel size of 14 μ m. For each frontal and lateral cross-section, a rectangular region of interest (ROI) containing the cement-bone interface, was defined. The frontal ROI had a surface area of approximately 10mm \times 10mm, whereas the lateral one was of approximately 5mm \times 10mm. The CT images were processed using Mimics 13.0 software (Materialise, Belgium), and reconstructed by thresholding the cement and the bone regions based on CT grey scale intensity. The 3D reconstructions were utilized to assess the failure modes for the samples tested.

Failures were classified as one of the four types, following Miller et al. (2007):

Type I: Fracture occurred at the extreme of cement penetration with bone fragments remaining in the cement half of the sample.

Type II: Fracture occurred at the extreme of cement penetration with bone spicules remaining in the cement and cement fragments remaining in the bone.

Type III: Failure occurred beyond the extreme of cement penetration, but near the interface of the bone.

Type IV: Failure occurred at the extreme of cement penetration with cement fragments remaining in the bone but no bone spicules in the cement.

4.3.2. Results

The resultant applied load versus displacement curve for each specimen was recorded. All load-displacement curves exhibit a similar characteristic response, with an initial linear behaviour followed by non-linear hardening to reach peak load, then by substantial strain softening until complete debonding of the interface. A normalised apparent strength was used as a relative measure of interfacial strength and its relationships with the loading mode and the normalised cement penetration depth is shown in Fig. 4.4a, b. The average apparent strength was $4.09(\pm 3.66)$ MPa in shear compared with $1.48(\pm 0.85)$ MPa in tension.

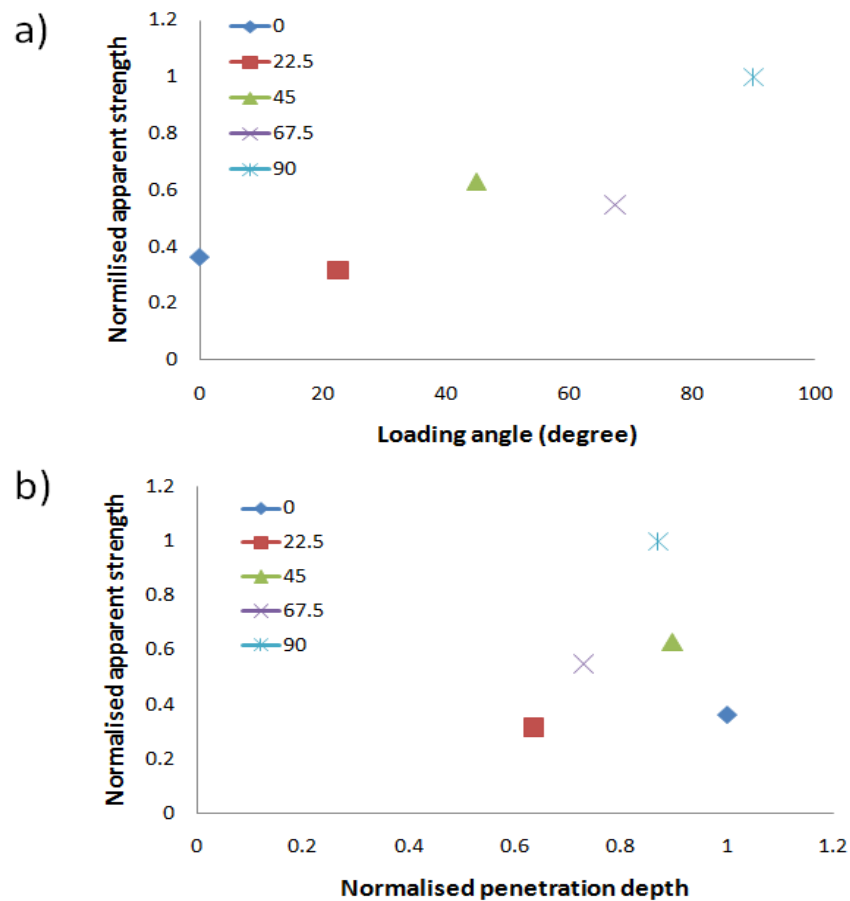


Figure 4. 4: (a) Normalised apparent interfacial strength as a function of the loading angle.
(b) Normalised apparent interfacial strength as a function of normalised cement penetration.

An average cement penetration was estimated in the present study. The cement penetration was measured at regular intervals of 0.5mm, from the edge of the specimen, following a method by Maher & McCormack (1999) and resulted to be $0.51(\pm 0.24)$ mm. The overall correlation (r^2) between the interfacial strength and the loading angle was found to be 0.77 (ANOVA, $p=0.07$) (Fig. 4.4a), while a value of 0.16 was obtained for the overall correlation with the depth of cement penetration (Fig. 4.4b). Interestingly, a correlation of $r^2=0.76$ was obtained for samples tested under shear loads; while for other loading angles this value is about $r^2=0.14$.

The morphologies of the failed samples are presented in Fig. 4.5, where the cement half of the samples was μ CT imaged. In the majority of the cases failure was found to be of either Type I or II for large loading angles, as opposed to Type IV or II for samples at low loading angles.

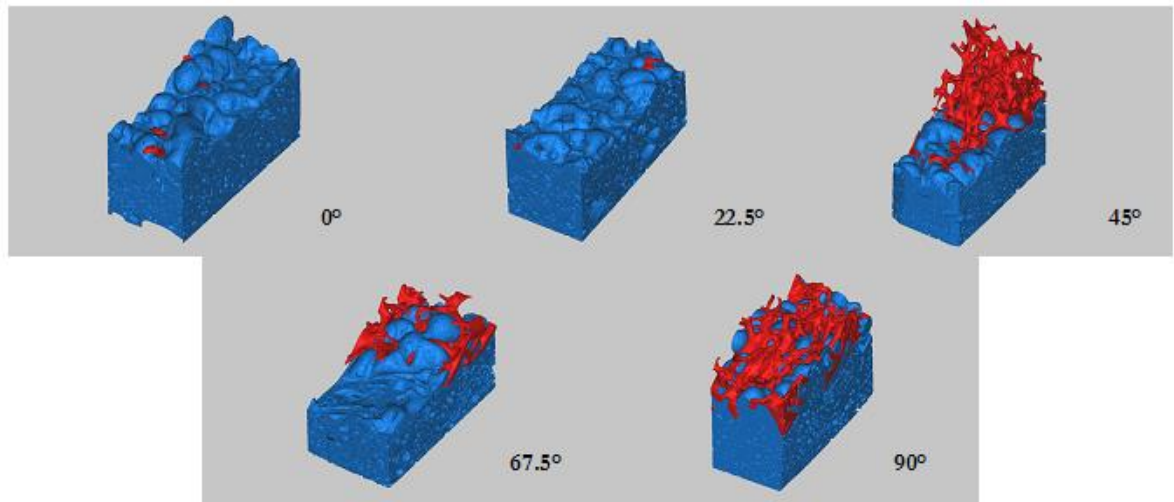


Figure 4. 5: Morphologies of some of the failed bone-cement interface samples tested under tensile (0°), mixed mode (22.5° , 45° , 67.5°) and shear (90°), examined using μ CT scanning. Note bone and cement are marked accordingly (bone-red; cement-blue) (Wang et al., 2010).

4.4.3. Discussion

There are significant variations between the results from specimens tested under the same or similar loading conditions, possibly due to multiple factors including the variation of the bone morphology, the effectiveness in cement penetration as well as local pre-existing damage at the interface. Nevertheless, some of the general trends may be observed. For example, despite the fact that the effect of the loading angle was not significant, there is a good correlation between the apparent interfacial strength and the loading angle ($r^2=0.77$) (Fig. 4.4a). The results are in broad agreement with those of Mann et al. (1999; 2001) in that the shear strength of bone-cement interface is greater than its tensile strength; and the loading angle significance with regards to the failure response was absent ($p>0.05$ (Mann et al., 2001)).

For cadaveric bone-cement specimens, the apparent strength in tension was reported to be 1.28 ± 0.79 MPa (Mann et al., 1997) compared with $1.48(\pm0.85)$ MPa from the present study; while for shear the present result of $4.09(\pm3.66)$ MPa is higher than the cadaveric result $2.25(\pm1.49)$ MPa (Mann et al., 1999), although a range of shear strength have been reported in the literature such as 4-11MPa from (Dohmae et al., 1988), 5-7MPa from (Bugbee et al., 1992) and 6-15MPa from (Bean et al., 1988).

The average cement penetration for tensile and shear tests in (Mann et al., 1997, 1999) was 1.5mm, while 2.2mm for mixed mode tests (Mann et al., 2001). These values are significantly higher than the average depth of penetration (0.51mm) achieved in the present study. Interestingly, however, the measured interfacial strength does not appear to correlate with the difference in the depth of cement penetration. There may be several reasons for this. Bone surface preparation techniques can affect the penetration depth, as reported in

Majkowski et al. (1993) from a depth of 0.2mm in unprepared bone, 0.6 to 1.4mm for brushing with surface irrigation to 4.8 to 7.9mm achieved when pressurized lavage. The effectiveness of cement penetration may be significantly affected also by blood circulation in vivo, a difference between 0.1 to 4.6mm in the presence of circulation and 0.8 to 8.2mm in the absence of circulation was reported (Juliussen et al., 1995). Although the penetration depth increased with the increase of pressure, there was a large variation in the penetration depths achieved under the same condition (Juliussen et al., 1995), possibly due to the difference in the bone morphologies of the patients. Some found interfacial shear strength to be dependent on the depth of cement penetration (MacDonald et al., 1993; Majkowski et al., 1993). This is consistent with the results obtained in the present study (partial correlation of $r^2=0.76$), although there seems to be no consensus on the issue (Krause et al., 1982; MacDonald et al., 1993; Majkowski et al., 1993; Mann et al., 2001).

The amount of bone interdigitated with cement was used to rationalise the variation in the strength of the interface, an improvement on cement penetration as a correlation parameter as reported previously ($r^2=0.025$, (Mann et al., 2001)); $r^2=0.22$, (Krause et al., 1982)), although the correlations are still not strong ($r^2=0.5-0.7$ (Mann et al., 1997, 1999, 2001)). It seems that neither the depth of the cement penetration, nor the amount of bone interdigitation, is solely responsible for the interfacial strength attained. Further work is necessary to identify the most relevant parameter(s) in defining the interfacial strength.

The morphologies of the failed samples are presented in Fig. 4.5, where the cement half of the samples was μ CT imaged. The effect of loading angle on the failure mode may influence the quantity of the remaining bone in the cement, which seems to increase with the increase of the loading angle, suggesting more effective “pull-out” of cement pedicles in tension than

in shear. In the studies of Mann et al., the highest values in the measured failure strength were obtained in Type IV mode, as found in the present study, i.e. failure occurred at the extreme of cement penetration with cement pedicles remaining in the bone, a case which rarely (<5%) occurred under mixed mode loading in (Mann et al., 2001) and none at all under tensile or shear in (Mann et al., 1997, 1999). In the current study, however, majority of the cases failed in either Type I or II at large loading angles, as opposed to Type IV or II for samples at low loading angles.

Admittedly, the experimental work was carried out using bovine bones in air, conditions considerably different from those in vivo, where the interface would not only experience more complex loading conditions but also be exposed to bone remodelling in response to the introduction of implants and possible degradation of the bone cement. Although bovine bones have been used in the studies of bone porosity and its influence on the measured interfacial fracture toughness (Graham et al, 2003), bovine trabecular bones is known to be slightly denser than human trabecular bones and with more of a plate-like structure as opposed to a rod-like structure found in human trabeculae. These architectural differences were not considered in this work, although they might also affect the interfacial behaviour.

4.4. Micromechanical compressive behaviour of bone-cement interface

4.4.1. Image-guided failure assessment (IGFA)

Step-wise compression testing of bone cement samples (n=5) in combination with time-lapsed μ CT imaging was performed using the novel loading stage (LS) device previously described. Modular end platens were used to apply compressive loads. The sample was first

glued (Permabond, UK) onto the lower compressive platen, whilst a small preload was applied through the top platen connected to the actuator ensuring a good end contact.

The specimens underwent μ CT analysis prior to testing and a complete data acquisition was performed. The CT scanner was set to a voltage of 55kV and a current of 145 μ A. With an isotropic voxel size of 20 μ m the image acquisition process was performed at a rotational step of 0.19° over 360° for 90min approximately and 3D reconstruction of the samples was obtained.

The samples were step-wise compressed at two selected displacements, corresponding approximately to the ultimate apparent strength and before the final failure, respectively. The former was defined as the first point on the load-elongation curve where the slope was zero (Ciarelli et al., 2000); whilst the latter when the second zero slope was reached. The slope readings were real-time captured together with the force-extension curve. At each displacement step the specimen was allowed to relax for about 15mins before CT imaging and data acquisition procedure were carried out. The 15-min window was deemed sufficient for the specimen to reach a stable state of stress relaxation, based on trial tests performed. All tests were carried out under displacement control at a constant cross head speed of 0.01mm/sec.

4.4.2. Cement penetration and bone morphology

The μ CT data sets obtained from the five test samples were used to determine the contact area (CA), cement penetration depth (CP), cancellous bone volume fraction (BV/TV); and to further assess the step-wise progressive microdamage evolution. The CT scan data of the specimens in the undeformed stage were imported into Mimics 14.01, where 3D objects

were created by thresholding the bone and the cement based on a protocol by Ding et al. (1999). The 3D cement object was then dilated by two voxels (40 μ m) and the contact volume was determined by a Boolean intersection between the cement and the bone. This volume was then divided by the dilatation thickness, resulting in an estimated contact area (Mann et al., 2008).

An average cement penetration over the specimen volume was calculated by measuring the cement penetration depth at a regular interval (0.5mm) from a reference line AB on a frontal cross section (Maher & McCormack, 1999), then repeating the measurement across the thickness of the specimen at the same interval (0.5mm) on a series of frontal cross sections to obtain a volume average, as shown in Fig. 4.6.

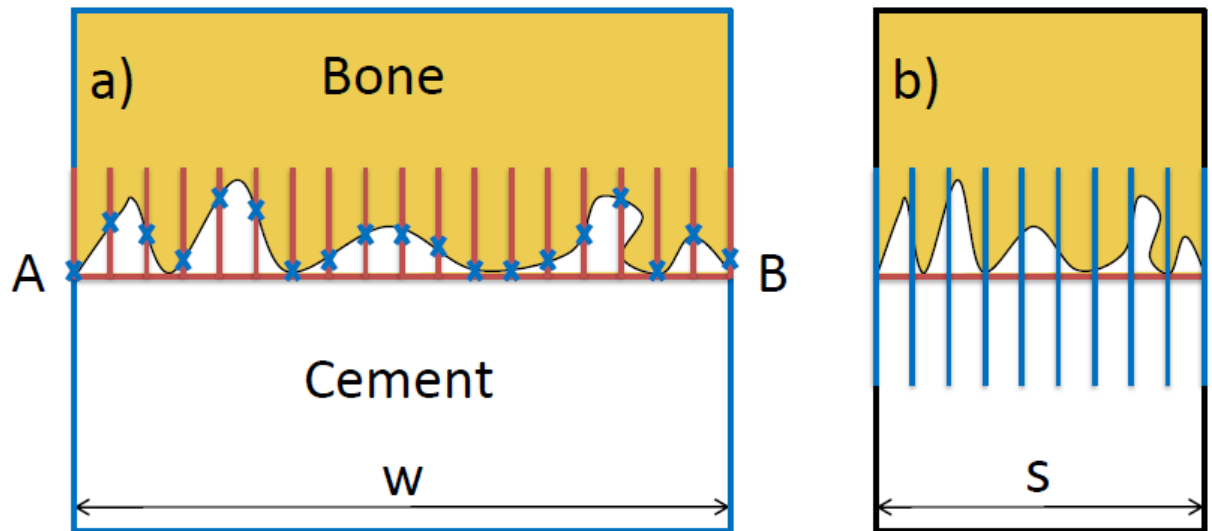


Figure 4. 6: The average cement penetration over the volume was estimated by (a) measuring firstly the penetration depths (marked “x”) at a regular interval (0.5mm) from a reference line AB on a frontal cross section (Maher and McCormack, 1999), where W is the width of the sample; (b) repeating the measurement across the thickness (S) of the specimen at the same interval (0.5mm) on all the frontal cross sections. A volume average was then obtained from all the measurements (Tozzi et al., 2012b).

The calculation of BV/TV for the cancellous bone in the bone-cement samples was carried out following a procedure reported by Guillen et al. (2011) and explained in detail in Chapter 5. A volume of interest (VOI, 6mm×8mm×5mm) was chosen to contain the interdigitated bone and part of the adjacent bone for all samples, over which BV/TV was calculated. The size of the VOI satisfies the continuum requirements of BV/TV calculation (Harrigan et al., 1988).

The microdamage at the bone-cement interface was recorded and evaluated at each step during the step-wise compression for all samples.

4.4.3. Results

The microdamage evolution in all five bone-cement interface specimens (BC01-BC05) showed similar patterns where the predominant deformation was found to initiate in the bone region, whereas the cement region appeared to be unaffected even towards the final failure stage. The main load transfer appears to occur in the bone-cement contact region, resulting in progressive microfailures including bending and buckling of trabeculae adjacent to the interdigitated region. Fig. 4.7 illustrates some details of the progressive microdamage during the step-wise compression in representative 3D sub-volumes.

No such deformation was found in the bulk interdigitated region; and there is no evidence of bone-cement interfacial debonding or cement cracking from the CT images.

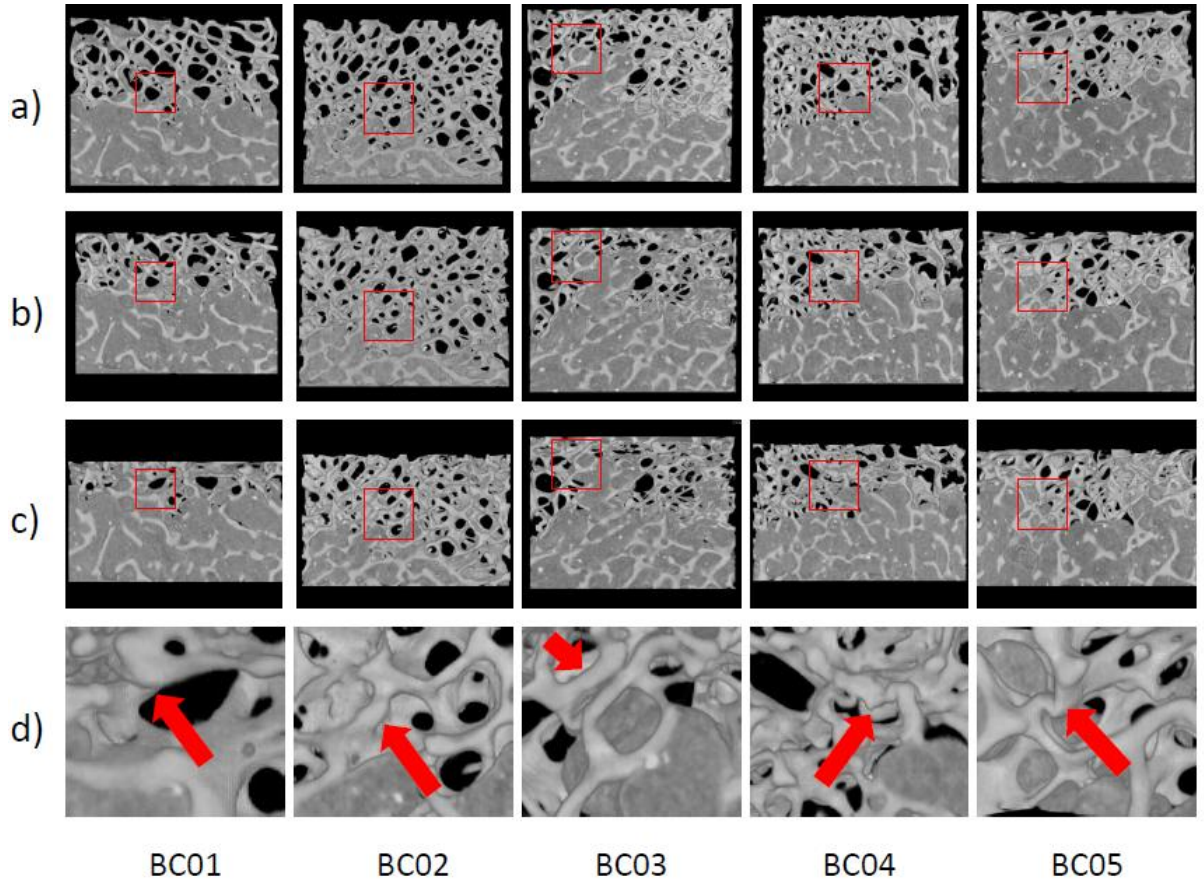


Figure 4. 7: 3D reconstructions of the five bone-cement interface samples tested (BC01-BC05) and selected sub-volumes (rectangles), where the progressive microdamage was monitored: (a) Unloaded; (b) at the ultimate stress; (c) failure state and (d) details of the local damage (mostly trabeculae buckling) in the bone-cement contact region, as indicated by the arrows (Tozzi et al., 2012b).

The contact area (CA) and cement penetration depth (CP) for the five samples were found to be $225.95(\pm 97.55)\text{mm}^2$ and $3.60(\pm 1.50)\text{mm}$, respectively. Although a strong correlation was found between CA and CP ($r^2=0.94$), there is no apparent correlation of CP (and consequently CA) with the ultimate compressive strength ($r^2=0.18$), as shown in Figs. 4.8a and d.

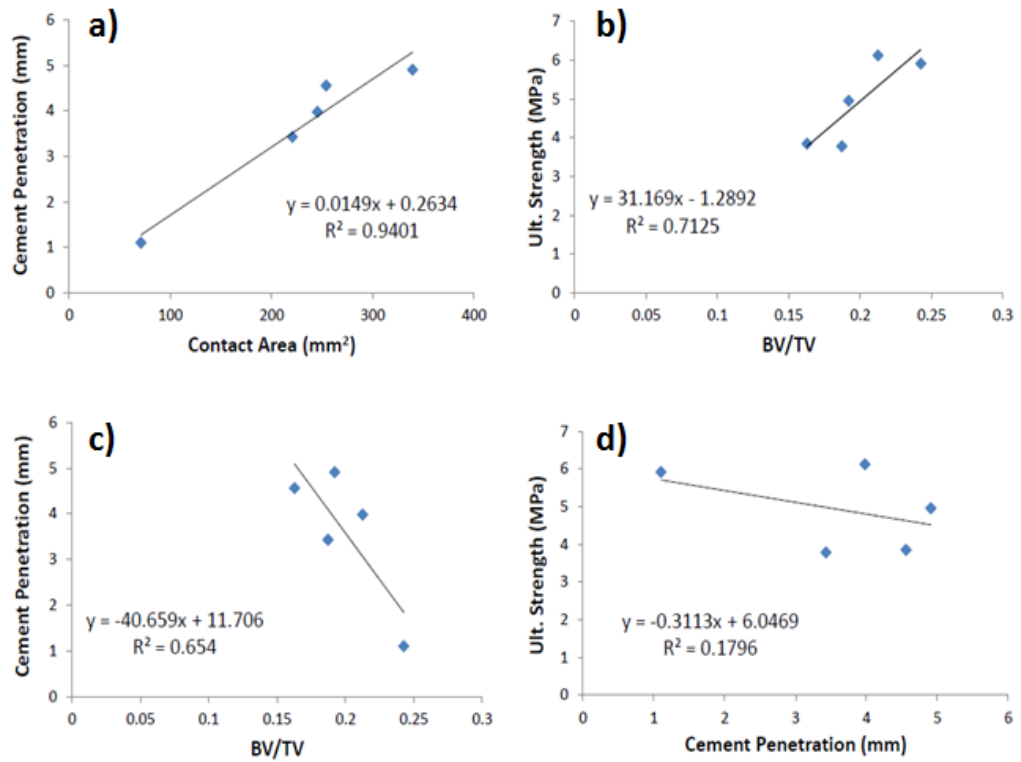


Figure 4. 8: Correlations between (a) cement penetration and bone-cement contact area; (b) interfacial strength and cancellous bone volume fraction (BV/TV); (c) cement penetration and BV/TV and (d) interfacial strength and cement penetration. The correlations range from very good ($r^2 = 0.94$ for (a)) to poor ($r^2 = 0.18$ for (d)) (Tozzi et al., 2012b).

The BV/TVs were calculated over the selected VOI for all the samples and the mean and standard deviation were found to be $0.20(\pm 0.03)$. A moderate correlation (Fig. 4.8b) was found between the BV/TV and the ultimate strength ($r^2=0.71$); whilst the correlation of BV/TV with CP is $r^2=0.65$ (Figs. 4.8c). The ultimate apparent compressive strength for the five samples was estimated at $4.93(\pm 1.10)$ MPa.

The stress-displacement responses of the five samples tested are shown in Fig. 4.9.

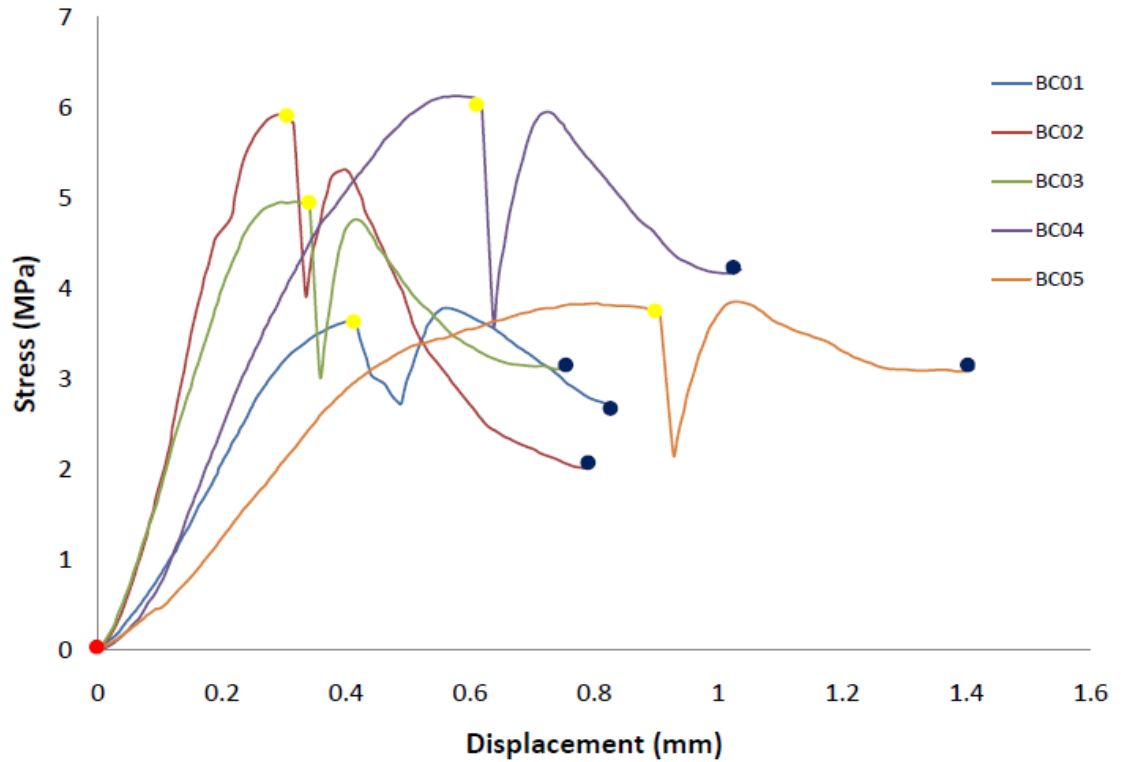


Figure 4. 9: The stress-displacement curves obtained from the compressive testing of the five samples. The dots (red=unloaded, yellow=ultimate strength, dark blue=failure) on the experimental curves indicate the positions at which the samples were imaged under a fixed displacement. The discontinuities are due to the stress relaxation in the samples under a constant displacement (Tozzi et al., 2012b).

4.4.4. Discussion

Three dimensional full volume visualisation of damage evolution in bone-cement interface during static compression shows that the bone region sustained most of the deformation overall. As the load was transferred to the cement mantle, damage in the bone-cement contact region developed, resulting in progressive failures of the trabeculae mainly due to buckling (Fig. 4.7). These findings were confirmed with the FE simulations (Figs. 4.10 and 4.11) reported by Tozzi et al. (2012b), and are consistent with the finding of Mann et al. (2008) that the contact interface was responsible for most micromotions of the bone-cement interface.

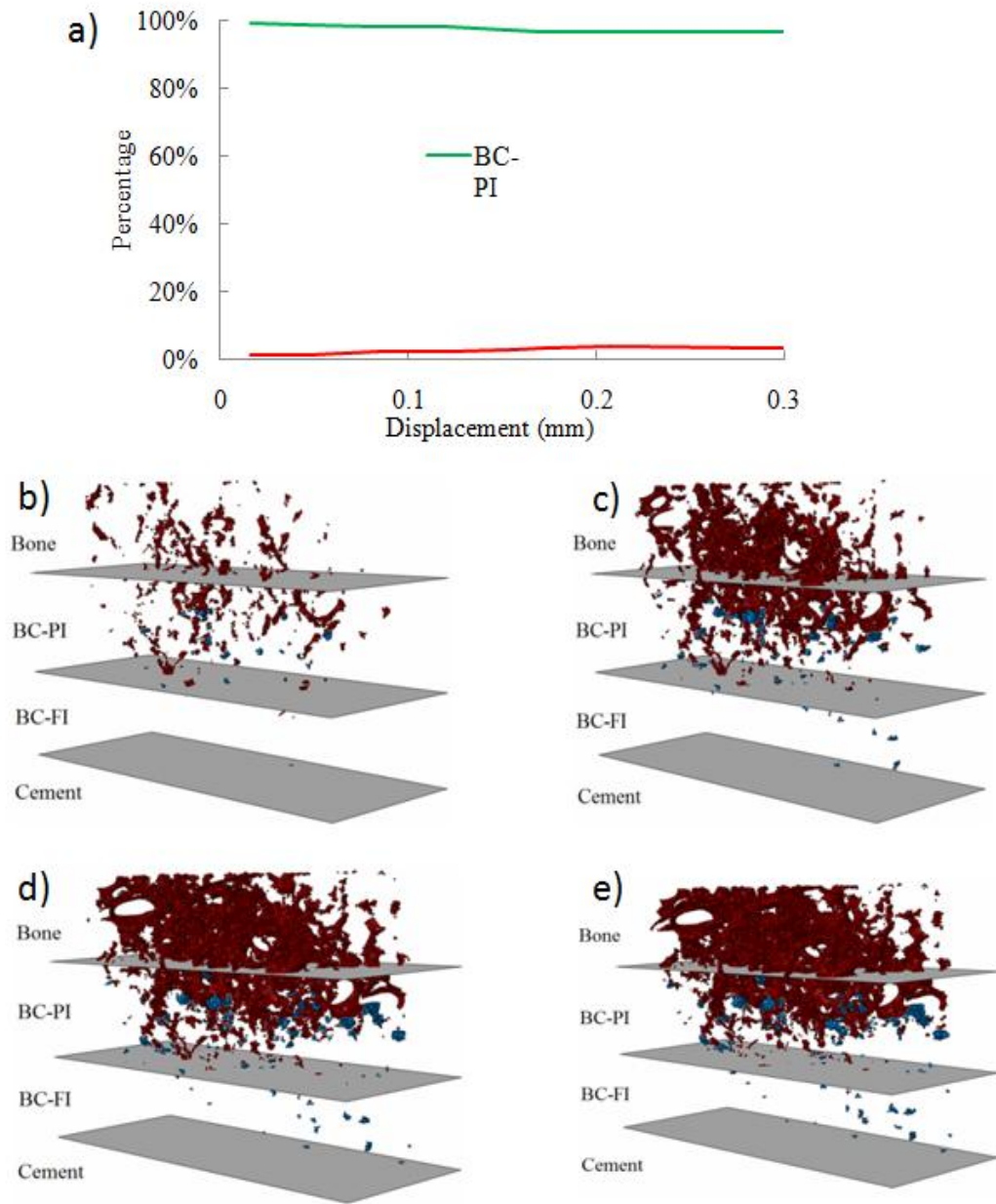


Figure 4. 10: a) The percentage of energy dissipated from the bone in BC-PI (Bone-Cement Partially Interdigitated) and BC-FI (Bone-Cement Fully Interdigitated) regions. b)-e) The predicted progressive development of yielded elements from the FE model under a compressive displacement of: b) 0.05mm; c) 0.1mm; d) 0.2mm and e) 0.3mm. Yielding appears to initiate mainly in the Bone and partially interdigitated (BC-PI) regions, developing across the fully interdigitated (BC-FI) region. Red: Bone; blue: Cement (Tozzi et al., 2012b).

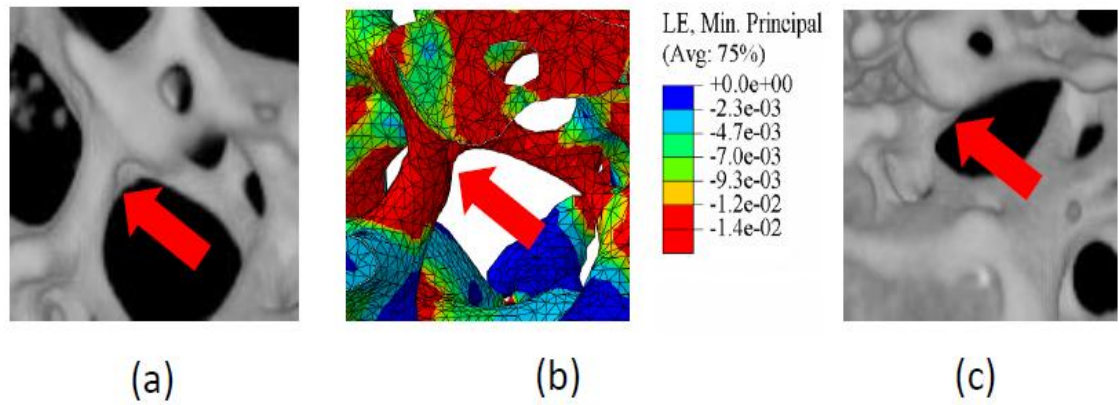


Figure 4. 11: (a) CT image of a typical bone-cement interface sub-volume (Fig. 4.7a, BC01). (b) The local predicted minimum principal strain distribution of the same sub-volume from the FE model and (c) the damage sustained at the failure state during the experiment (Fig. 4.7c, BC01). The high local compressive strain predicted (b) may explain the local buckling damage (c), as indicated by the arrows (Tozzi et al., 2012b).

The effect of cement penetration on the bone-cement interfacial behaviour has been a major interest (Krause et al., 1982; Mann et al., 1997, 2001; Leung, 2008; Waanders et al., 2010). Interestingly, although the CA correlates with CP, consistently with the report of Waanders et al. (2010), neither CA nor CP correlates with the apparent interfacial compressive strength measured in the current study. The pressurisation technique used for the specimen preparation resulted in an overall CP depth of $3.60(\pm 1.50)$ mm, a value within the optimal range of 3–5mm considered for bone-cement interface (Krause et al., 1982; MacDonald et al., 1993). As opposed to in the cases of tensile and shear (Mann et al., 1997, 2001, 2008), where moderate correlations between the interfacial strength and cement penetration were found, the current study shows virtually no correlation between the compressive strength and cement penetration depth. This is likely due to the difference in the loading mode which dictates the failure mechanism. In tensile and shear cases, cement pedicles being “pulled-out” serves as the main failure mechanism, where the role of cement penetration depth may

be important; whilst this mechanism is absent from compression as compressive failures are attributed mainly to trabeculae bulking in the bone and the bone-cement contact regions. In addition, insufficient load transfer to the bulk interdigitated volume also supports this line of argument. Furthermore, the bone used in Mann et al was from the medullary cavity of femurs with substantial cortical bone and some trabecular bone, whilst only trabecular bone taken from iliac crest was used in the present study. This may also explain some of the differences in our results and those of Mann et al. (1997, 2001, 2008).

The ultimate strength of cancellous bone is known to be strongly correlated with the apparent density (or BV/TV) of the bone (Keaveny et al., 2001). The average BV/TV from this study is $0.20(\pm 0.03)$ and it has a correlation ($r^2=0.71$) with the ultimate interfacial strength. This may be explained by the role of the bone and the cement in load sharing as constituents of the bone-cement composite structure, although the bone appears to carry most of the loads (Figs. 4.7 & 4.10).

The current study describes the essential response of bone-cement interface under static compression, a common loading mode experienced in tibial, femoral components of knees and acetabular components, and reflective of the immediate post-operational state of cement fixation, where cement cracking due to fatigue loading and bone cracking/remodeling are absent.

4.5. Conclusions

The results reported in this chapter may be useful in the analysis of bone-cement interface in joint replacements with and without defects under different loading conditions.

The measured interfacial strength does not appear to correlate with the depth of cement penetration, and the effect of the loading angle seems more consistent in dictating the failure of the bone-cement interface under mixed mode loading conditions, as confirmed by both mechanical testing and post failure morphologies.

Time-lapsed μ CT imaging and in situ step-wise compression showed that load transfer in bone-cement interface occurred mainly in the bone cement contact region, resulting in progressively developed deformation due to trabeculae bending and buckling. Contact area and cement penetration depth were found to be closely related, although neither correlates with the measured compressive interfacial strength. A moderate correlation was found between the BV/TV and the ultimate interfacial strength.

Chapter 5

Characterisation of cancellous bone analogous materials for biomechanical testing

5.1. Introduction

The main results discussed in this chapter were partially obtained at the University of Siegen (Germany) within a research partnership (British Council-DAAD, ARC1354) with the Universities of Portsmouth (UK), and reported in the following papers:

- Guillen, T., Zhang, Q.-H., Tozzi, G., Ohrndorf, A., Christ, H.-J., Tong, J., 2011. Compressive behaviour of bovine cancellous bone and bone analogous materials, microCT characterization and FE prediction. *J. Mech. Behav. Biomed. Mater.* 4, 1452-61.
- Guillen, T., Ohrndorf, A., Tozzi, G., Tong, J., Christ, H.-J., 2011b. Compressive fatigue behaviour of bovine cancellous bone and bone analogous materials under multi-step loading conditions. *Adv. Eng. Mat.*, doi: 10.1002/adem.201180060.

In this chapter, AlSi7Mg and CuSn12Ni2 open-cell foams were characterised using Scanning Electron Microscopy (SEM) and μ CT. Static compression testing together with multiple-step cyclic tests were carried out to determine whether the foams might be suitable as cancellous bone mechanical models for biomechanical studies. Mechanical and morphometrical properties were obtained for the three foams and compared with the corresponding values obtained from bovine iliac crest cancellous bone.

Some preliminary results, under mixed-mode loading conditions, of foam-cement interface mechanical testing of the interface are also presented.

5.2. Materials and specimen

The open-cell metal foams used in this work were precision-cast, where an open-cell polymer model was used as a template to create casting moulds into which metallic alloys were cast. The desired cell size, relative density and geometry were determined by the appropriate polymer foams. The polymer foams were filled with a mould casting slurry which was then baked to harden the casting material (Fig. 5.1a).

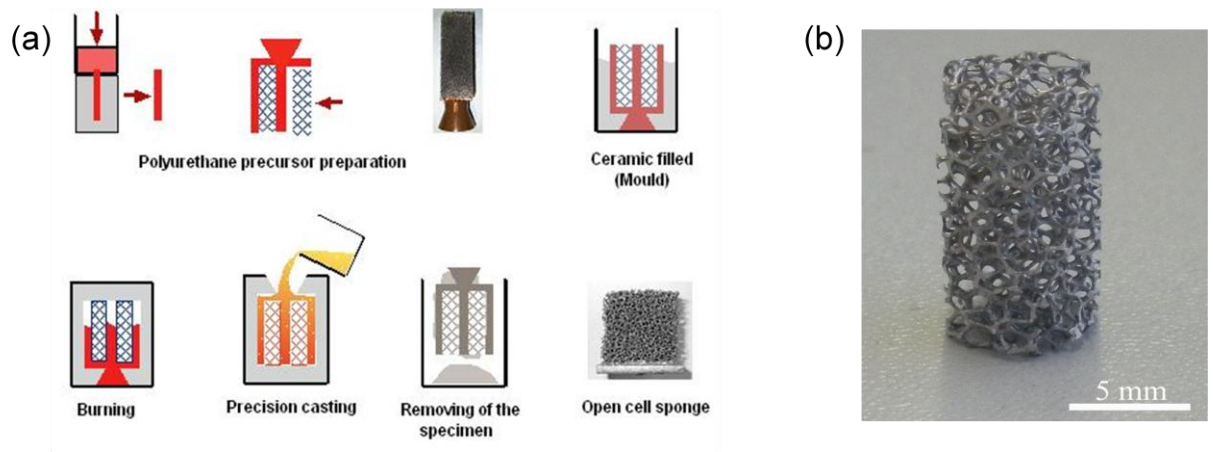


Figure 5. 1: (a) The open-cell production process and (b) the cylindrical foam specimen used in this study.

The mould materials were then removed to leave the metal foam with the prescribed morphology of the original polymer foam (Girlich & Franzke, 2001). Open-cell AlSi7Mg foams (m-pore GmbH) with porosities of 30ppi (pores per inch) and 45ppi were produced, together with a 30ppi CuSn12Ni2 foam. The foams were obtained in sheets with dimensions of 200x200x10mm³. Specimens from each type were machined to a cylindrical shape with a diameter of 8mm (Fig. 5.1b). Cancellous bone samples of the same diameter were harvested from fresh bovine iliac crest, using a milling drill. The iliac crest was chosen to obtain the bone samples with no preferential orientation in its trabecular network. The bone samples

were kept frozen till testing (Linde & Sorensen, 1993); and were thawed and stored in a SBF solution (Kokubo & Takadama, 2006) for approximately 24h before testing.

The three foam materials were also used to interdigitate with acrylic bone cement to create foam-cement interface samples. Samples were prepared with the same procedure used for the bone-cement interface coupons (finger packing) as reported in Chapter 4.

5.3. Compressive behaviour of cancellous bone and open-cell metallic foams

5.3.1. Material characterisation

Ten cylindrical specimens for each of the three foams and bone were examined prior to mechanical testing using μ CT (XT H 225 X-Tek, Ltd). The scanner settings used for the three different materials were 55kV-130 μ A for the trabecular bone, 60kV-140 μ A for the AlSi7Mg foams and 90kV-65 μ A for the CuSn12Ni2 foam, respectively. With an isotropic voxel size of 15 μ m the image acquisition process was performed at a rotational step of 0.19° over 360° and 3D reconstruction of the samples was obtained.

The calculation of the morphometric parameters was carried out by importing the CT images into MicroView v2.1.2 software (GE Healthcare) and using a uniform threshold procedure (Ding et al., 1999). A circular region of interest (ROI) containing only cancellous bone/foams (diameter 7.2mm) was defined in order to exclude the surface debris as a result of the cutting operation. The cylindrical volume of interest (VOI) for each sample was of approximately 7.2mm diameter by 13mm (height). For each specimen the following morphometric parameters were determined over the entire VOI, including volume fraction,

BV/TV or FV/TV, trabecular thickness (Tb.Th) or strut diameter and trabecular separation (Tb.Sp) or cell spacing. The reconstructed bone and foams are shown in Fig. 5.2.

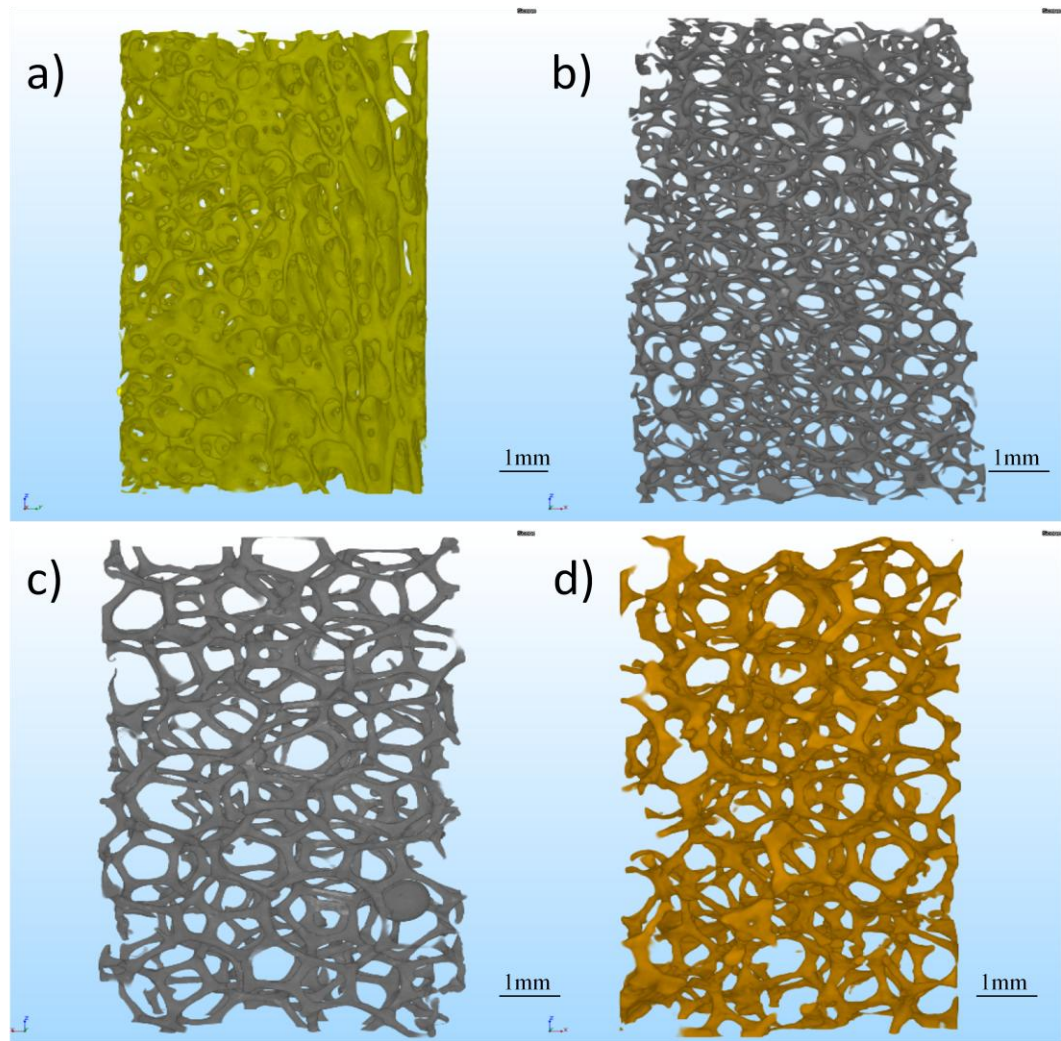


Figure 5. 2: 3D μ CT reconstructions of (a) bovine trabecular bone; (b) 45ppi AlSi7Mg foam; (c) 30ppi AlSi7Mg foam and (d) 30ppi CuSn12Ni2 foam (Guillen et al., 2011).

In addition, the bone specimens were cleaned with ethanol in an ultrasonic system for approx 3h, then dehydrated in a desiccator for approximately 24h and finally sputtered with gold to allow the examination using the SEM (Philips XL 30). SEM was also carried out for

AlSi7Mg and CuSn12Ni2 foams. The morphometric parameters were then manually estimated based on the micrographs.

5.3.2. Mechanical testing

Compression testing was carried out on a MTS 810 servo-hydraulic testing system under displacement control. The measured mechanical properties were reported to be influenced by the experimental procedures (Keaveny et al., 1997) and specimen geometry (Linde et al., 1992; Keaveny et al., 1993). Hence efforts were made to minimise the test variables. Custom made end-caps were manufactured from hardened steel. The ends of each cylindrical specimen were then embedded about 3mm into the end-caps with acrylic resin to minimise the end artifacts (Keaveny et al., 1997). A custom made aligning system was also used to ensure the alignment of the sample with the loading axis. The length of the cylinders after fixation was of approximately 13mm, thus achieving an aspect ratio of 1.625:1, within the range of 1.5 to 2, as recommended by DIN 50134 (2008) for compressive testing of metallic cellular materials; and not far from the 2:1 suggested by Keaveny et al. (1993) for testing of bones. Strain measurements were performed using a 15mm gauge-length extensometer (Sandner EXA R15-5X) attached to the end-caps (Keaveny et al., 1997). The elastic modulus, yield stress and ultimate strength were determined. The experimental set-up used for the mechanical testing is shown in Fig. 5.3.

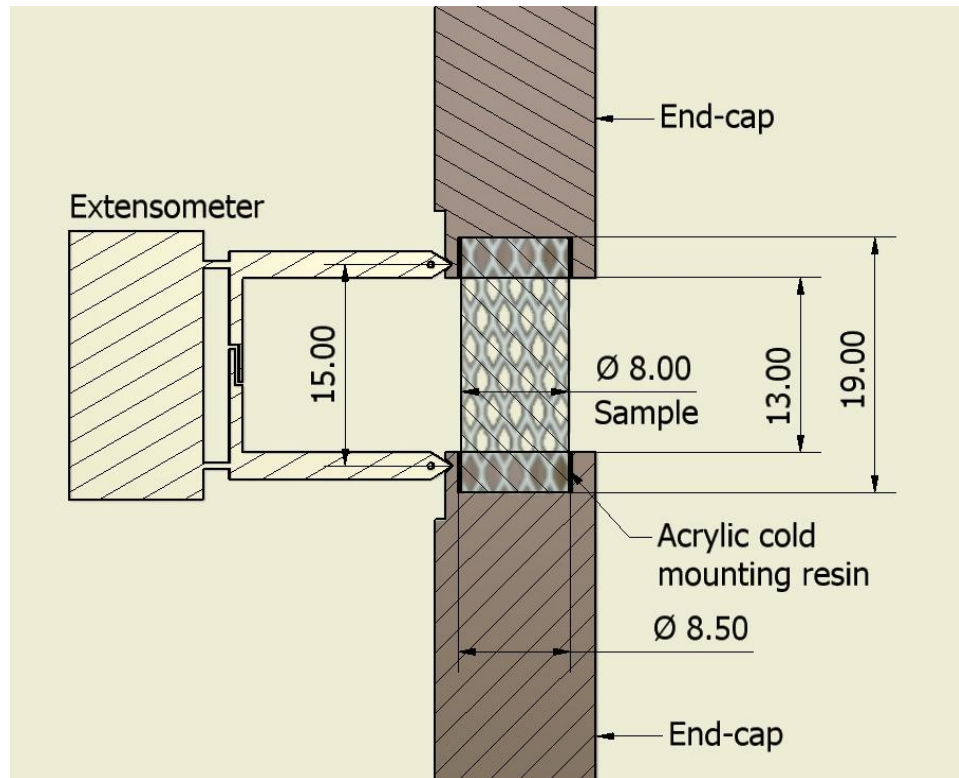


Figure 5. 3: Schematic of the experimental setup. The sample was embedded into the endcaps using acrylic resin. Strains were measured using an extensometer attached to the end-caps (Guillen et al., 2011).

Tests were carried out following the protocol for cellular metals (DIN 50134, 2008). Preliminary compression tests were performed at a constant strain rate of $0.01/\text{sec}$ on the three foams ($n=5$ for each type) and the bone samples ($n=5$) to estimate the mechanical properties, including R_{20} , R_{70} (20% and 70% of the compressive strength, respectively) and ultimate compressive strength (σ_{ult}). The tests ($n=10$ for each material) were then carried out at an initial strain rate of $0.001/\text{sec}$ under a preload of approximately 2% of the σ_{ult} until R_{70} was reached. The load was then reduced to R_{20} at a strain rate of $1/\text{sec}$. Finally the samples were re-loaded from R_{20} until failure at a strain rate of $0.1/\text{sec}$. The ultimate compressive strength was defined as in Ciarelli et al. (2000) and Li & Aspden (1997). The protocol was slightly modified for the testing of bones, where R_{40} was used instead of R_{70} .

The elastic modulus was measured directly from the unloading linear stress-strain relationship between R_{70} (R_{40} for the bones) and R_{20} . The yield stress was estimated using the 0.2% offset method (Keaveny et al., 1994).

5.3.3. Results

A summary of the morphological and mechanical properties of bone and foams are presented in Table 5.1 and 5.2 respectively. The elastic modulus of AlSi7Mg foams is about 30% (30ppi) and 40% (45ppi) that of bone; whilst the elastic modulus of CuSn12Ni2 foam is about 30% higher than that of bone. Further, the yield stresses of all foams are much lower than that of the bone (AlSi7Mg foams are only 8-10%; whilst CuSn12Ni2 foam is about 50% of the yield stress of the bone). The ultimate compressive strengths follow a similar trend (11-13% for AlSi7Mg, 50% for CuSn12Ni2 of the ultimate compressive strength of the bone).

Table 5. 1: A summary of the morphometry results from all the tests performed on the three foams and bone (Guillen et al., 2011).

Morphometry	Bone		45ppi AlSi7Mg		30ppi AlSi7Mg		30ppi CuSn12Ni2	
	SEM	CT	SEM	CT	SEM	CT	SEM	CT
BV/TV FV/TV	NA	0.25 (± 0.07)	NA	0.16 (± 0.008)	NA	0.06 (± 0.005)	NA	0.12 (± 0.007)
Cell spacing Tb.Sp (mm)	0.807 (± 0.213)	0.613 (± 0.282)	1.783 (± 0.448)	0.82 (± 0.035)	2.08 (± 0.45)	2.034 (± 0.069)	2.075 (± 0.370)	1.432 (± 0.022)
Strut thickness Tb.Th (mm)	0.142 (± 0.035)	0.177 (± 0.027)	0.142 (± 0.02)	0.156 (± 0.005)	0.196 (± 0.02)	0.142 (± 0.005)	0.291 (± 0.081)	0.198 (± 0.002)

Table 5. 2: The mechanical properties obtained from testing of the three foams and bone (Guillen et al., 2011).

Mechanical Properties	Bone	45ppi AlSi7Mg	30ppi AlSi7Mg	30ppiCuSn12Ni2
σ_{ult} (MPa)	9.36 (± 1.45)	1.19 (± 0.09)	1.05 (± 0.2)	4.54 (± 0.31)
σ_y (MPa)	8.02 (± 1.19)	0.84 (± 0.09)	0.67 (± 0.16)	3.75 (± 0.52)
E (MPa)	800 (± 138)	314 (± 63)	244 (± 63)	1068 (± 206)

The stress-strain curves of the four sets of materials are presented in Fig. 5.4. The mechanical responses of the two AlSi7Mg foams (Figs. 5.4b-c) seem to be similar, with a significant range of stress hardening (up to 3% strain) following the initial elastic response, then “softening” till about 10% strain.

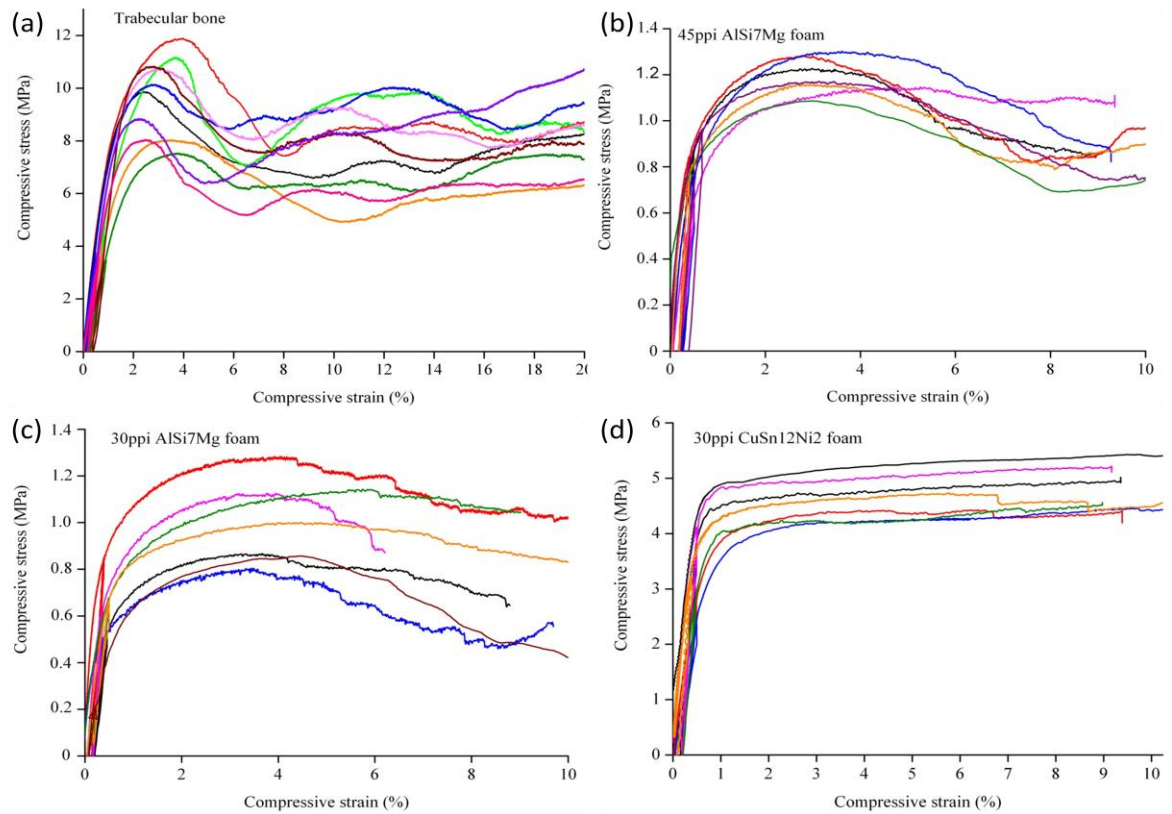


Figure 5. 4: Mechanical responses as measured of (a) bovine trabecular bone; (b) 45ppi AlSi7Mg; (c) 30ppi AlSi7Mg and (d) 30ppi CuSn12Ni2 foam (Guillen et al., 2011).

CuSn12Ni2 foams (Fig. 5.4d) showed only limited hardening following the elastic response, whilst no softening was observed in the same strain range (up to 10%) as in the AlSi7Mg foams. The stress-strain behaviour of the bone is shown in Fig. 5.4 (a), where linear behaviour was observed at the beginning of the compression. This was followed by a brief but significant hardening period, then by a prolonged softening with considerable reduction of stresses due to progressive failure of the trabeculae. For large strains between 9-12%, a slight increase in the stress may be observed also, due to the densification process of the trabeculae.

The SEM results were found to be broadly similar to those obtained from the μ CT (Table 5.1).

The microstructures of the four samples from the SEM analysis are shown in Fig. 5.5.

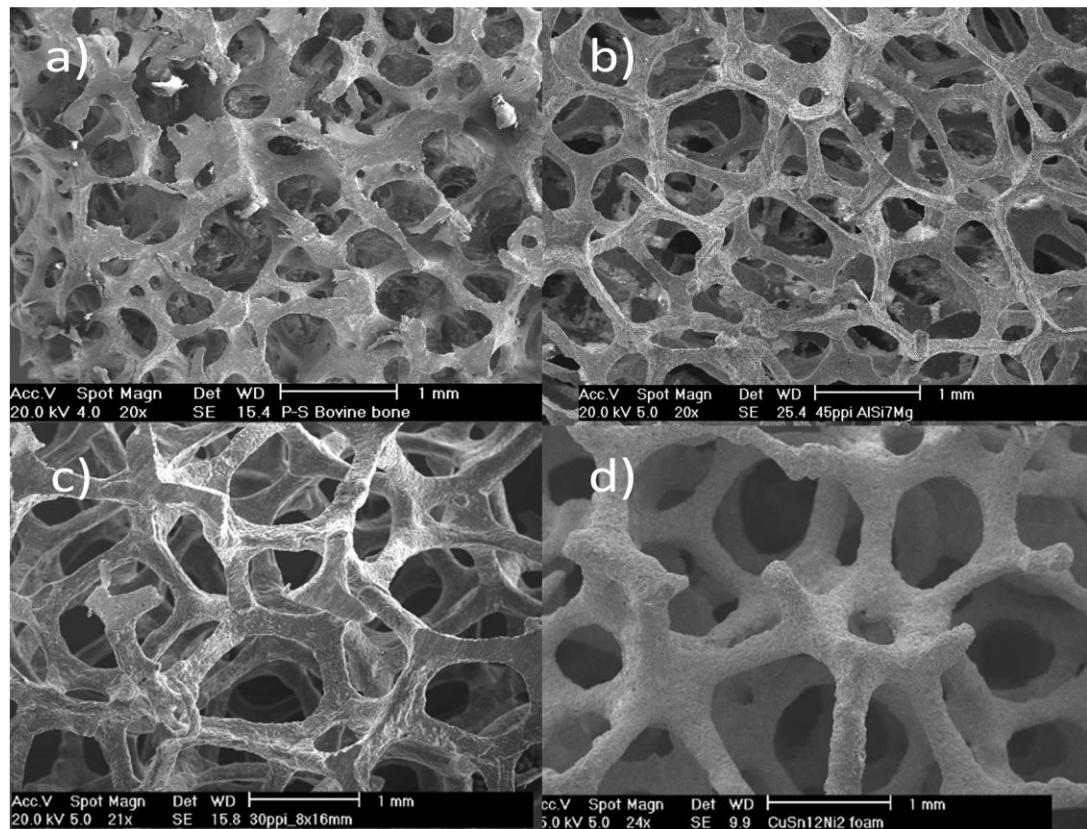


Figure 5. 5: SEM micrographs of (a) bovine trabecular bone; (b) 45ppi AlSi7Mg; (c) 30ppi AlSi7Mg and (d) 30ppi CuSn12Ni2 foam (Guillen et al., 2011).

5.3.4. Discussion

The bone and foams tested in this study exhibited the usual deformation behaviour observed in cellular materials (Gibson & Ashby, 1997; Gibson, 2005). The adoption of a compression test standard for cellular materials (DIN 50134, 2008) and an appropriate gripping arrangement (Fig. 5.3) have reduced the testing variables, hence more consistent results in the measured elastic modulus, yield stress and ultimate strength are obtained. Particular attention was placed on the determination of the elastic modulus. At the initial loading stage of the test, an increase in stress was observed, which, at first sight, appeared to be elastic. However, previous works revealed that the stress was only partially reversible, and some irreversible deformation processes in the cellular structure occurred during this loading stage (Banhart & Baumeister, 1998). Therefore, the measurement of the elastic modulus was carried out based on the unloading linear stress-strain relationship between R_{70} and R_{20} .

The “softening” or reduction of strength observed in the AlSi7Mg foams may be characteristics of this type of alloys, which are influenced by the manufacturing process. It was previously observed (Krupp et al., 2006) that in cast AlSi7Mg foams coarse Si and Mg₂Si precipitates were present along the grain boundaries. These precipitates can drastically reduce the effective properties of cell struts and trigger brittle failure. Another factor that may affect the deformation behaviour is the homogeneity of the foam. Poor homogeneity introduces localised deformation by the formation of deformation bands leading to cell collapsing. With increasing homogeneity the deformation of the foam structure becomes more uniform due to the more uniform distribution of the stresses. The results of current compression tests show that the ductility of the AlSi7Mg foam studied in this work is superior to that used in a previous study (Ohrndorf et al., 2006). A possible reason for this

improvement may be the development of advanced casting techniques, which resulted in not only a more homogeneous foam structure but also better microstructural properties of the cell struts. Most of the closed-cell metal foams commercially available show some anisotropy of their mechanical properties as a consequence of cell asymmetry obtained during the manufacturing processes. For many applications this anisotropy is considered as a disadvantage and much work is carried out in order to reduce anisotropy and irregularity of the foam structure. On the other hand, precision casting process might be used to produce “tailored” anisotropy by suitably defining the precursor material. Another possibility to produce foam materials with a specific degree of anisotropy is to use an isotropic precursor material and stretch it prior to embedding with a ceramic mould material. The stretched shape of the cells will give the metal foam an anisotropy that could be controlled by the degree of stretching.

In comparison with the AlSi7Mg foams, failure of cell struts in CuSn12Ni2 foam, if any, seems to be insignificant from the deformation behaviour of the foam. Consequently, an approximately linear-perfect plastic stress-strain curve was obtained. Similar behaviour was found in a ductile foam Cu11Zn3.5Si (Ohrndorf et al., 2005), where elastic deformation behaviour during the loading stage of a compression test (up to 10% strain) was followed by a pronounced plateau regime up to 70% strain. The ductility of the cell struts, as well as the homogeneity of the architecture of the foam, may be responsible for the deformation behaviour of the foam. The high mechanical strength achieved in this bronze foam, as opposed to those of the two aluminum foams, may be attributed to the bulk material properties.

Detailed studies of cancellous bone at tissue level (van Rietbergen et al., 1995) showed that the predominant deformation mode in bones is bending, where the trabeculae can be strained in tension even under apparent compressive loading. Unlike foams, where a general trend of buckling may be observed, brittle fracture due to the presence of minerals was observed to be more evident in bones (Nazarian & Muller, 2004), although collagen network within the trabeculae may keep the mineral phase together even at large strains. These findings are consistent with some IGFA results for the three foams and cancellous bone, where the final failure was characterised by predominant buckling in the foam material and a mixture of bending, buckling and brittle failure in cancellous bone (Guillen et al., 2011c). It needs to be said that the bone samples were selected from iliac crest where no pronounced anisotropy exists in the trabecular network, and this permits a direct comparison with the foams which are essentially isotropic.

There are significantly higher scatters in the data obtained from the bone ($\pm 2.5\text{MPa}$), than those from AlSi7Mg foams ($\pm 0.25\text{MPa}$ for 30ppi; $\pm 0.1\text{MPa}$ for 45ppi) and that of CuSn12Ni2 foam ($\pm 0.5\text{MPa}$). This is not unexpected as mechanical properties of trabecular bone are affected by density, structural anisotropy, marrow influence and viscoelastic properties (Nicholson et al., 1997; Carter & Hayes, 1977; Martin et al. 1998), factors that can vary greatly with site and subject. From a biomechanical characterization point of view, substitution of bones with appropriate foams may be a reasonable step forward to remove some of the uncertainties.

Although the morphological parameters of the foams and the bone used here are not too dissimilar (Table 5.1, Figure 5.2), the apparent mechanical responses differ considerably between the foams and the bone (Table 5.2). These differences may be explained by the

differences in the mechanical properties of the base materials as well as their microarchitecture details, although it remains a challenge to find foams that represent both the apparent mechanical properties of trabecular bone and its microarchitecture. However, FE simulations performed on the three materials (Guillen et al., 2011) showed that, regardless of the difference in their architectures, the yielded elements appear to be distributed throughout the whole structure for both bone and foam. This is interesting as although the global mechanical responses differ considerably between the foams and the bone, the damage distribution and evolution with load are broadly similar. This finding might be useful in the use of foams as bone substitute materials for material testing purposes using scaled models, for example.

Gibson (2005) described a general relationship between the apparent mechanical properties of cancellous bone and its density; and indicated that low-density trabecular bone may resemble open-cell foams. The metallic foams used in the present study are in the low-density range of 177-355kg/m³ for cancellous bones. The 30ppi and 45ppi AlSi7Mg foams might be related to bones with average densities of 177kg/m³ and 193kg/m³, respectively; whereas a higher value of 355kg/m³ might be represented by the 30ppi CuSn12Ni2 foam.

The main goal of this study was necessarily limited to the evaluation of the morphological and mechanical properties of some of the available metallic foams as possible candidates for cancellous bone substitute materials for mechanical testing purposes. Caution must be exercised in cases such as assessing pull-out strength of screws though, where local microstructural properties may dictate the outcome.

5.4. Compressive fatigue behaviour of cancellous bone and bone analogous materials under multi-step loading conditions

5.4.1. Mechanical testing and SEM analysis

The monotonic compression results previously obtained (see Paragraph 5.3) were used to set up the cyclic tests. Fatigue tests on a MTS 810 servo-hydraulic testing system equipped with a 2kN load cell were performed, at a frequency of about 1Hz. A preload of 5N was applied to the samples to ensure good end contact throughout the test, and this preload was subsequently kept constant as the maximum stress (σ_{\max}) in a cyclic test (Fig. 5.6).

The multi-step testing was carried out in two series: short multi-step tests and long multi-step tests. A total of six samples were used in each test series.

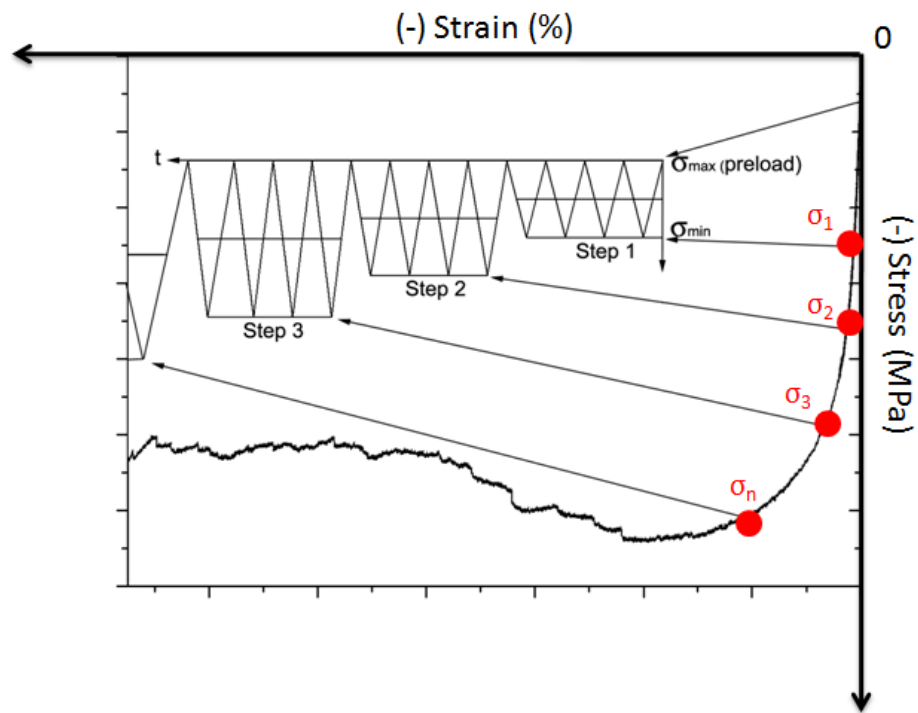


Figure 5. 6: A schematic representation of a multi-step compressive fatigue test (Guillen et al., 2011b).

Short multi-step tests

Foams and bones were cyclically tested in compression at a frequency of 1Hz. Each sample was tested under multiple stress amplitudes in steps by keeping the maximum stress constant and increasing the minimum stress incrementally. The stress amplitude of each step was kept constant for a given number of cycles between 300 and 500 cycles. The step duration for the 30ppi AlSi7Mg and 45ppi AlSi7Mg foams was 300 cycles, whilst the step duration for the 30ppi CuSn12Ni2 foams and bones was 500 cycles. For all foams, the preload was kept at 5N while for bone was 10N. For all materials, the first step had a minimum stress (σ_{\min}) of $0.5\sigma_{\text{ult}}$ (Fig. 6.9). The value of σ_{\min} at the first step was chosen so that the sample remains essentially elastic. For the subsequent steps the σ_{\min} was increased by 0.1MPa per step for the 30ppi AlSi7Mg and the 45ppi AlSi7Mg; 0.2MPa for 30ppi CuSn12Ni2 and 0.5MPa for the cancellous bone until failure.

For further analyses the secant modulus (E) was calculated, where E was defined as $\Delta\sigma/\Delta\varepsilon=(\sigma_{\max} - \sigma_{\min})/(\varepsilon_{\max} - \varepsilon_{\min})$. The secant modulus was measured at steps during the test to monitor the onset and development of fatigue damage. According to some authors (Michel et al., 1993; Palissery et al., 2004; Dendorfer, 2008; Gibson et al., 2010), fatigue failure for cellular materials may be defined in terms of specific reduction in the secant modulus. In this work, failure was considered as a reduction by more than 10% of the secant modulus as compared to the secant modulus of the first cycle (E_0).

Cyclic yield stress (σ_{cy}) of samples was obtained and used for setting the long multi-step tests. Cyclic yield stress was defined as the stress amplitude where irreversible deformation of the cell-struts took place. The measurement of the cyclic yield stress was obtained by analysing E

in each step of the fatigue test and cyclic yield stress was considered as a reduction by more than 5% of the secant modulus compared to the E_0 (Michel et al., 1993; Ganguly et al., 2004).

Long multi-step test

Long multi-step tests were carried out for all materials at four steps of 5,000 cycles and a frequency of 2Hz. The values for σ_{\min} were chosen as: $0.4\sigma_{cy}$, $0.6\sigma_{cy}$, $0.8\sigma_{cy}$, and σ_{cy} . The purpose of this test was to examine the long term cyclic behaviour of the materials. Special attention was placed on the pseudo-saturation regime (PSR) (Dendorfer, 2008) of the strain vs cycle curve.

For both short and long multi-step tests the minimum strain (ϵ_{\min}) and the maximum strain (ϵ_{\max}) were monitored and strain vs cycle curve and hysteresis loops obtained. In addition, mean strain was also obtained, so that the influence of mean strain as well as strain amplitude on the evolution of deformation of foams and bones may be assessed.

Failure analysis

Fracture surfaces of the specimens after multi-step tests were analysed using SEM. The specimens were carefully extracted from the grips post testing. In order to avoid additional damage, the samples were cold embedded and ground until the deformed cell struts were found. Prior to the SEM analyses the samples were prepared using the same procedure reported above (Paragraph 5.3).

5.4.2. Results

Fig. 5.7 shows the monotonic compression and the multi-step fatigue test results of the 30ppi AlSi7Mg foam. With the determination of σ_y and σ_{ult} in the compression test (Fig. 5.7a),

it was possible to establish an initial stress step for the multi-step fatigue test. The fatigue test started with a minimum stress of $0.5\sigma_{ult}$, where linear elastic response was observed (Fig. 5.7b). Thereafter, the main phases of the cyclic deformation behaviour may be classified as deformation in the transient region, evolution of the deformation during the pseudo saturation region (PSR), deformation at the defined failure and catastrophic failure (Michel et al., 1993; Moore & Gibson, 2001, 2003, 2003b; Ohrndorf et al., 2006).

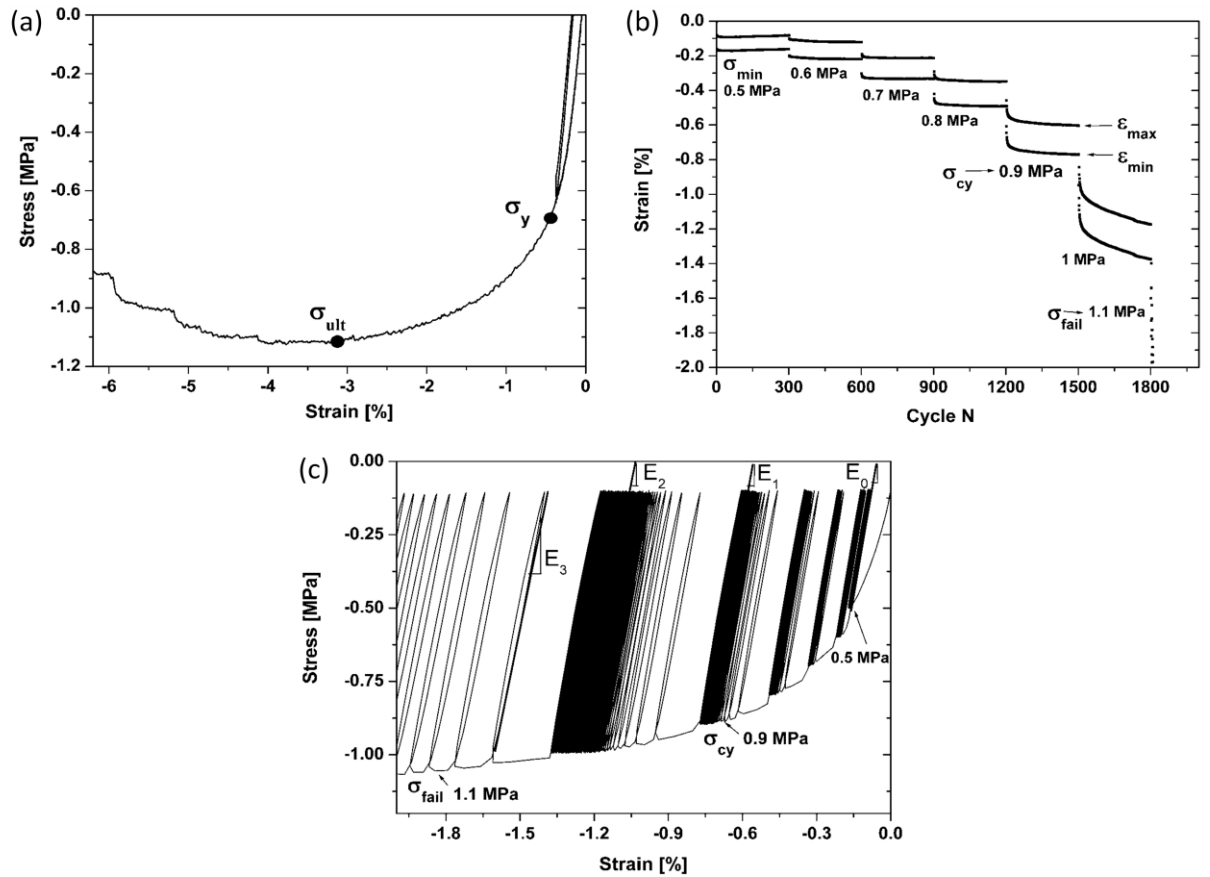


Figure 5. 7: Mechanical behaviour of 30ppi AlSi7Mg: (a) Monotonic compression test curve, (b) max/min strain of the short multi-step test, (c) hysteresis loops of the short multi-step test (Guillen et al., 2011b).

The cyclic yield stress σ_{cy} of 30ppi AlSi7Mg was about 0.9MPa (Fig. 5.7b and c). This value was slightly higher than the compressive yield stress σ_y (Fig. 5.7a). Fig. 5.7b shows that after the transition phase, the mean strain remains mostly constant during each step. However, at the step with σ_{min} at 0.9MPa a decrease in the mean strain appeared, indicating possibly irreversible deformation of the struts.

The secant modulus at the first step ($\sigma_{min}=0.5\text{MPa}$) was $E_0 \approx 309\text{MPa}$. E was then measured at the subsequent steps and found that at $\sigma_{min}=0.9\text{MPa}$ the secant modulus (E_1) was reduced by approx. 5% (Fig. 5.7c). Finally, at a stress near to catastrophic failure ($\sim 1.1\text{MPa}$), the reduction of E_3 was around 20%. At this point, a drastic increase of the compressive mean strain was measured just prior to the catastrophic collapse of the cell struts. This behaviour was found to be common for all the metallic foams. For the 45ppi AlSi7Mg foam similar values of σ_{cy} and σ_{fail} were obtained (Table 5.3).

Table 5. 3: Comparison of monotonic (see Table 5.2 in Paragraph 5.3) and multi-step fatigue test results (Guillen et al., 2011b).

	30ppi AlSi7Mg	45ppi AlSi7Mg	30ppi CuSn12Ni2	Bone
Compressive stress at cyclic yield σ_{cy} (MPa)	0.75-1.0	0.85-1.1	3.5-4.4	3.0-5
Compressive stress at failure σ_{fail} (MPa)	1.0-1.4	1.1-1.5	3.8-5	3.5-6.5
Compressive mean strain at cyclic yield ϵ_{cy} (%)	0.4-0.7	0.75-1.3	0.35-0.7	0.5-0.6
Compressive mean strain at failure ϵ_{fail} (%) (cyclic test)	1.1-2.0	1.2-2.3	1.2-1.8	1.7-2
Monotonic yield strain (%)	0.42-0.75	0.41-0.65	0.6-0.73	1.27-1.96
Monotonic strain at σ_{ult} (%)	3.5-5.9	2.8-5.4	4.5-9.4	2.2-3.8

The monotonic compression and the multi-step fatigue test results of the 30ppi CuSn12Ni2 foam are shown in Fig 5.8. The compressive mean strains at σ_{cy} were in the range of 0.35 to 0.70, again, close to that for monotonic yield strain (0.6–0.73) (Table 5.3). For the secant modulus at σ_{cy} approximately 3.6MPa was found. This value was selected to set up the long multi-step tests of the bronze foam samples.

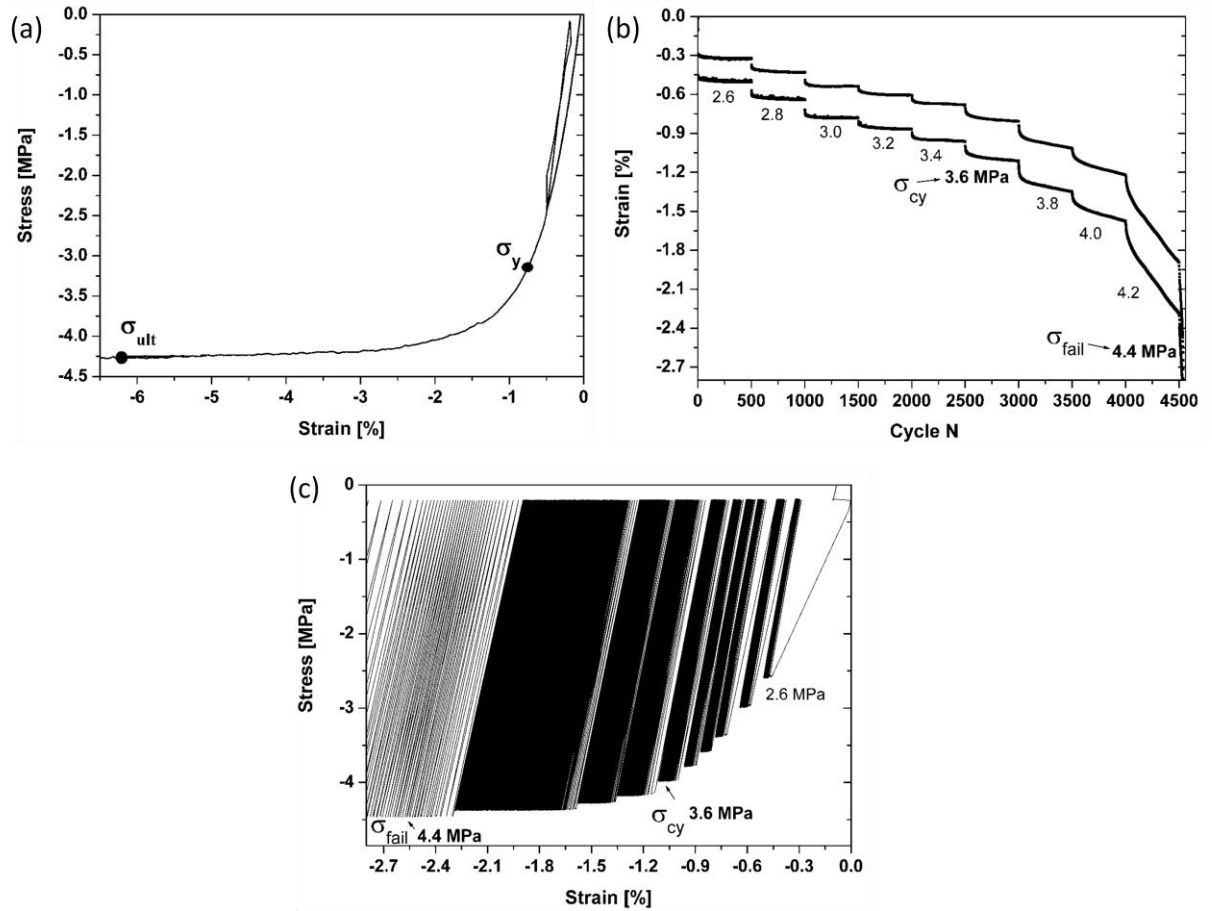


Figure 5. 8: Mechanical behaviour of 30ppi CuSn12Ni2: (a) Monotonic compression test curve, (b) max/min strain of the short multi-step test, (c) hysteresis loops of the short multi-step test (Guillen et al., 2011b).

The behaviour of trabecular bones during monotonic compression and multi-step tests is shown in Fig. 5.9. As in the foams, it is possible to see an increase of the residual strain during the fatigue test.

Fig. 5.9b shows that the compressive value of ϵ_{\min} increases at each step, whilst the slope of ϵ_{\max} stays almost linear until σ_{cy} , when the strain rate increased drastically due to the progressive damage accumulation. Similarly to the foams, hysteresis loops in Fig. 5.9c show non-linearity and the shift of the loops. Nevertheless, the hysteresis loop shift showed by bones was continuous along the entire multi-step test. The cyclic yield stress σ_{cy} was found to be between 3 and 5 MPa. These values were obtained when E decreased by more than 5%.

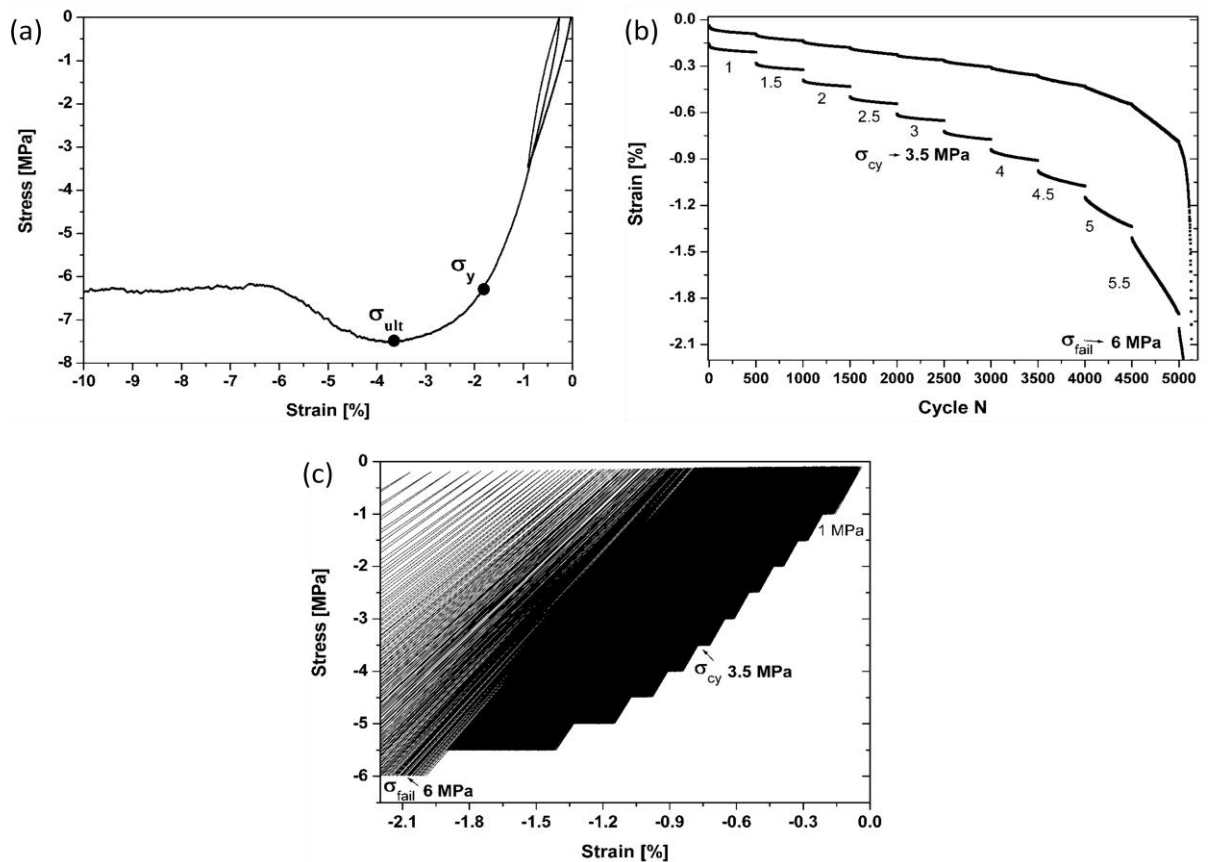


Figure 5. 9: Mechanical behaviour of trabecular bone: (a) monotonic compression test curve (b) max/min strain of the short multi-step test, (c) hysteresis loops of the short multi-step test (Guillen et al., 2011b).

Fig. 5.10 shows the long multi-step test results for the 45ppi AlSi7Mg foam and trabecular bone.

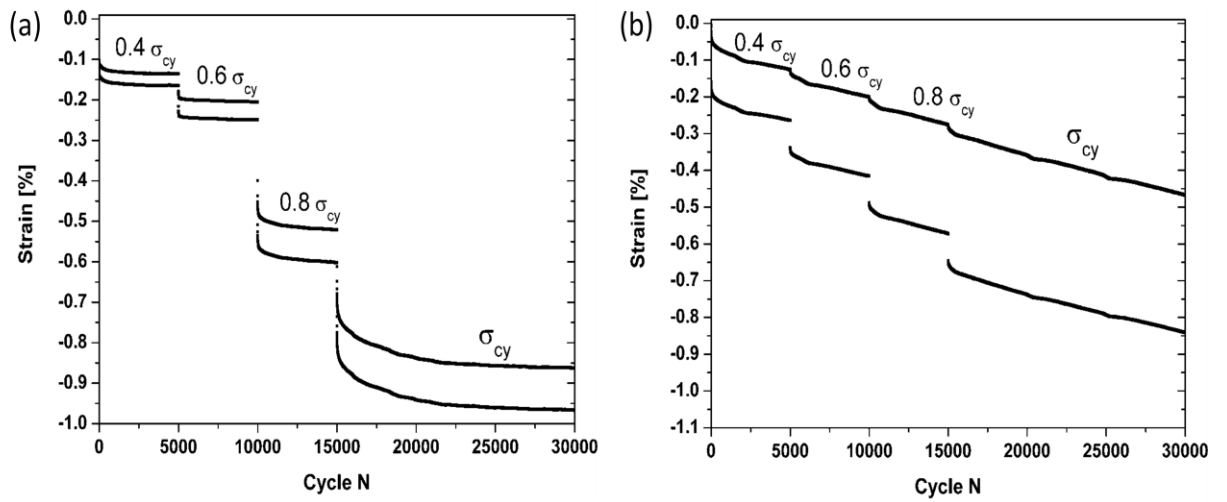


Figure 5. 10: Strain development from a multi-step fatigue test of (a) 45ppi AlSi7Mg and (b) trabecular bone (Guillen et al., 2011b).

The cyclic deformation behaviour of the foam was similar for the long multi-step test to that of short multi-step test. The values of the mean strain at failure were very similar to those showed in Table 5.3. The samples were cyclically loaded at steps up to 30,000 cycles. About 5,000 cycles were allowed at the last step (σ_{cy}) to record damage accumulation leading to catastrophic failure.

Bones in the long multi-step test show similar behaviour to that observed in the short multi-step tests. Damage accumulated continuously during the test, and catastrophic failure was observed at 30,000 cycles. A comparison of the short and the long multi-step test results reveals that failure occurs only when a compressive mean strain of 1.7-2% is reached. This value is similar to the mean failure strain found for the foam samples (Table 5.3).

SEM micrographs of the failed samples are shown in Fig. 5.11. White vertical arrows indicate the loading directions during the fatigue tests.

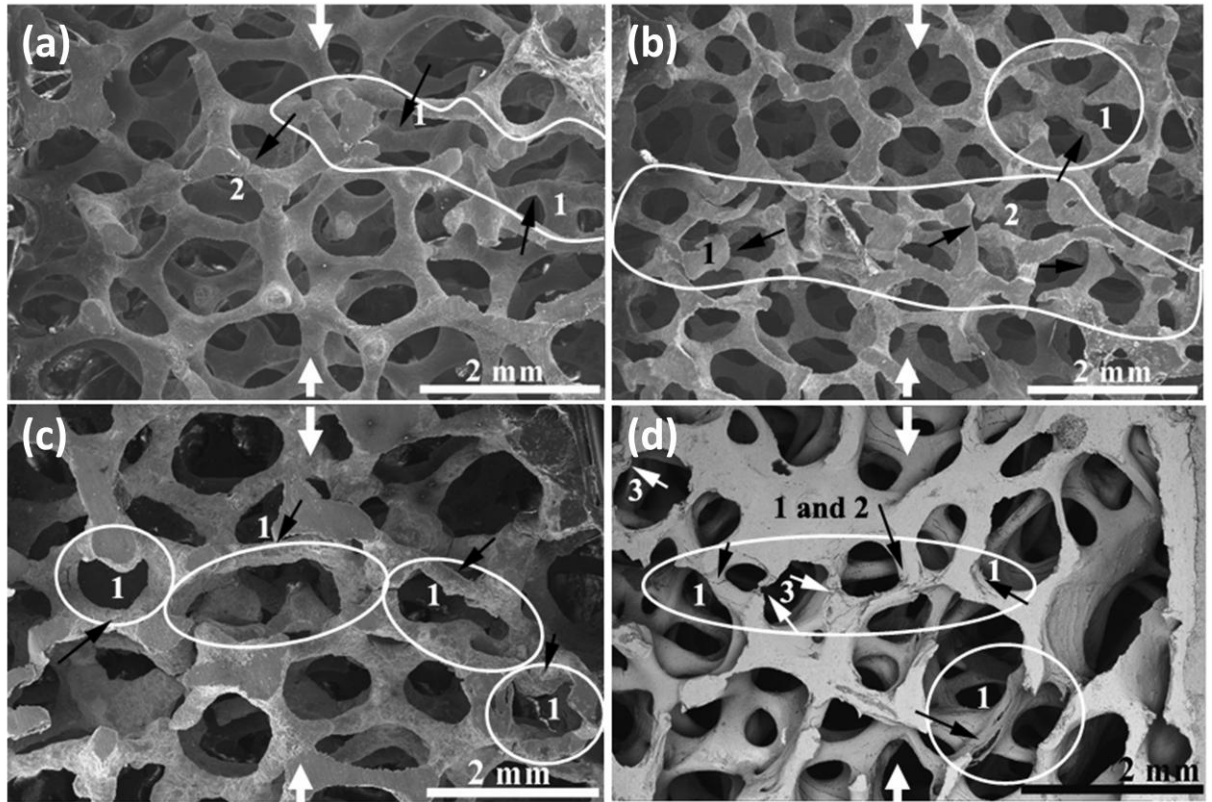


Figure 5. 11: Failure analysis of the samples using SEM (failure mode 1: buckling, 2: rupture, 3: splitting): (a) 30ppi AlSi7Mg, (b) 45ppi AlSi7Mg, (c) 30ppi CuSn12Ni2 foam and (d) trabecular bone (Guillen et al., 2011b).

SEM analysis revealed that the main failure mechanism of the foams was buckling (1), caused by the accumulation of residual strains during the tests. 30ppi AlSi7Mg and 45ppi AlSi7Mg show additional rupture without buckling (2).

A BSE detector was used to examine bone samples as it is more sensitive in detecting failures on the surface of the bone compared to the SE detector. BSE micrographs of the tested

trabecular bone reveal that the samples failed by a combination of buckling (1), splitting (3) and brittle failure (2) in some trabeculae (Fig. 5.11d).

5.4.3. Discussion

The short multi-step test showed how the increase of the residual strain, or the accumulated plastic deformation, was the principal mechanism for material failure. This phenomenon is also called cyclic creep (Carter, 2005) which results in a shift of the hysteresis loops along the strain axis. In addition, broadening and increased nonlinearity of the hysteresis loops were also observed when the mean strain exceeded σ_{cy} (Fig. 5.7(c)). It seems that cyclic creep takes place principally in the transient region of each step and the compressive mean strain at cyclic yield is very similar to the monotonic yield strain (Table 5.3). The cyclic creep rate is highest at the beginning of every step. Subsequently, the rate reduces during the pseudo-saturation phase. This cyclic creep transition phase is short for the steps at stress amplitudes of 0.5, 0.6 and 0.7 MPa, whilst at 0.9MPa and above saturation is no longer attainable.

For the 30ppi CuSn12Ni2 foam cyclic deformation is also principally due to cyclic creep. The strain rate in the transitional phase of each step appears to change slower than that in the AlSi7Mg foams (Fig. 5.8b and c). This may be explained by the more ductile behaviour and homogeneity of the bronze foam. Fig. 5.8b shows that the cyclic creep at the transition phases increases for stress closer to σ_{ult} , although the increment (0.2MPa) was larger than that for the AlSi7Mg foams (0.1MPa). The mean cyclic strains of the bronze foam are similar to those obtained for the AlSi7Mg foams.

The compressive residual strain in bones increased with a largely constant rate during the entire test, as opposed to the foams where transient behaviour occurred primarily at step

changes. Unlike in the foams, the cyclic mean strain at yield was in the range of 0.5-0.6, much lower than the range of the monotonic yield strain (1.27-1.96); although the compressive mean strain at failure (1.7-2) is close to the monotonic yield strain (1.27-1.96, Table 5.3). It was noticed also that bone deformation at the steps accumulated continuously until failure. This seems to suggest that the secant modulus may be used to characterise the damage of the structure. Based on previous studies (Pattin et al., 1996; Zioupos et al., 1996; Yeni & Fyhrie, 2002), this behaviour can be explained by the gradual alteration of trabecular bone stiffness caused by damage accumulation. This may lead to the initiation of microcracks resulting in catastrophic failure of the bones. Consistent with the previous analysis, trabecular bone seems to deform by cyclic creep also, but in this case the residual strain accumulated continuously during the entire test.

The aim of the long multi-step tests was to observe the long term behaviour of the samples at stress amplitudes below and equal to σ_{cy} . In these cases the samples were not tested until failure, and the attention was placed on the cyclic deformation behaviour of bones and foams during the long periods under selected stress amplitudes.

The analyses revealed that cyclic creep plays again an important role in the deformation behaviour of the samples at each step for both foam and bone. The cyclic creep rate increased when σ_{min} approached σ_{cy} . Comparing the results of the short and long multi-step test of foams and bones, it seems that the compressive mean strains at yield are similar for both foams and bone, and they are again close to the monotonic yield strain, at least for the foams. In terms of σ_{cy} and σ_{fail} , the values are similar between 30ppi CuSn12Ni2 foam and the bone. Although both foams and bone fail by cyclic creep, the deformation behaviour in the transient region of each step was different for both materials. Fig. 5.10a shows that ϵ_{max} of

foams decreased abruptly during the transient region of the each step. This behaviour may be explained by rapidly developing microdamage (possibly collapse) in the cell struts that occurred during the initial stage of each step. Bones show a gradual decrease of ϵ_{\max} (Fig. 5.10b). The behaviour during the transient regions in the bone suggests that the effect of microdamage in the trabeculae during the step changing plays a less significant role in the overall deformation compared to the foams. The damage in bones was found to be accumulated continuously during the fatigue test until failure.

The failure analysis may explain the behaviour of AlSi7Mg foams during the multi-step tests. These samples showed the largest increment in compressive mean strain during the transition phase of each step. This may be considered as evidence of a combination of brittle and buckling failure behaviour of AlSi7Mg foams, as opposed to pure buckling failure in 30ppi CuSn12Ni2 foam where ductile behaviour seems to prevail. The characteristic failure mode observed in the bone may be explained by its complex nature, where characteristics such as viscoelasticity, irregular cell architecture and a hierarchical structure of the trabeculae are present, which may lead to a more complex deformation mechanism compared to those in the foams.

5.5. Foam-cement interface under mixed-mode loading conditions: a preliminary study

5.5.1. Experimental procedures

The AlSi7Mg (30ppi and 45ppi) and CuSn12Ni2 (30ppi) foam materials were used to interdigitate with acrylic bone cement (Simplex P, Stryker, UK) to create foam-cement interface samples. The rectangular foam coupons were machined to size

(16mm×10mm×5mm) using a low speed diamond saw with constant mineral oil irrigation to avoid abnormal high temperatures with consequent local melting of the foams (Fig. 5.12).

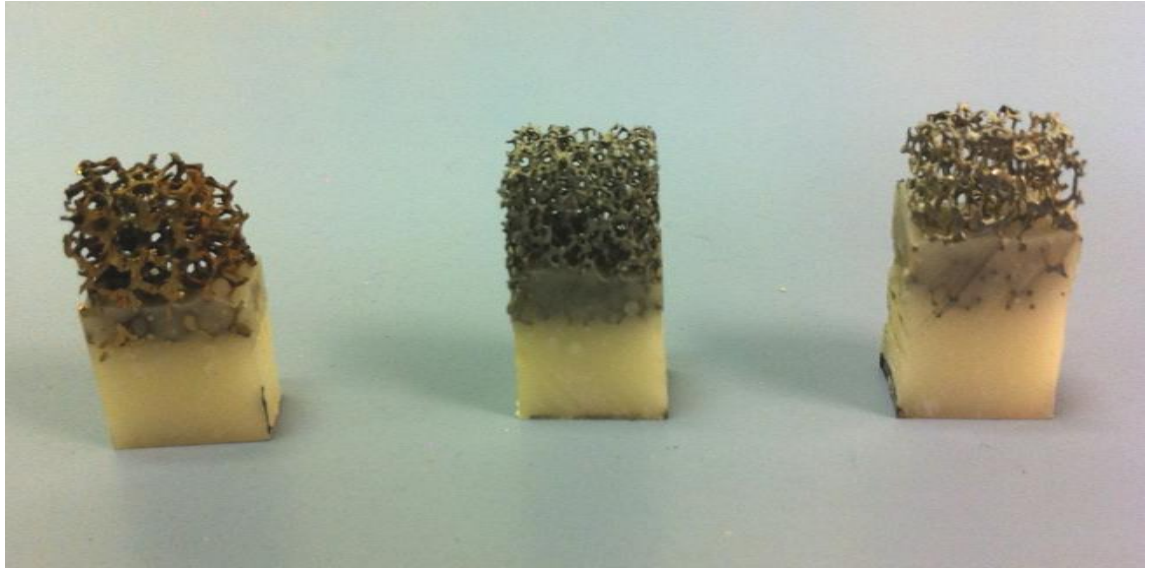


Figure 5. 12: The foam-cement interface samples used in this study.

The same mixed-mode experimental setup described in Paragraph 4.3 was used. A total of 30 samples were tested at 5 selected loading angles (0° , 22.5° , 45° , 67.5° and 90°); 2 tests were carried out at each loading angle for each foam material.

SEM analysis and stereoscopic microscopy of the fractured surface were carried out to assess both failure type and morphology.

The results were then compared with some of those obtained for bone-cement interface under the same loading conditions, to evaluate the potential suitability of the three foams as cancellous bone substitutes for cemented biomechanical testing.

5.5.2. Results and discussion

Fig. 5.13 shows an overview of the stress-displacement behaviour of the three foams and bone-cement interface for the five different loading conditions. The initial quasi-elastic section was followed by bending of the individual cells and struts. Consequently, a general reduction of the slope is observed until the ultimate stress was reached. The post-yield softening behaviour is characterised by discontinuities as a result of the progressive buckling and fracture of the struts at the interface. This behaviour seems more evident under tensile (0°) and mixed-mode until 67.5° , and differs from that observed in bone-cement samples, where the cement pedicles were pulled-out of the bone matrix gradually until a critical stage was reached when fast fracture occurred. The mixed-mode tests between 0° and 67.5° seem to suggest that foam material dictates the mechanical response of the foam-cement interface, with CuSn12Ni2 foams exhibiting the highest strength. Some variations in the measured ultimate strength were observed, especially for AlSi7Mg. This can be explained by the variation of cell and strut dimensions as well as slight differences in the dimension of the samples. Ultimate stresses for both foams and bone tested at the first 3 loading angles (0° , 22.5° and 45°) did not change significantly. Although the variation in ultimate stresses was small, the energy absorbed seemed to be reduced by the increment of the shear component, resulting in a less prolonged softening. At 67.5° , where shear is the main component, an important increase in the properties was observed for CuSn12Ni2 and bone, whilst only moderate for the two AlSi7Mg. Fig. 5.13e shows the behaviour of the samples tested at 90° . The load applied in this case was pure shear and the highest values of strength were obtained for all samples.

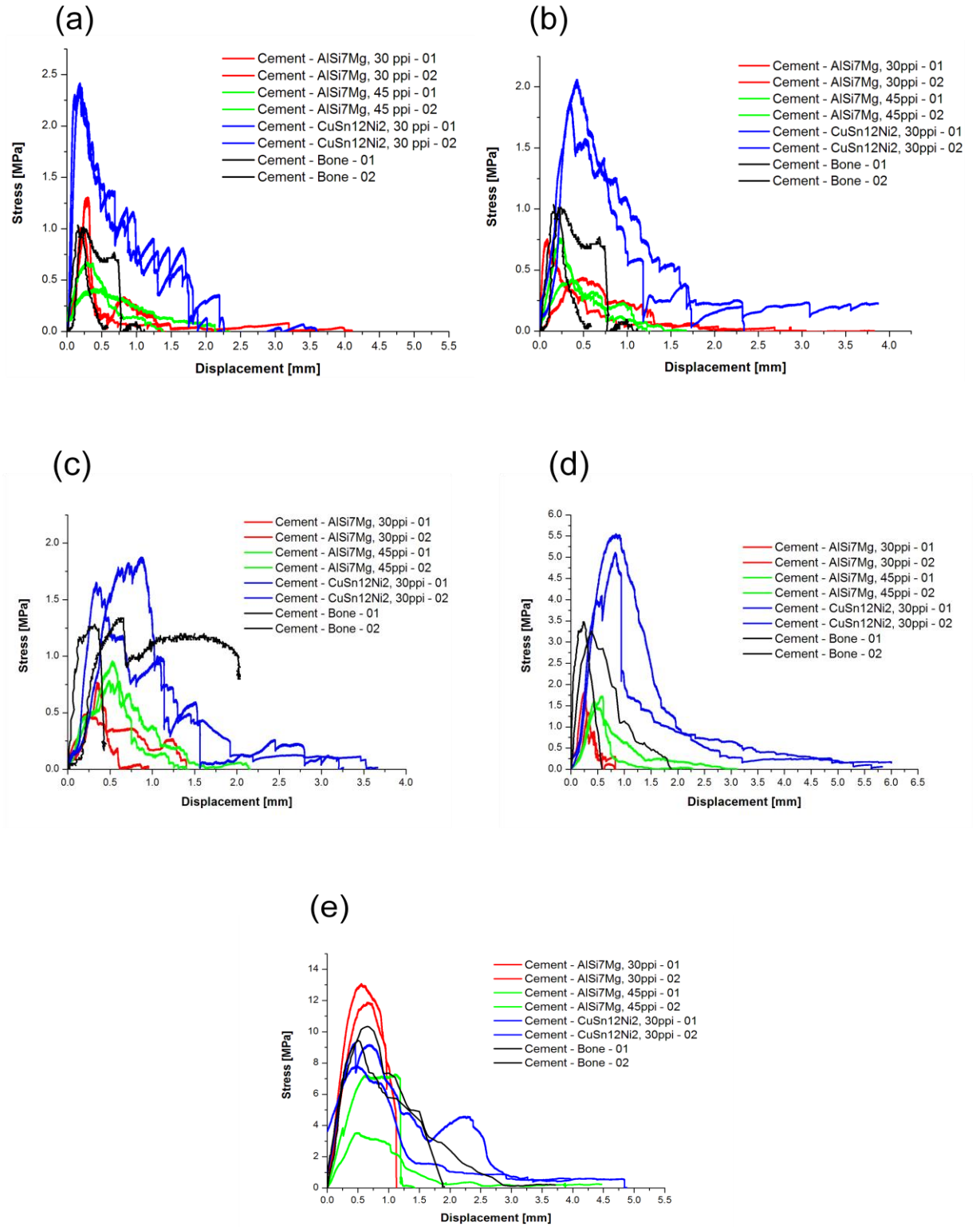


Figure 5. 13: The stress-displacement responses of the three foam-cement and bone-cement composites tested under mixed-mode loading conditions at angles of (a) 0°, (b) 22.5°, (c) 45°, (d) 67.5° and (e) 90°.

Although the measured strengths differed, there was no clear difference in the behaviour of the three types of foam-cement interface. Moreover the mechanical response of the 30ppi AlSi7Mg was found to be higher than that of 30ppi CuSn12Ni2 for shear loading condition, suggesting that shear may have obscured any difference due to foam material properties.

Linear regression analysis (Fig. 5.14) revealed a positive ascending slope with a general increment of the stresses towards the 90° load angle, with a similar trend reported for the bone-cement interface (Wang et al., 2010). A moderate correlation for cancellous bone and the three foam materials was found with r^2 values falling in the range between 0.52 and 0.72. Due to the small number of samples tested for each loading condition, no statistical analysis was performed.

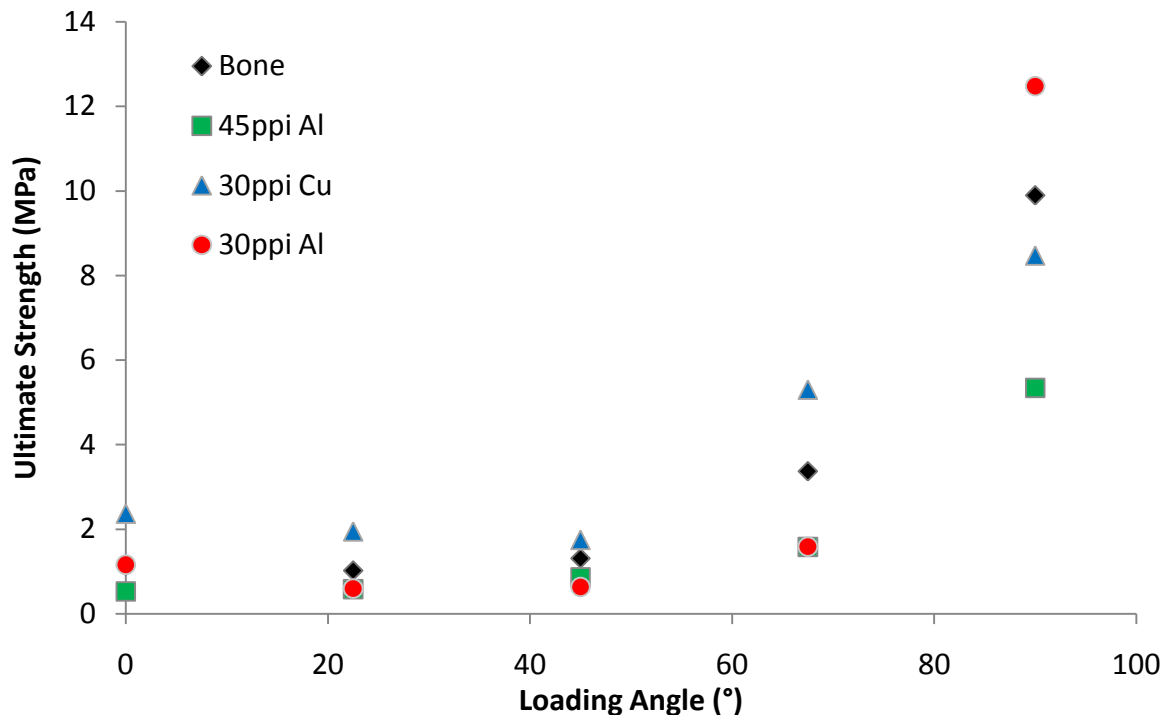


Figure 5. 14: The apparent interfacial strength as a function of the loading angle.

The failure types (as defined in Chapter 4) of each failed sample were analysed with optical stereoscopy and SEM. Two types of failure mode were identified in the current study, independent of the foam materials. Type III failure was found for the mixed-mode tests at 0°, 22.5°, 45° and 67.5° and type I for the 90° case. In these cases the main failure occurred completely in the foam material without any apparent damage at the interface and the distance of the broken cell struts from the interface layer can vary, although does not appear to be related to the loading angle. Fig. 5.15 shows a 30ppi AlSi7Mg foam-cement sample after testing at 0°.

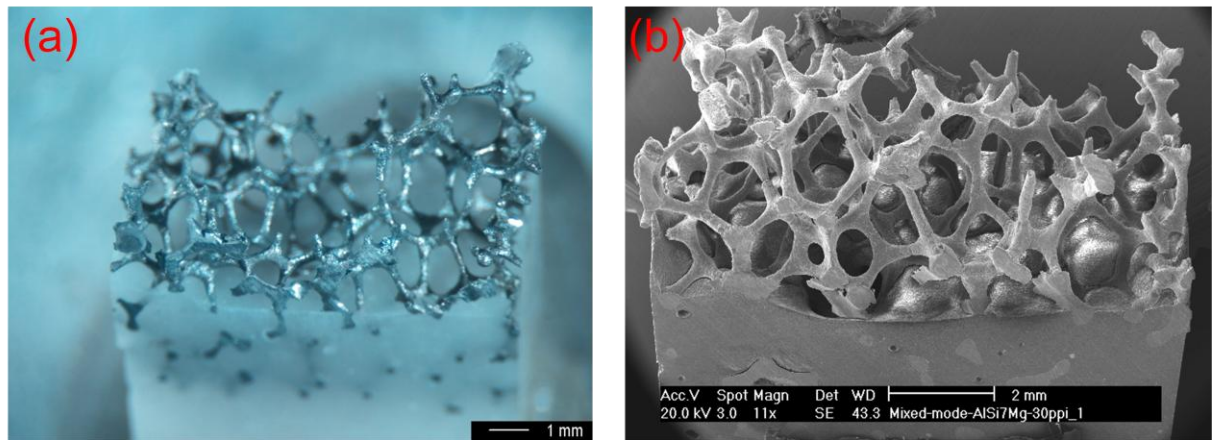


Figure 5. 15: (a) Optical and (b) SEM image of a failed foam-cement interface tested at 0°.

The foam-cement samples tested at a loading angle of 90° (shear) showed different failure (type I) compared to that of the samples under other loading angles. Fig. 5.16 shows that the shear load was concentrated on one side of the interface, resulting in a “cut-like” failure adjacent to the loading point, while failure by bending and rupture of the cell struts occurred in the far side. Despite a small number of sample were tested, these observations seem to indicate once again the important role of the loading condition, in particular shear (90°), in the overall behaviour of the foam-cement composites.

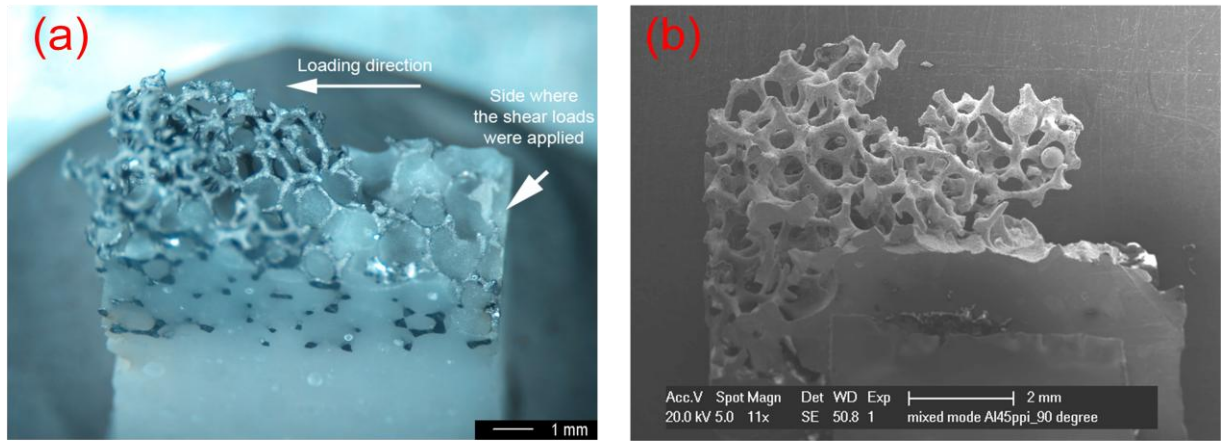


Figure 5. 16: Optical (a) and SEM (b) of a failed foam-cement interface sample tested at 90°.

5.6. Conclusions

In summary μ CT characterisation and compressive mechanical testing of three open-cell metallic foams and bovine trabecular bone seem to suggest that, despite the apparent differences in their respective microstructures and mechanical properties, the deformation behaviour is similar across the bone and the foams.

The multi-step tests were useful in assessing the cyclic deformation behaviour of metallic foams and bones, reducing the number of samples needed for a cyclic analysis of the materials. The results showed that the increase of residual strain, or accumulated plastic deformation, was the principal deformation mechanism. This effect is observed when σ_{cy} was exceeded; thereafter cyclic creep rate accelerated until catastrophic failure occurred. Cyclic creep plays an important role in cyclic deformation of trabecular bone also, and the residual strain increased continuously during the entire test. Optical microscopy and SEM revealed that the foams failed principally due to buckling caused by the accumulation of residual

strains during the test; whilst bones failed mainly by a combination of buckling and brittle fracture.

Bone-cement interface mechanical responses fell within the foam-cement experimental results for the selected mixed-mode loading conditions. The apparent strength as a function of the loading angle showed a similar trend with a moderate correlation for all the samples tested. The preliminary results seem to confirm the importance of loading condition in dictating interfacial responses under predominantly shear loads.

The results reported in this chapter might be of interest in the exploitation of foams as substitutes of trabecular bones for further biomechanical studies.

Chapter 6

Discussion, conclusions and future work

6.1. Discussion

The main aim of this work was to provide a better understanding on how the bone-cement interfacial behaviour in THR, with particular regards to the acetabular side of the implant, is affected by environmental (i.e. wet conditions, temperature), structural (i.e. depth of cement penetration, contact area, BV/TV) and operational (i.e. loading condition) variables.

The analyses were conducted at different levels of studies and using different investigation techniques (i.e. hip simulator, μ CT, IGFA, SEM), from a macro level (acetabular reconstructs in wet environment) to a micro-level (bone/foam-cement interface coupons under selected loading conditions). Prior to employing specific open-cell metallic foams for a biomechanical characterisation of the interface, their suitability as cancellous bone substitutes was also explored.

At Portsmouth, a unique hip simulator for fixation endurance testing of cemented acetabular reconstructs was developed, where realistic physiological loading waveforms (Bergmann et al., 2001) were realised (Zant et al., 2008). More recently, a combined load profile representative of patient routine activities was also used to study further the long-term structural integrity of acetabular fixation (Wang et al., 2009). All previous studies (Tong et al., 2008; Zant et al., 2008; Wang et al., 2009) showed that bone–cement interfacial debonding was observed to start from superior-posterior quadrant, and the damage development became more extensive with the increase of the body weight. Furthermore, the debonding appeared to start from the dome region of the acetabulum (zone II), developing towards zones I & III. However, all the previous investigations have been conducted in dry conditions, where the effect of environment on the fixation integrity was not considered. In this work

(Chapter 3) bovine acetabular reconstructs were tested in a controlled wet environment at 37°C, through a new experimental protocol for the first time. The preliminary results have provided a more balanced failure scenario for damage initiation and evolution of the reconstructs in both dry and wet conditions. In this sense, although interfacial debonding at the bone-cement interface was identified as the failure mechanism in wet consistently with dry experiments, reduced survival lives were observed for the wet case (i.e. about 100,000 compared to 200,000 cycles vs 3-4 million cycles in dry), where debonding seemed to start from the rim of the acetabulum (zones I) developing towards zone II, a phenomenon consistent with clinical observation (Garcia-Cimbrelo & Munuera, 1992).

To better understand the failure mechanisms and mechanics of the bone-cement interface, a more detailed analysis was needed. The scale of investigation was reduced (Fig. 6.1) from the whole hemi-pelvic acetabular reconstruct (macro), to small bone-cement interface coupons (micro) tested under selected loading conditions (tensile, compression, mixed-mode, shear), using different experimental techniques, from servo-hydraulic testing machines to μ CT imaging and image-guided failure assessment (IGFA).

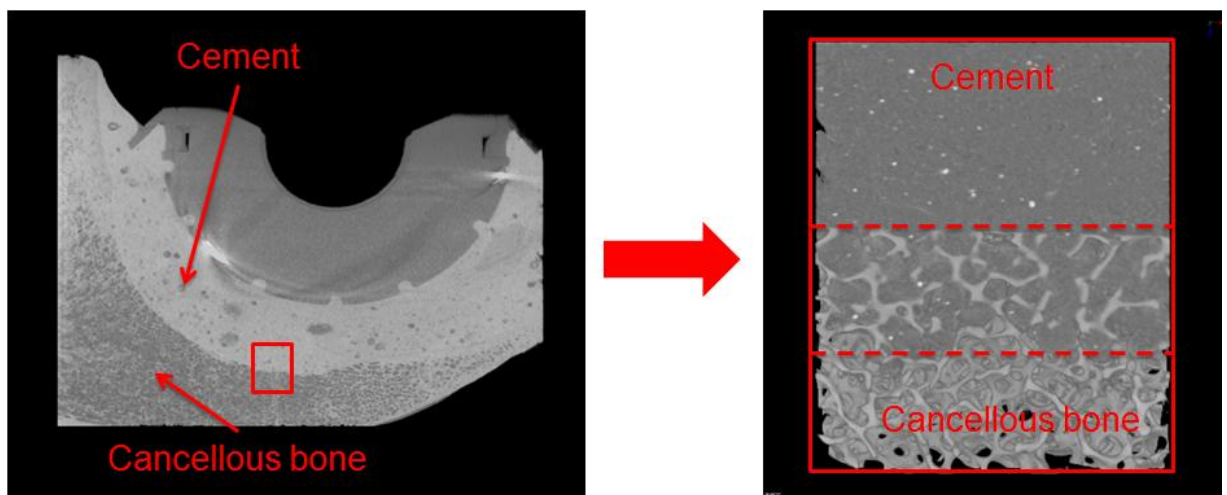
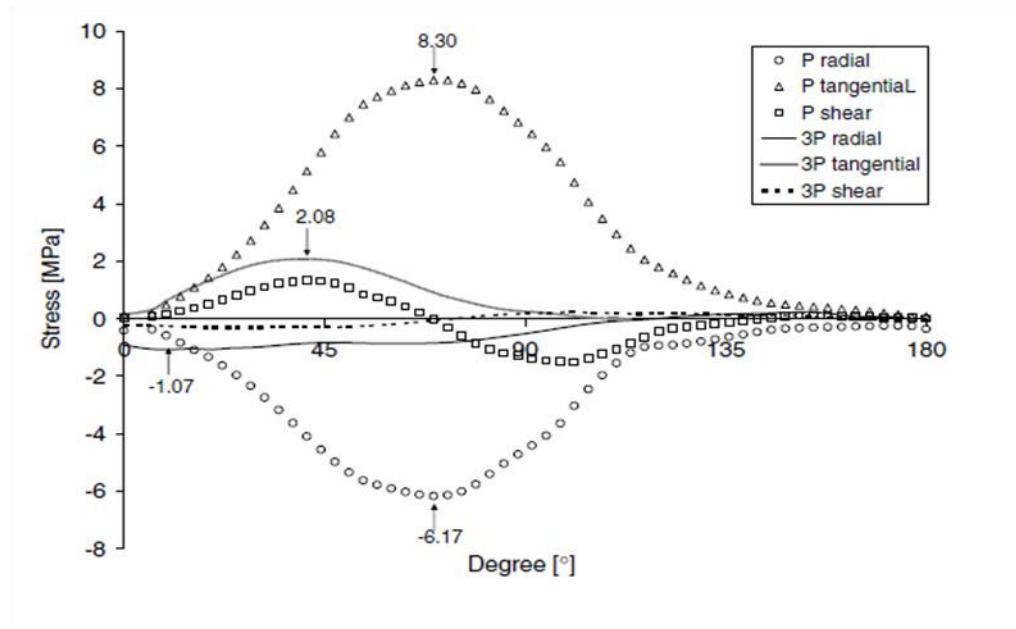
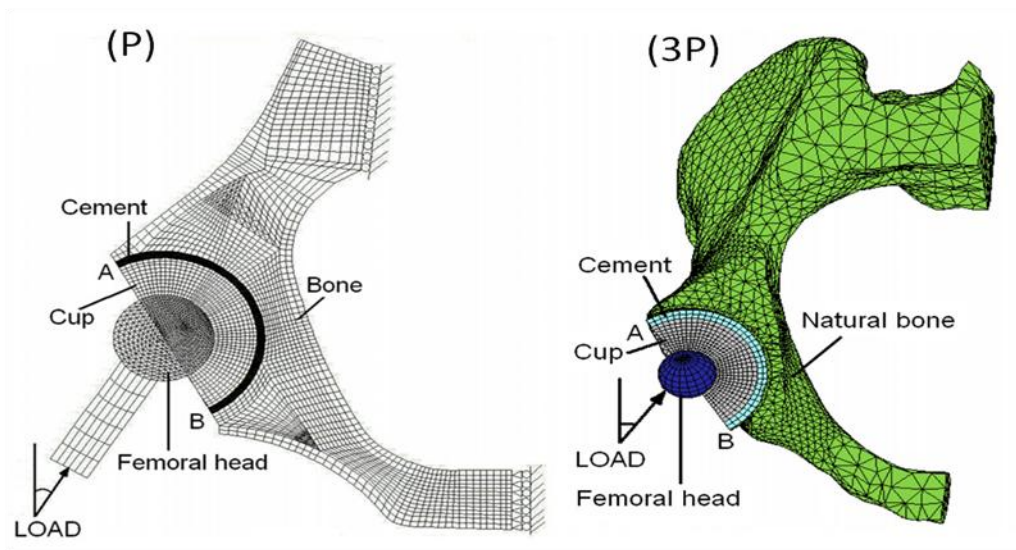


Figure 6. 1: Bone-cement interface transition from macro to micro level.

These typical loading conditions were observed in a typical cemented acetabular replacement, as shown in Fig. 6.2 (Tong et al., 2008).



(a)



(b)

Figure 6. 2: (a) Local stress distributions at the bone-cement interface at peak hip contact force during normal walking; (b) the plane strain pelvic bone (P) and 3D pelvic bone (3P) models (Tong et al., 2008).

In order to have a direct term of comparison, the interface coupons were prepared and the results mainly compared with the work of Mann and his associates (Mann et al., 1997, 1999, 2001, 2008, 2009; Miller et al., 2012; Kim et al., 2004). In Chapter 4 an attempt was made to answer two very important research questions: (1) Is the bone-cement interfacial strength correlated to measurable parameters such as cement penetration into the cancellous bone ($r^2=0.025-0.7$ (Krause et al., 1982; Mann et al., 1997, 1999, 2001)) or loading direction ($r^2\approx 0.6$; $p>0.05$ for tensile, mixed-mode and shear (Mann et al., 2001))? (2) What is the main damage mechanism at the interface under uniaxial compression, which is very important in acetabular implants? All the results obtained in the present study seem to suggest that, despite the difference in the loading conditions and the preparation methods, there was virtually no correlation between the interfacial strength and cement penetration ($r^2=0.16-18$). However, the interfacial strength was found to correlate quite well with the loading angle ($r^2=0.77$), despite the overall effect of the loading angle on the failure response was not significant (ANOVA, $p=0.07$). With regards to the second question, the damage evolution at the bone-cement interface under uniaxial compression showed that trabecular bone sustained most of the load, which was transferred as a result of progressive failure of trabeculae in the proximity of the interfacial contact region, although there are some significant scatter in the results ($SD=\pm 1.10\text{MPa}$). There may be several reasons for all these, but the most important of all might be the general mechanical and morphological anisotropic behaviour of the cancellous bone used to interdigitate with the bone cement.

The work reported in Chapter 5 aims to overcome the problem related to the trabecular bone anisotropy by using three open-cell metallic foams as cancellous bone analogous models for cemented biomechanical studies. Furthermore, materials with known consistent

and controllable mechanical properties within the range of properties of cancellous bone may provide valuable alternatives to cadaver bones as test substrates especially in terms of availability (Szivek et al., 1995). Three different foams (30ppi, 45ppi AlSi7Mg and 30ppi CuSn12Ni2) were characterised in the current study under uniaxial and fatigue compression. The results showed that, despite the CuSn12Ni2 foam exhibited better static mechanical properties and the AlSi7Mg (45ppi) a closer morphology compared to cancellous bone, the cyclic behaviour was similar across the materials with accumulated plastic deformation (cyclic creep) as the principal failure mechanism. Finally, all the three metallic foams were used to interdigitate with bone cement to create foam-cement interface samples that were tested under mixed-mode loading conditions. Although only preliminary, the results obtained seem to confirm the role of the loading angle, in particular shear condition (90°), in dictating the failure mechanism at the interface. However, further systematic studies are needed to obtain a more comprehensive understanding of the bone-cement mechanics.

6.2. Conclusions

Failure at the bone-cement interface has been identified as the predominant failure mode in cemented arthroplasty, where the lasting integrity of the bond between the bone and the cement decisively defines the long-term stability of the implant. Although observations seem to suggest that debonding of the bone-cement interface is mainly responsible for the fixation failure, the contribution and the relative importance of mechanical and biological events on the implant failure is still a matter for discussion. The micromechanical behaviour of the interface is complex due to the substantial heterogeneity, anisotropy, variation of the morphology of the cancellous bone and the loading conditions. As a result, the bone-cement

interfacial properties are difficult to determine by means of conventional experimental techniques; whilst the introduction of bone analogous materials, with controllable microstructures and adjustable densities, may help to obtain a better understanding of the bone-cement mechanics providing more consistent results.

In this work, an investigation of the bone-cement interfacial properties has been carried out using a range of experimental techniques, from structural endurance testing to image-guided failure assessment (IGFA) using μ CT. The following main conclusions may be drawn:

Based on the fixation endurance testing of cemented acetabular reconstructs using the hip simulator in wet condition under a combined loading block in a saline environment:

- Preliminary results seem to indicate that the saline solution, when combined with physiological loading conditions, plays an important role in reducing the survival lives of the reconstructs.
- Although debonding at the bone-cement interface has been found as the main failure mechanism in both wet and dry conditions, damage in wet condition seems to appear after only 200,000 cycles as opposed to 3-4 million cycles under the same combined loading block in dry conditions (Wang et al., 2009).
- The debonding process seems to start and develop from DeLee & Charnley zone I, as opposed to zone II in dry conditions, where the combined action of mechanical loading and fluid environment is most effective and more consistent with clinical observations.

Based on the bone-cement interfacial behaviour under mixed-mode loading conditions and step-wise micromechanical compression:

- Bone-cement interfacial strength and cement penetration were not correlated ($r^2=0.16$). The effect of the loading angle seems to be more consistent in dictating the failure mechanism of the interface under mixed-mode loading conditions ($r^2=0.77$), although its overall effect on the failure response is still unclear (ANOVA, $p=0.07$). The findings were confirmed by both mechanical testing and post failure characterisation.
- The experimental results from time-lapsed μ CT imaging and in situ step-wise compression showed that load transfer in bone-cement interface occurred mainly in the bone-cement contact region. The damage resulted in progressively developed microfailures due to trabeculae bending and buckling.
- Although contact area and cement penetration depth are closely related ($r^2=0.94$), neither correlates with the measured compressive interfacial strength. A moderate correlation was found between the BV/TV and the ultimate strength of the interface ($r^2=0.71$).

Based on the mechanical and morphometrical characterisation of open-cell metallic foams under uniaxial compression and multi-step fatigue; including preliminary foam-cement interface investigation under mixed-mode loading conditions:

- μ CT characterization and compressive mechanical testing of AlSi7Mg and CuSn12Ni2 open-cell foams and bovine trabecular bone seem to suggest that, despite the apparent differences in their respective microstructures and mechanical properties, the general deformation behaviour is similar across the bone and the foams.
- The multi-step fatigue results showed that the increase of residual strain, or accumulated plastic deformation, is the principal deformation mechanism. Cyclic creep plays an important role in cyclic deformation of foams and trabecular bone, and

the residual strain was found to increase during the test. However, the deformation behaviour at the transient region of each step is different for foams and bone.

- Optical microscopy and SEM revealed that the foams failed principally due to buckling caused by the accumulation of residual strains during the test; whilst bones failed mainly by a combination of buckling, splitting and brittle fracture.
- Bone-cement interface mechanical responses seem to compare well with those of foam-cement interface for a given mixed-mode loading condition. The apparent strength correlates with the loading angle ($r^2=0.52-0.72$) for all the samples tested. The preliminary results seem to confirm that loading condition is more important than material properties in dictating the interfacial responses under predominantly shear loading conditions.

6.3. Future work

6.3.1. Fixation endurance testing of acetabular reconstructs

Admittedly, the main findings presented in this work represent a preliminary study on the effect of wet environment on the fixation of cemented acetabular reconstructs, with only two samples tested. More samples need to be tested under the same physiological loading and environmental conditions, to verify the current results and to provide statistical information. Furthermore, samples prepared with other orthopaedic procedures (i.e. reaming & drilling of the acetabulum to enhance cement penetration in bone and implant stability) could be tested using new combined loading blocks with more physiological routine activities (i.e. introduction of stumbling which can produce a joint reaction force of 870%BW

according to Bergmann et al. (2001)). These may provide valuable information for clinical applications.

6.3.2. Bone-cement interface micromechanics

In recent years the development of new experimental imaging techniques has provided important tools for the micromechanical investigation of different cellular materials. In the current study μ CT, along with IGFA, was used for the examination of bone-cement interfacial behaviour under step-wise compression. A micromechanical analysis of the interface under cyclic and mixed-mode loading conditions may be carried out to fully develop IGFA in more practical applications. Furthermore, damage progression was only visually assessed so far without any quantitative information on the relative micromotion and strain levels at the interface. In this sense techniques such as digital volume correlation (DVC) may be used in the determination of displacement and strain fields throughout the whole volume of the interface, overcoming the limitations of two-dimensional surface investigation.

6.3.3. Bone analogous materials characterisation

The same characterisation procedures used in the present study for open-cell foams may be used for preclinical assessment of commercially available scaffolds for tissue engineering applications. Detailed mechanical and morphological investigations, using trabecular bone as a control group, may provide valuable information to the manufacturers for the development of the next generation of scaffolds.

Chapter 7

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7. References

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Chapter 8

Appendix

8. Contributions to publications produced during the course of this work

8.1. Journal papers

Wang, J.-Y., Tozzi, G., Chen, J., Contal, F., Lupton, C., Tong, J., 2010. Bone–cement interfacial behaviour under mixed mode loading conditions. *J. Mech. Behav. Biomed. Mater.* 3, 392–398.

Guillen, T., Zhang, Q.-H., Tozzi, G., Ohrndorf, A., Christ, H.-J., Tong, J., 2011. Compressive behaviour of bovine cancellous bone and bone analogous materials, microCT characterization and FE prediction. *J. Mech. Behav. Biomed. Mater.* 4, 1452-61.

Guillen, T., Ohrndorf, A., Tozzi, G., Tong, J., Christ, H.-J., 2011. Compressive fatigue behaviour of bovine cancellous bone and bone analogous materials under multi-step loading conditions. *Adv. Eng. Mat.*, doi: 10.1002/adem.201180060.

Tozzi, G., Zhang, Q.-H., Tong, J., 2012. 3D real-time micromechanical compressive behaviour of bone-cement interface: experimental and finite element studies. *J. Biomech.* 45, 356-363.

Tozzi, G., Lupton, C., Heaton-Adegbile, P., Tong, J., 2012. Damage evolution in acetabular reconstructs under physiological testing in a saline environment. *J. Biomech.* 45, 405-408.

Zhang, Q.-H., Tozzi, G., Tong, J., 2012. Micromechanical damage of trabecular bone-cement interface under selected loading conditions: a finite element study. *Computer Methods in Biomechanics and Biomedical Engineering*, revised & resubmitted.

Madi, K., Tozzi, G., Zhang, Q.-H., Tong, J., Cossey, A., Au, A., Hollis, D., Hild, F., 2012. Computation of full-field displacement in a scaffold implant using Digital Volume Correlation and Finite Element Analysis. *J. Biomech.*, submitted.

8.2. Conference papers

Tozzi, G., Guillen, T., Ohrndorf, A., Christ, H.-J., Tong, J., 2010. Characterization of aluminum foam as a cancellous bone substitute. 17th European Society of Biomechanics (ESB) Congress, Edinburgh, Scotland, UK.

Tong, J., Wang, J.-Y., Tozzi, G., Chen, J., Lupton, C., 2010. Bone-cement interfacial behaviour under mixed mode loading conditions. 17th European Society of Biomechanics (ESB) Congress, Edinburgh, Scotland, UK.

Tozzi, G., Zhang, Q.-H., Lupton, C., Tong, J., 2011. A 3D real-time micromechanical study of bone-cement-interface. 23rd International Society of Biomechanics (ISB) Congress, Brussels, Belgium.

Tozzi, G., Lupton, C., Tong, J., 2011. Damage evolution in acetabular reconstruct under physiological testing in a biological environment. 23rd International Society of Biomechanics (ISB) Congress, Brussels, Belgium.

Zhang, Q.-H., Tozzi, G., Tong, J., Guillen, T., Ohrndorf, A., Christ, H.-J., 2011. Post yield behaviour of bovine trabecular bone and bone analogous materials under compression. 23rd International Society of Biomechanics (ISB) Congress, Brussels, Belgium.

Madi, K., Booker, A., Tozzi, G., Hsu, Y.-H., Lupton, C., Tong, J., Cossey, A., 2011. Viscoelastic modelling and full-field strain field computation of scaffold implants for cartilage knee repair. 2nd International Conference on Tissue Engineering (ICTE), Lisbon, Portugal.

Guillen, T., Tozzi, G., Ohrndorf, A., Tong, J., Christ, H.-J., 2011. Substituting cancellous bone with open-cell metal foam in biomechanical testing of cemented hip joints. 7th International Conference on Porous Metals and Metallic Foams (MetFoam), Busan, Korea.

Tozzi, G., Heaton-Adegbile, P., Tong, J., 2011. Long-term assessment of cement fixation in acetabular replacements under physiological testing in a biological environment. 24th International Society for Technology in Arthroplasty Congress (ISTA), Bruges, Belgium.

Tozzi, G., Zhang, Q.-H., Tong, J., 2011. Micromechanical behaviour in bone-cement interface under compression. 24th International Society for Technology in Arthroplasty Congress (ISTA), Bruges, Belgium.